

Antiferromagnetic Materials: Characterisation techniques

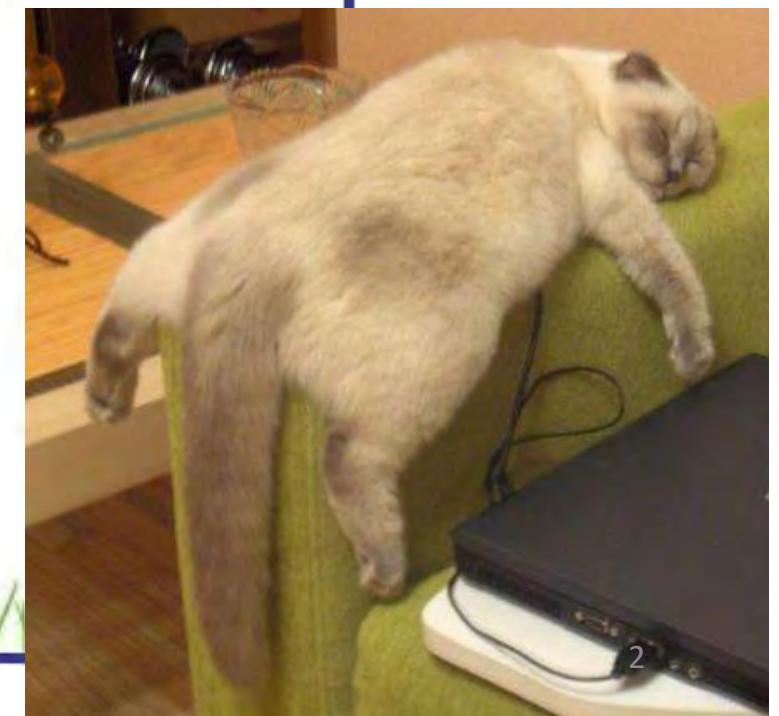
Bryan Gallagher, Nottingham University

Experimental techniques which can be used to characterize AF materials.

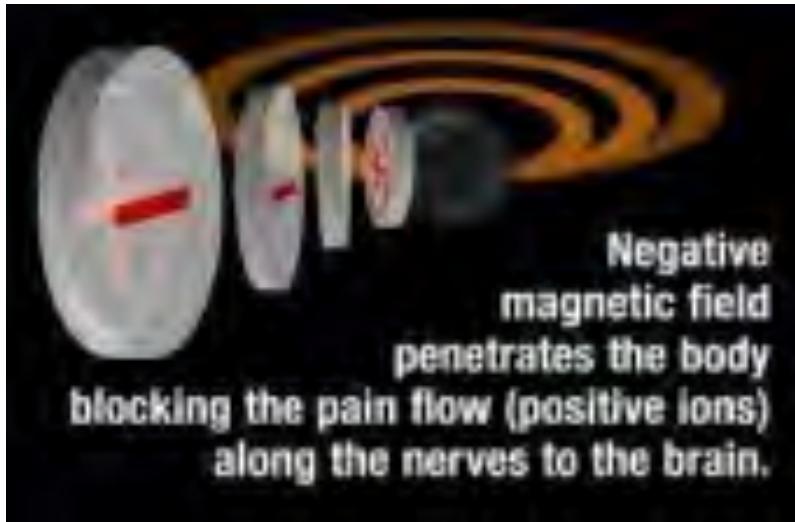
*Exclusive Jewellery
for Pets- Combined
with the power
of Magnets!*



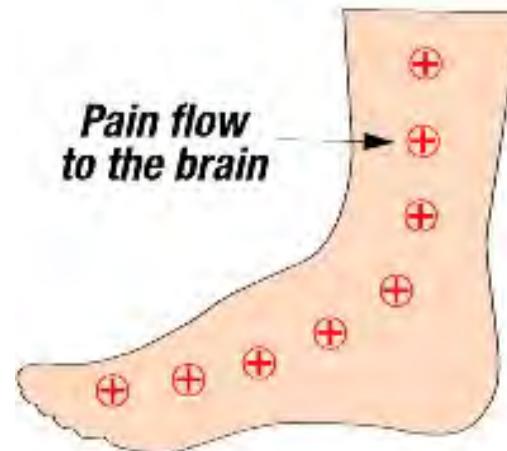
MagnetAnimal
Energise your pet



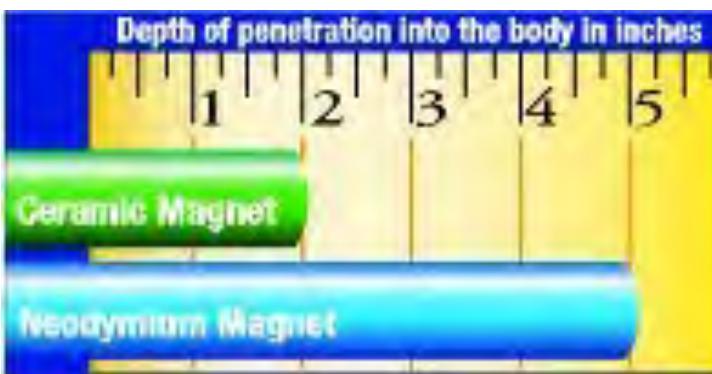
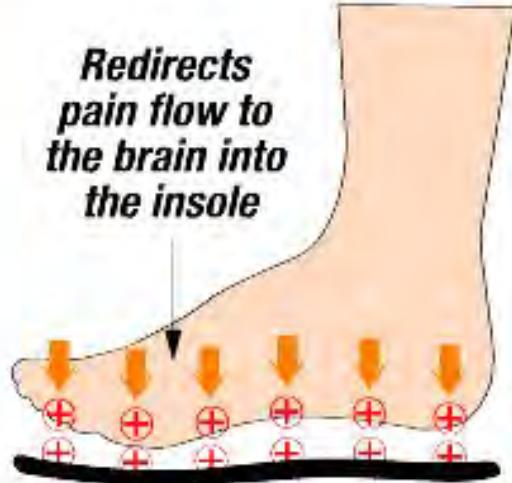
Dr. Bakst Magnetic Shoe Inserts



BEFORE



AFTER



Talk Contents

1. Materials Growth

2.1 Structural Characterisation: X-ray Scattering

2.2 Structural Characterisation: Transmission Electron Microscopy

3.1 Magnetic Characterisation: Neutron Scattering

3.2 Magnetic Characterisation: Magnons

3.3 Magnetic Characterisation: Magnetic Resonance

3.4 Magnetic Characterisation: Remanence

3.5 Magnetic Characterisation: Susceptibility

3.6 Magnetic Characterisation: “Large” Fields

3.7 Magnetic Characterisation: ZFNMR

4.1 Critical Behaviour : Specific heat

4.2 Critical Behaviour: Resistivity

4.3 Critical Exponents

5. X-ray magnetic dichroism

6. Spin Polarised Scanning Tunnelling Microscopy

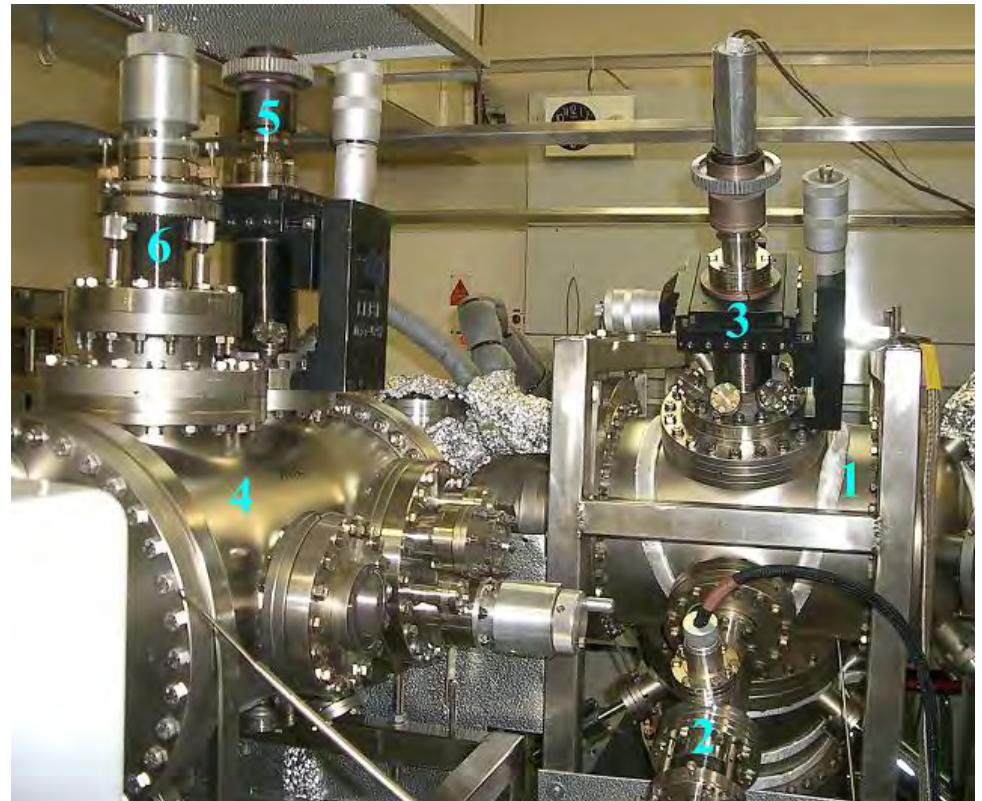
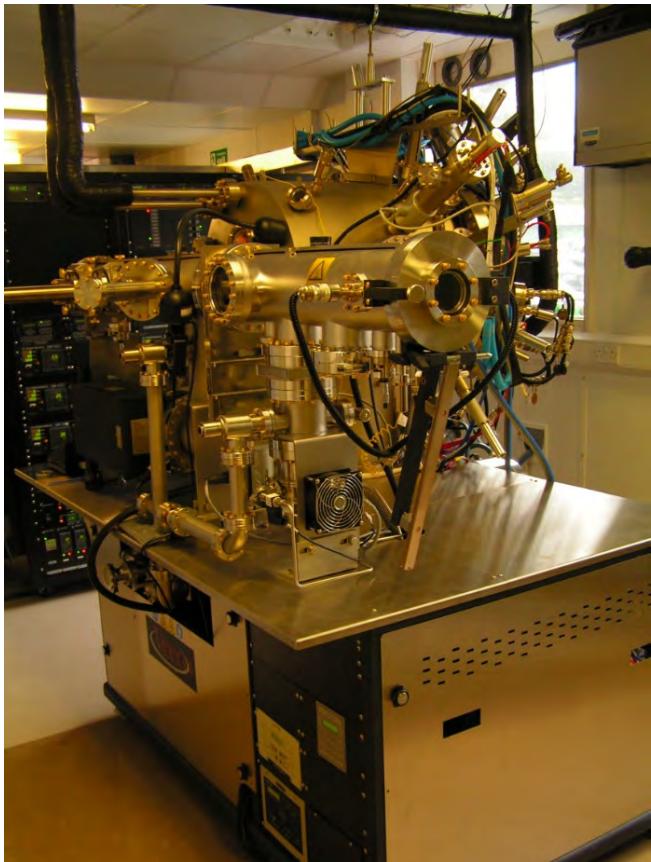
7.1 Electrical Transport: Anisotropic Magnetoresistance

7.2 Electrical Transport: Tunnelling Anisotropic Magnetoresistance

1. Materials Growth

Molecular Beam Epitaxy (MBE)

Richard Campion 9.30 Friday



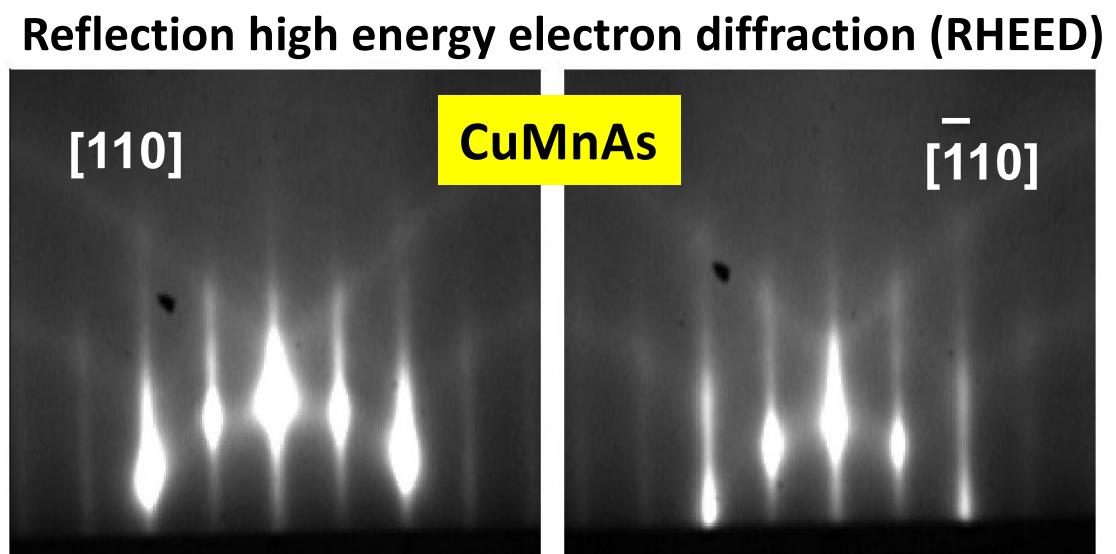
Molecular Beam Epitaxy

Can achieve (with regular calibrations)

- Very good epitaxy
- Very good control of stoichiometry
- Accurate growth temperature control
- Monolayer by monolayer growth
- Abrupt interfaces

Very slow
Very expensive

In situ monitoring of crystal structure

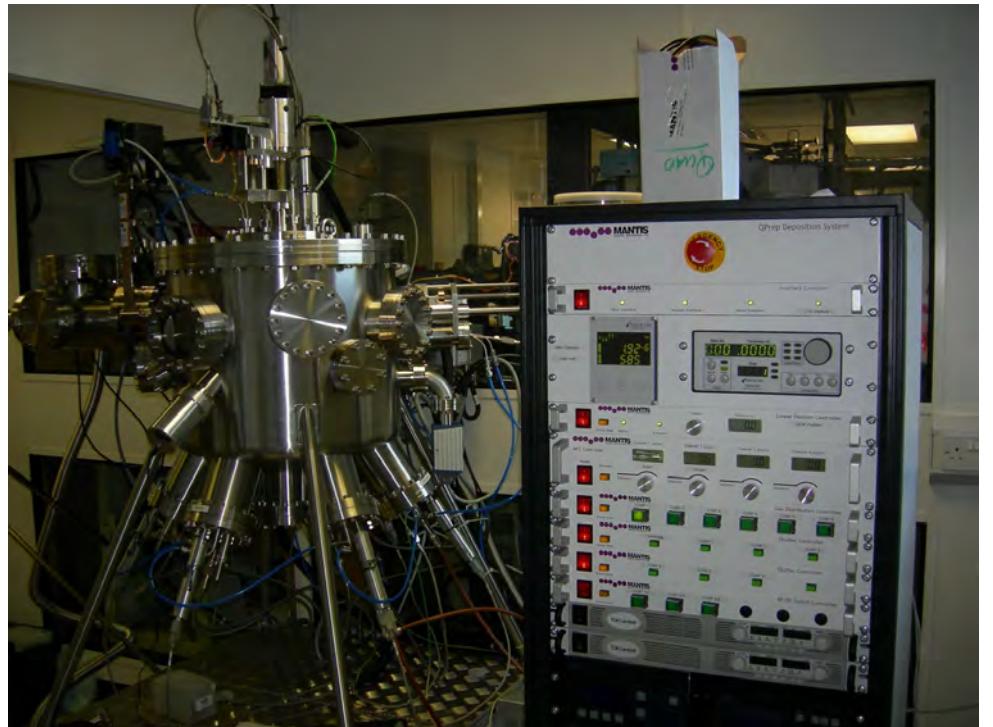


Good for materials structurally compatible with commercial semiconductors ⁶

Limited number of elements in a specific machine

Sputter Deposition Facilities

- Very flexible
- Relatively fast and low cost
- Can achieve close to epitaxial growth
- Very wide range of elements
- Can co-deposit from multiple sources
- Can sputter in reactive gas (O_2) or from composite targets (e.g. MgO)



Good to have access to MBE and sputtering

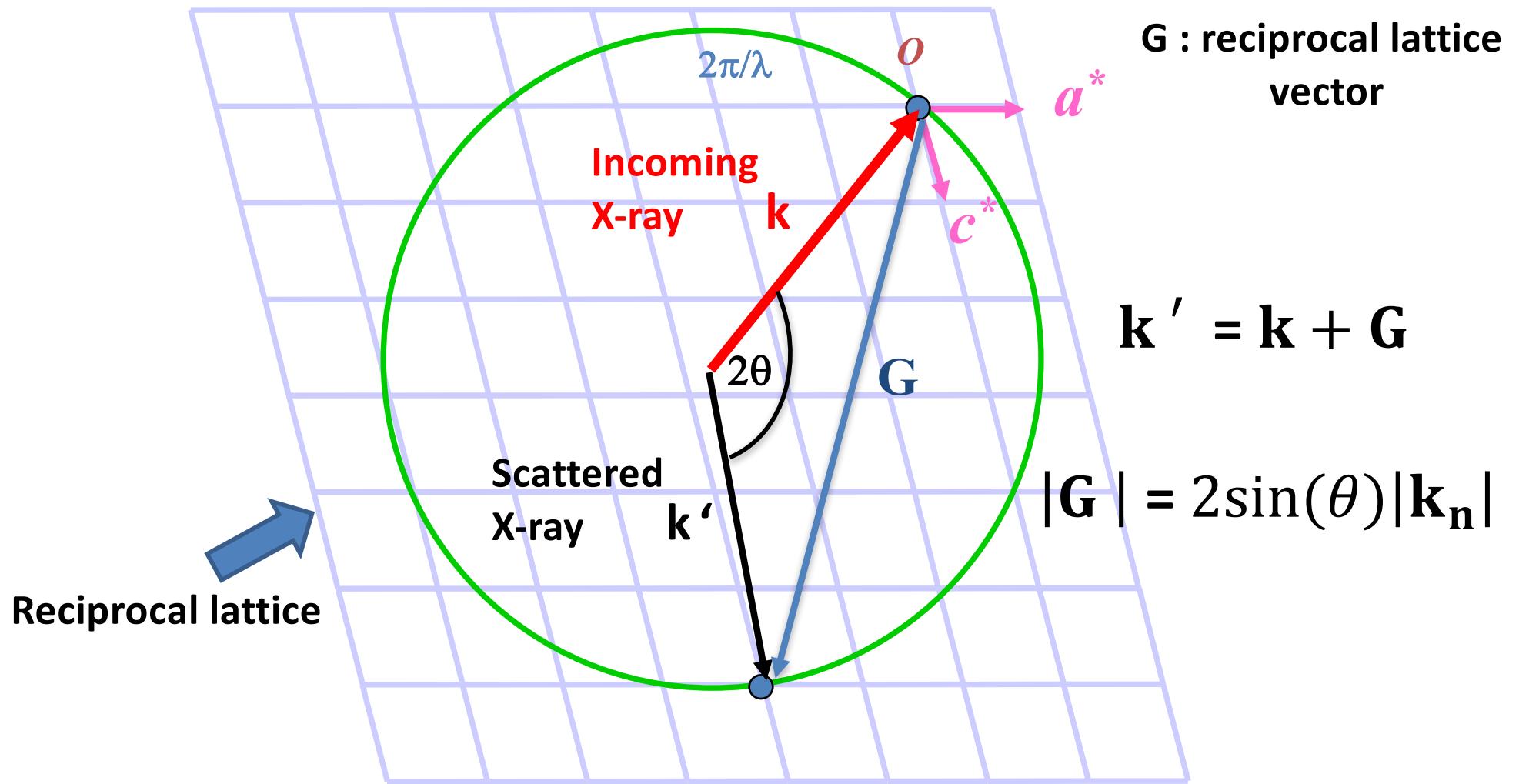
2.1 Structural Characterisation: X-Ray Diffraction (XRD)

- X-ray diffraction can reveal crystallographic structure.
- Want x-ray wavevector, $k \approx$ reciprocal lattice vector, $G \approx 10^{10} \text{ m}^{-1}$.
- For $k = 10^{10} \text{ m}^{-1}$ x-ray energy is $\epsilon = 21 \text{ keV}$.
- Scattering almost elastic since $\epsilon \gg$ phonons and magnons energies (~25meV at room temperature).



Phillips X-Pert high-resolution X-ray diffractometer

Elastic X-Ray Scattering



Bragg's Law in Reciprocal Space (Ewald Sphere)

X-ray Scattering Amplitude

$$A(\mathbf{Q}) \propto \int \rho(\mathbf{r}) \exp[i\mathbf{Q} \cdot \mathbf{r}] d\mathbf{r} \propto \sum_j f_j(\mathbf{Q}) \exp[i\mathbf{Q} \cdot \mathbf{r}_j] \sum_T \exp[i\mathbf{Q} \cdot \mathbf{r}_j]$$

Q scattering vector $\rho(r)$: electron charge density $f_j(Q)$ atomic structure factor
 T translation vectors

Inverse problem: infer real space structure from reciprocal space structure

Peak Positions



Translational symmetries

Atom positions in basis

Peak Intensities



Atom types in basis

Order / disorder

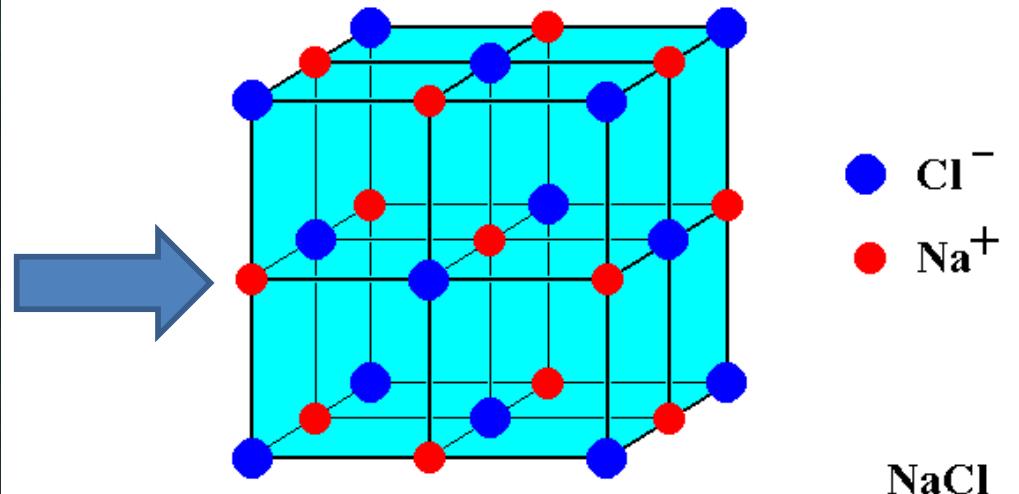
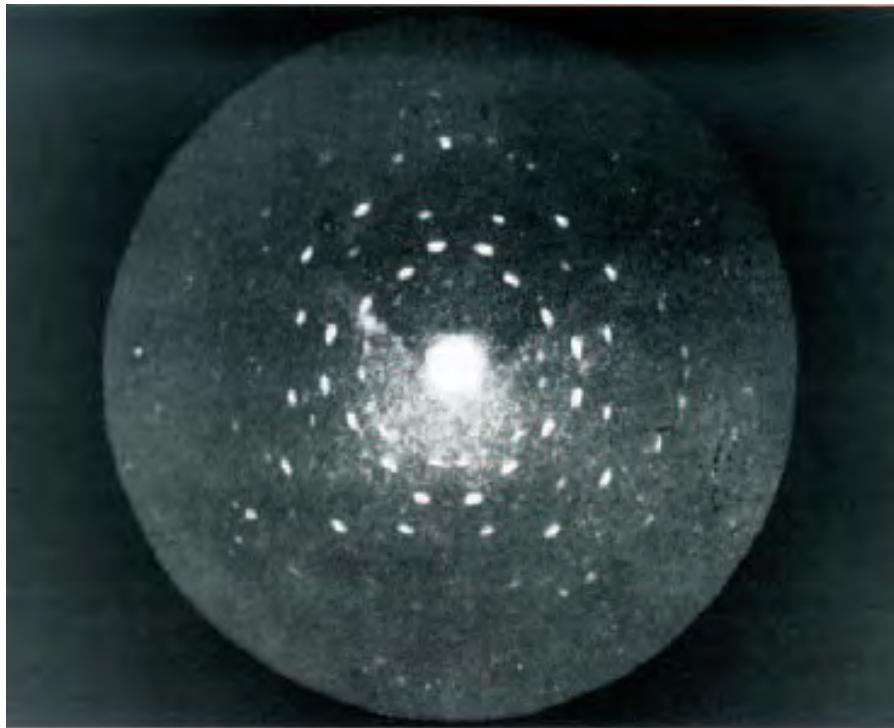
Peak Widths



Extent of periodicity

crystallite size / film thickness

X-ray diffraction from NaCl crystal

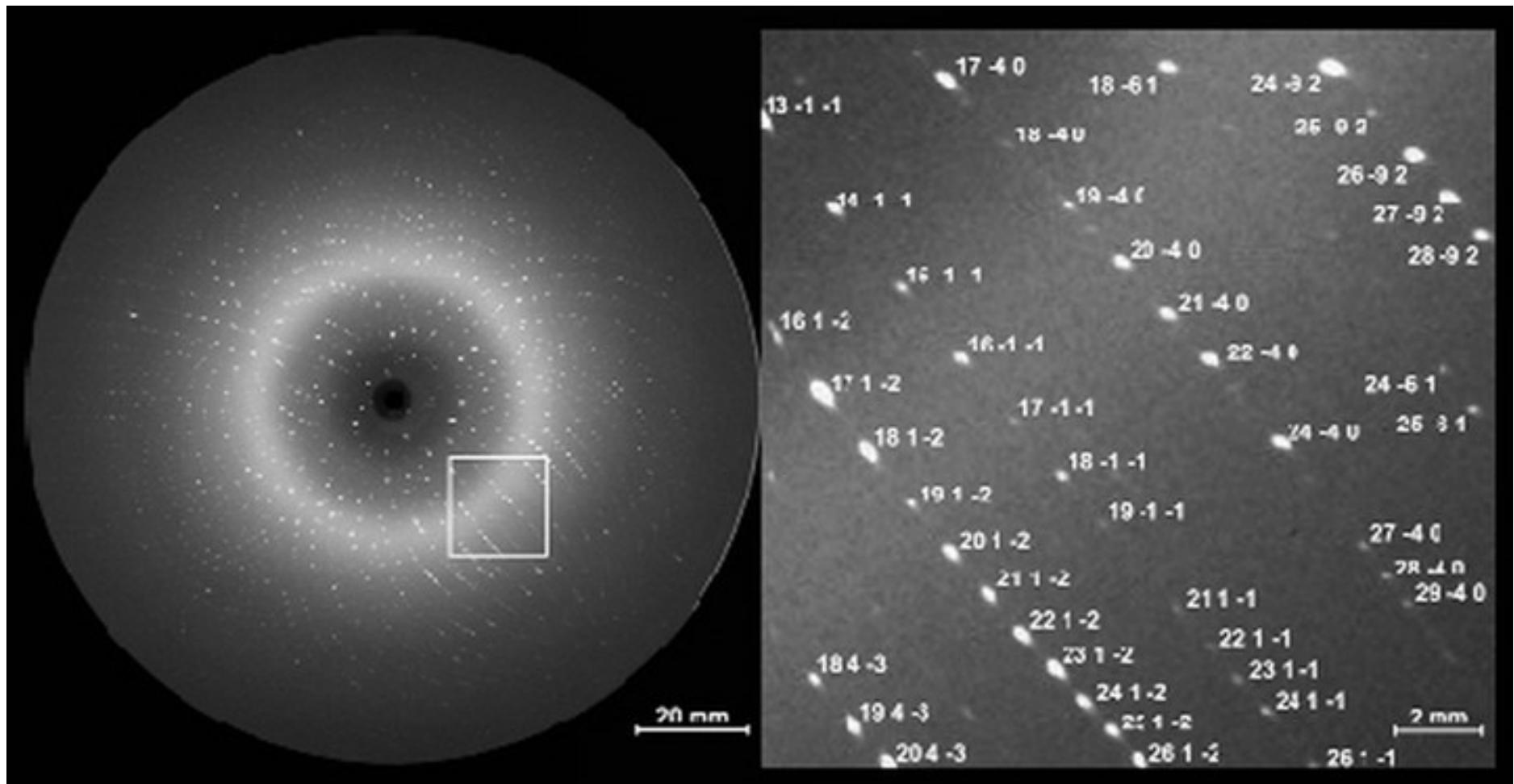


One of the first diffraction photographs taken M. von Laue in 1912

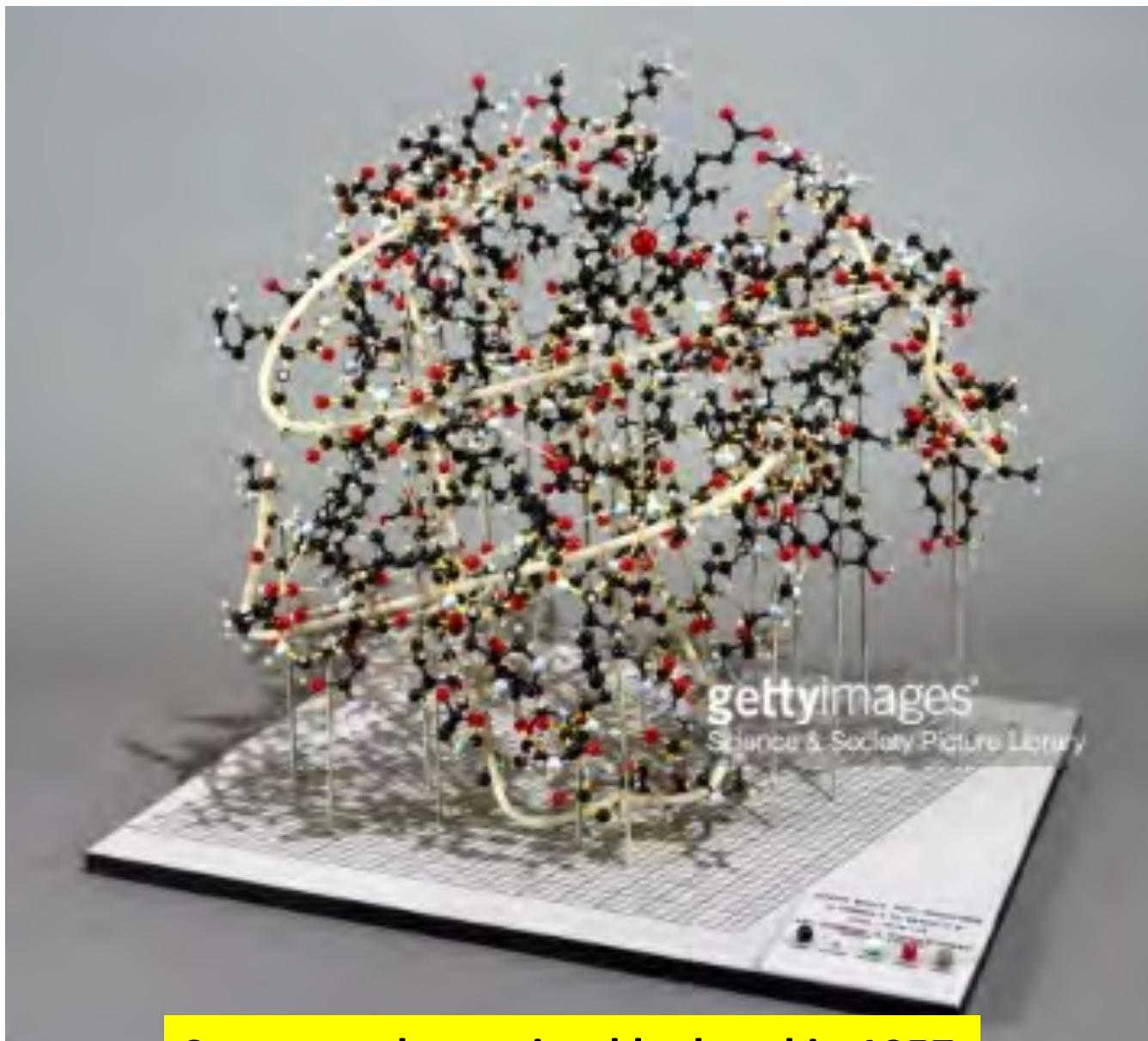
Crystal structure determined from Laue pattern by W. H. and W. L. Bragg in 1913

X-ray diffraction from protein Myoglobin

Image: ~3000 diffraction spots → positions of ~ 3000 atoms in protein



Myoglobin

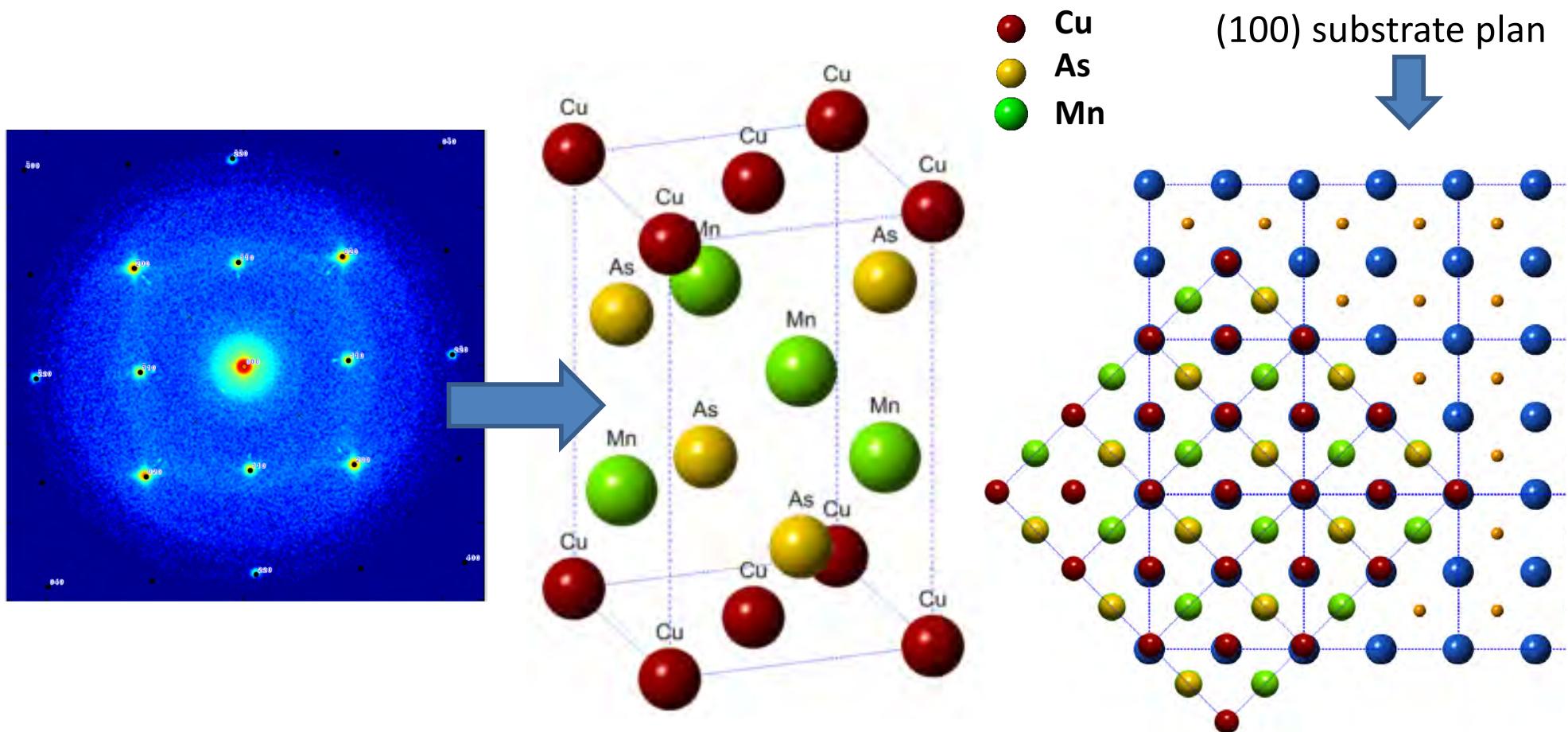


Structure determined by hand in 1957

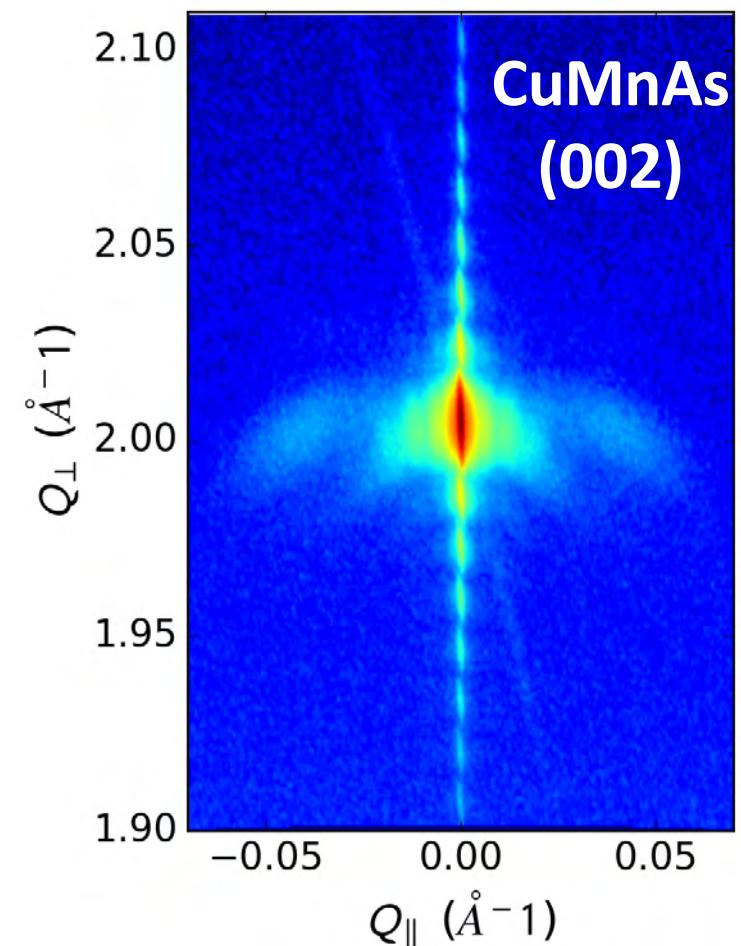
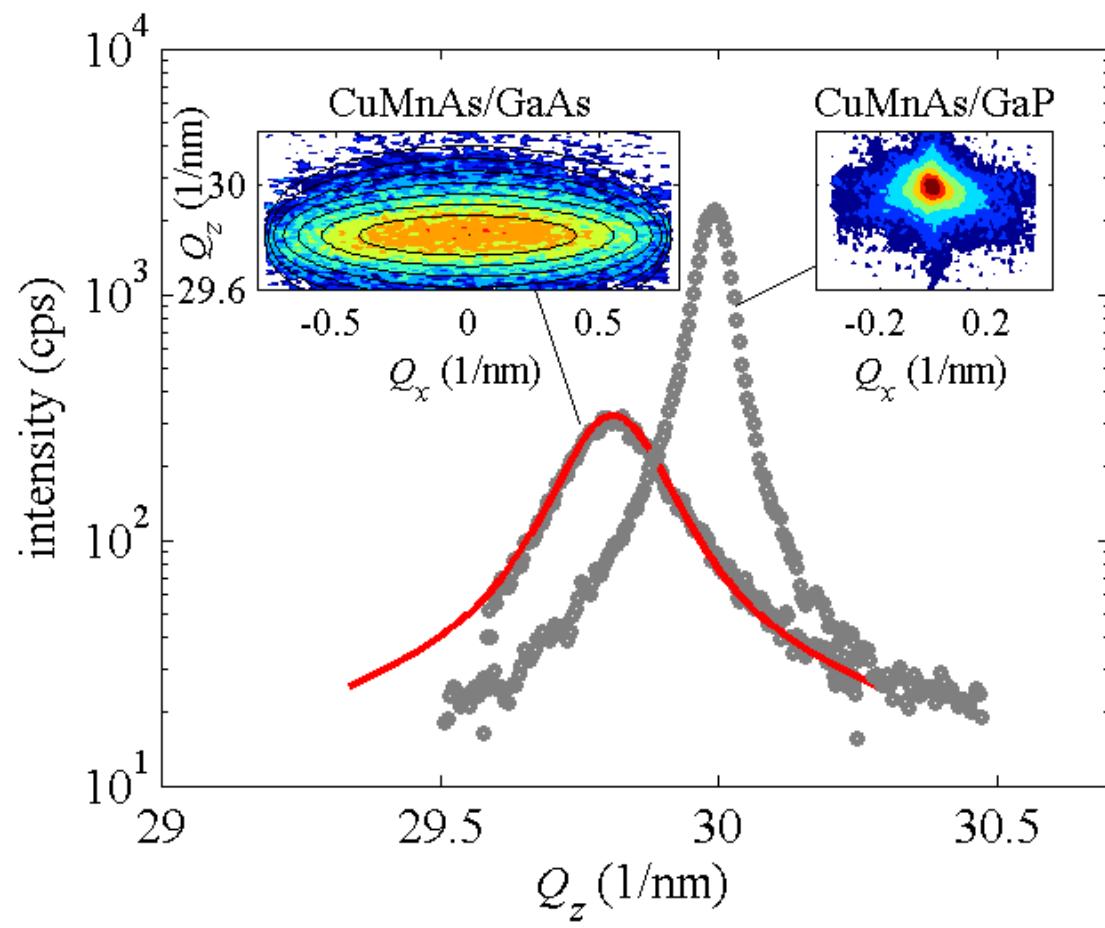
AF CuMnAs Grown on (100) GaAs or GaP

Structure resolution: Cu₂Sb type structure Tetragonal P4/nmm

$$a = b = 3.820 \text{ \AA} \quad c = 6.318 \text{ \AA}$$

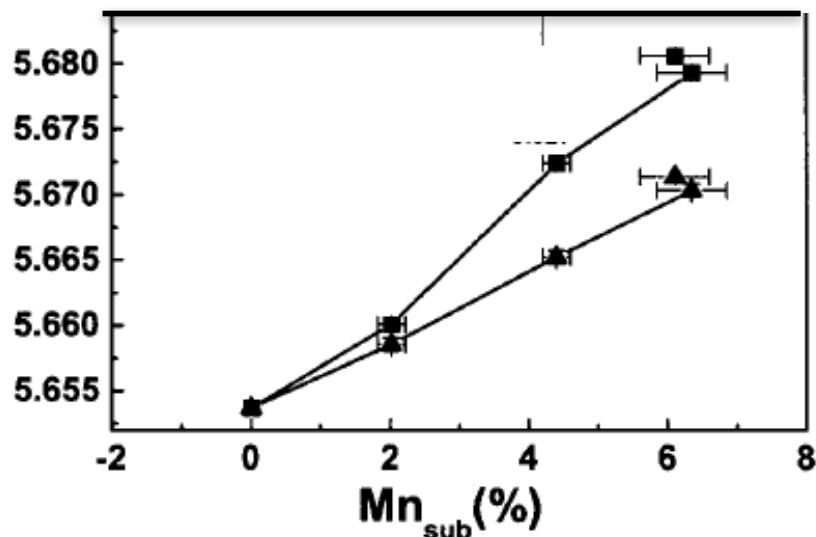
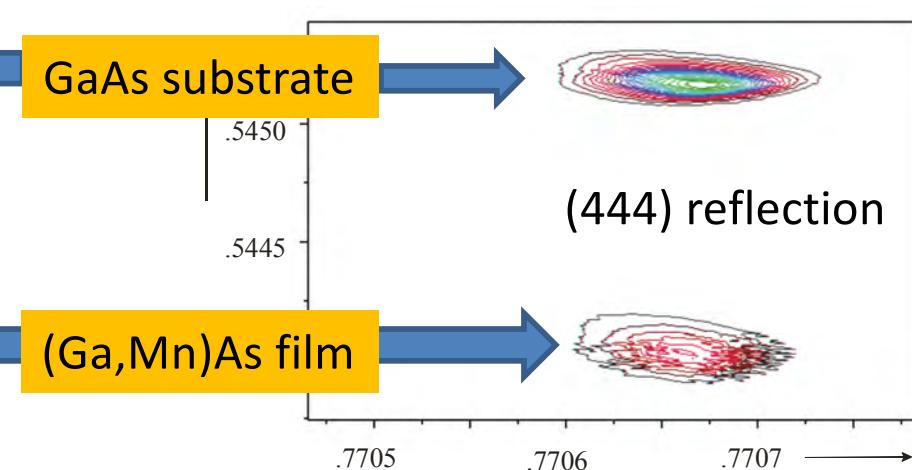
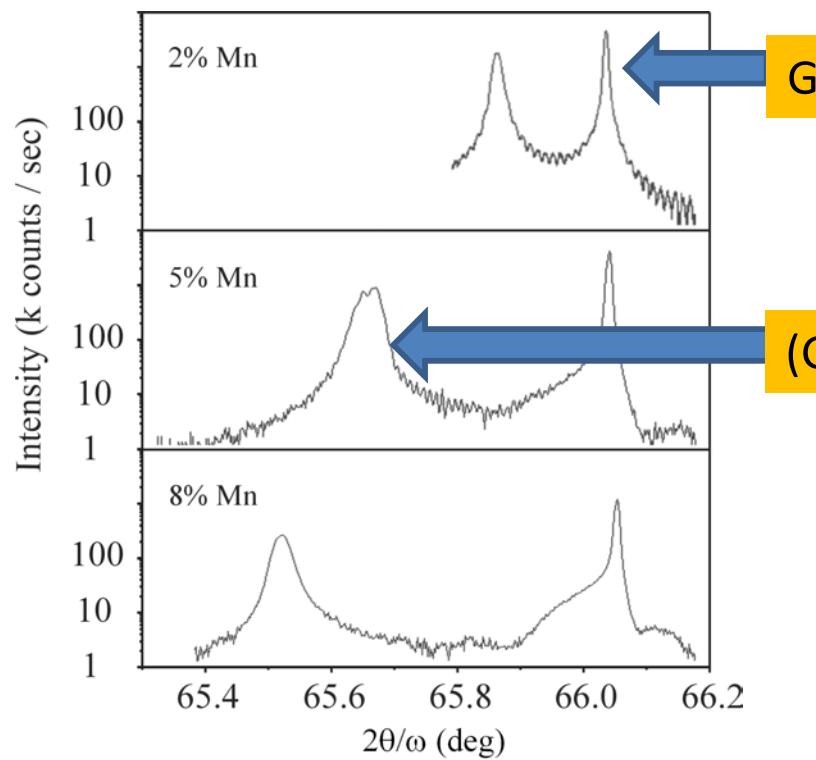


Crystal quality of CuMnAs



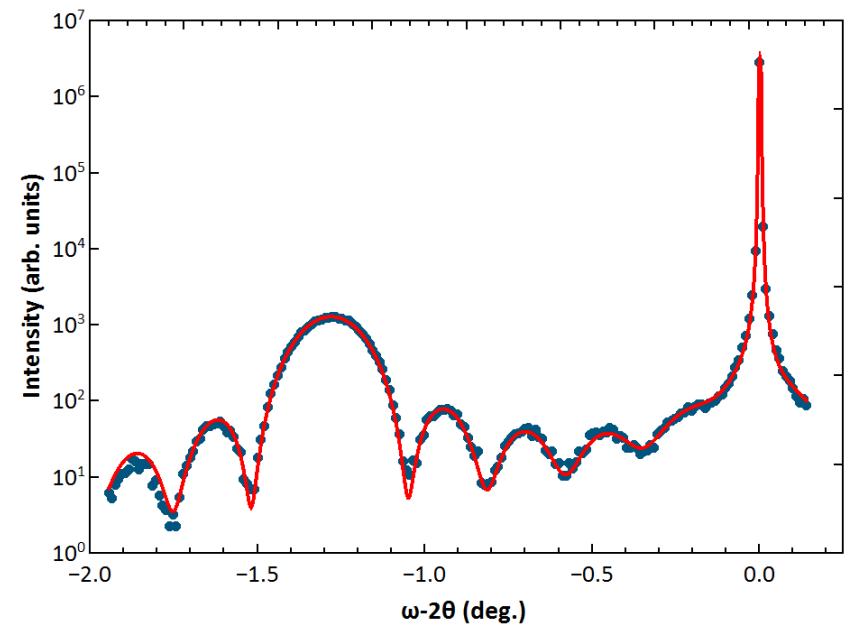
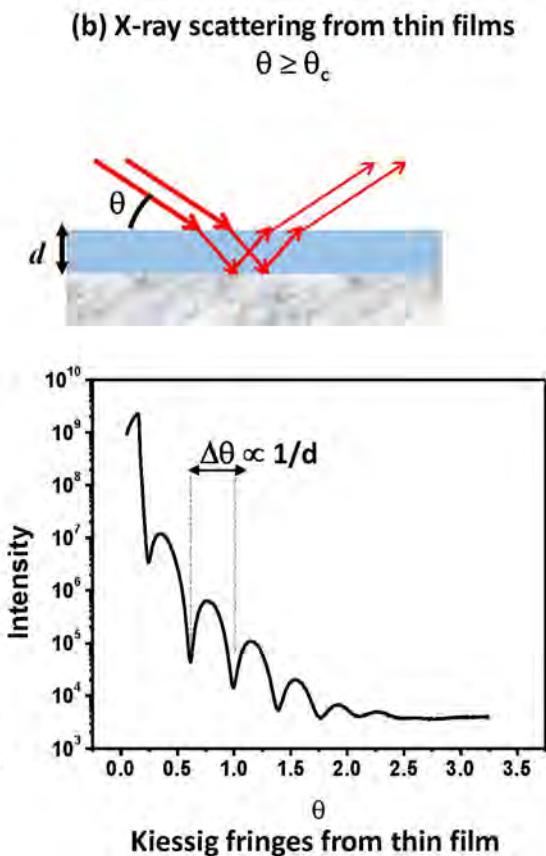
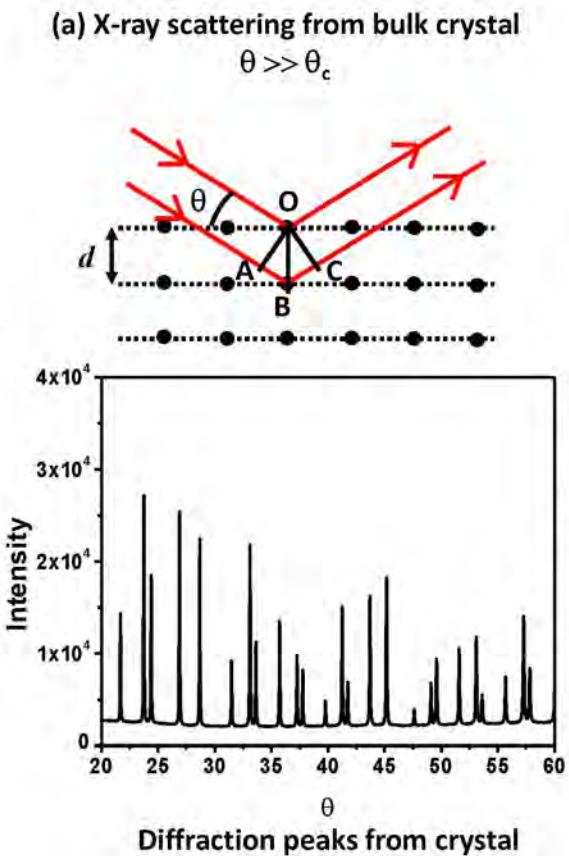
- **Narrower peaks** on better lattice matched GaP substrate: **Higher quality**
- Relaxed mosaic block structure on GaAs
- Fully strained on GaP

FM (Ga,Mn)As Grown on GaAs



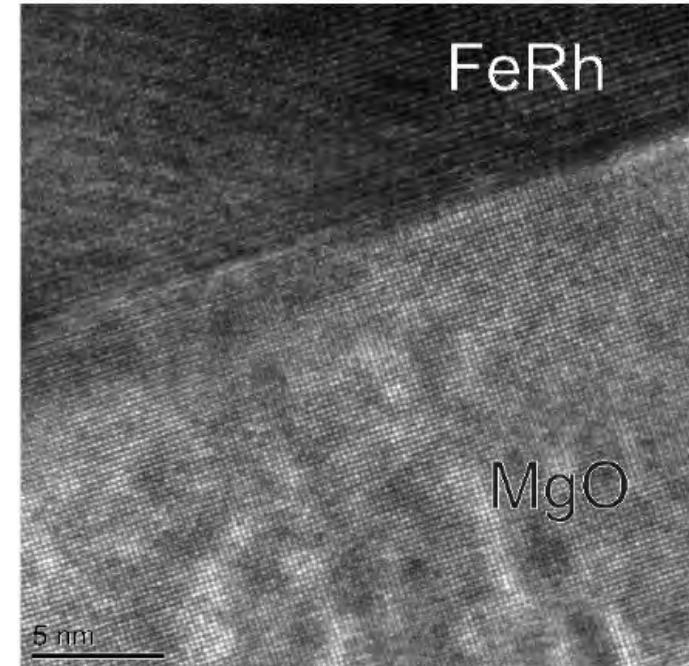
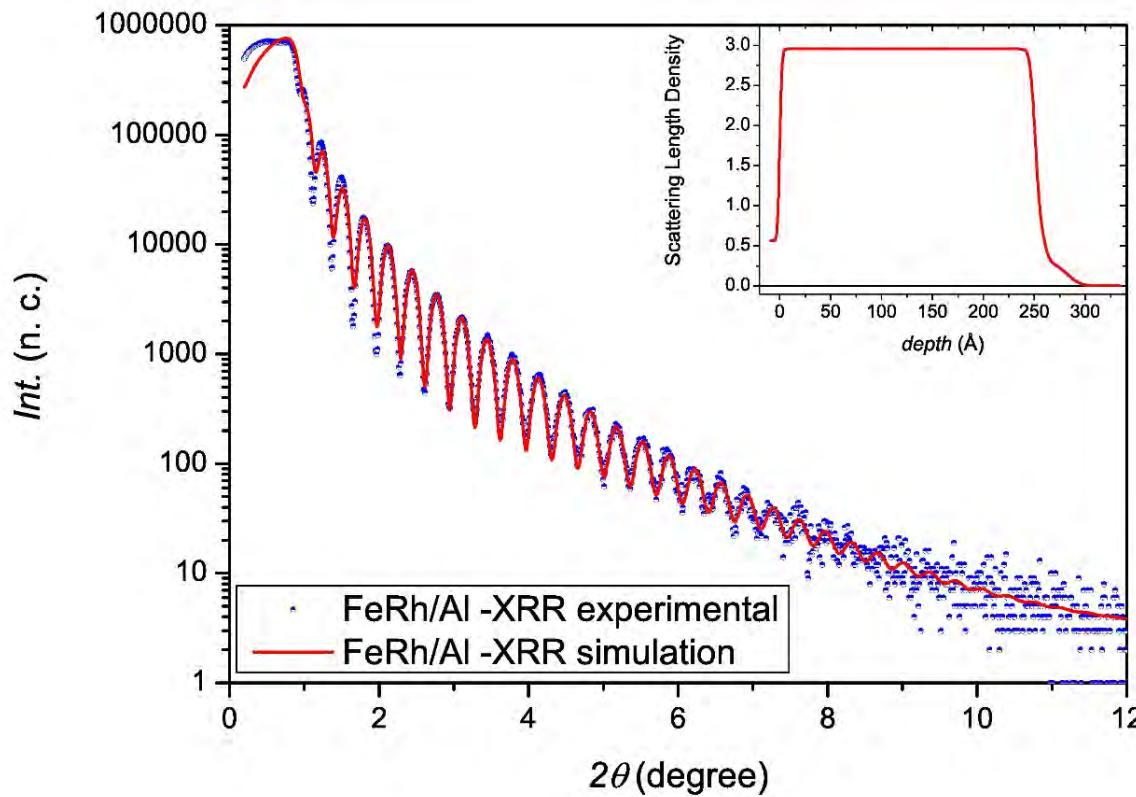
Expansion of the out of plane lattice constant with Mn concentration (as grown and annealed samples)

Kiessig fringes from thin films



Symmetric 004 reflection from 22.5 nm epitaxial film of $\text{Si}_{1-x}\text{Ge}_x$ with $x = 49\%$ on a Si(001) substrate

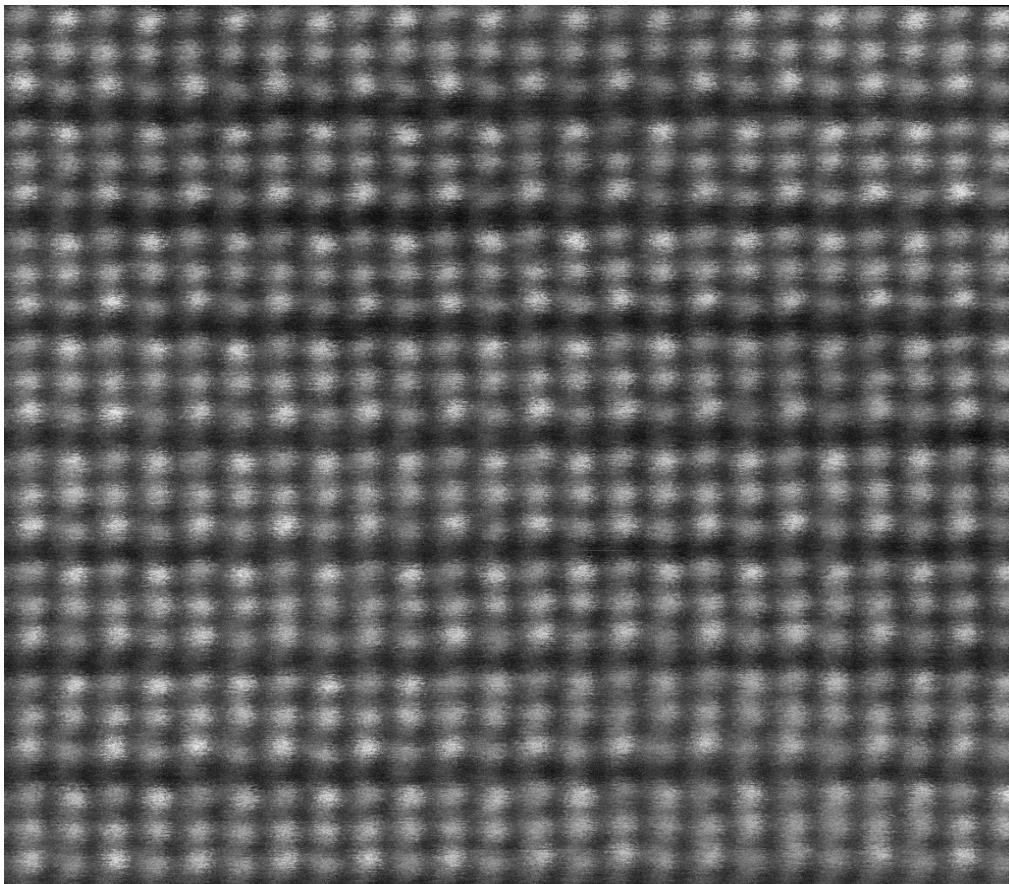
FM / AF FeRh thin film



X-ray reflectometry spectrum from a 25 nm thick FeRh epilayers capped with polycrystalline Al. The solid line is a fit using the FeRh density profile of the insert.

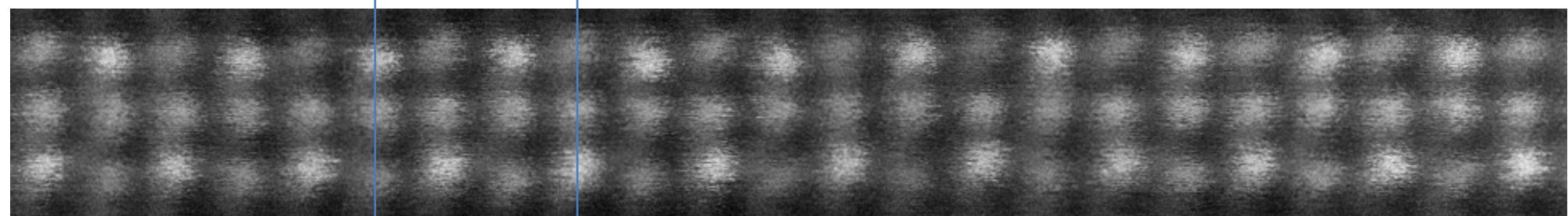
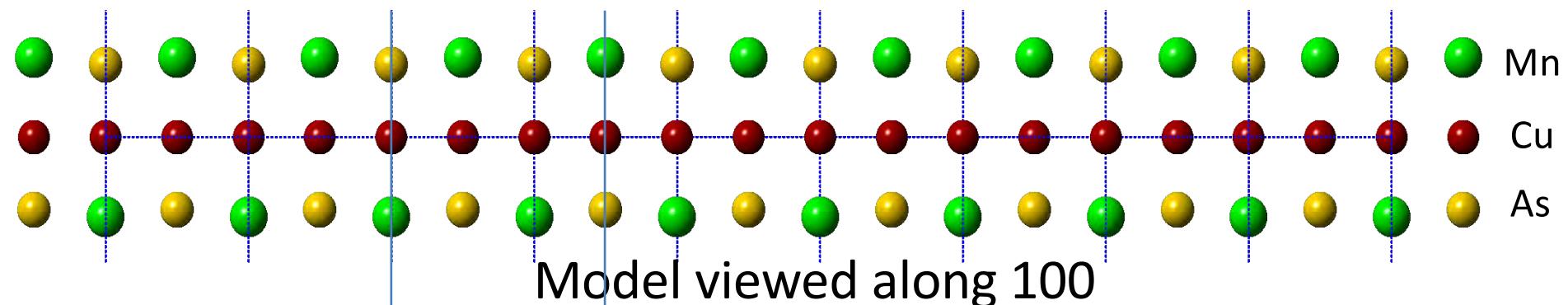
3.2 Structural Characterisation: Transmission Electron Microscopy (TEM)

Element (Z)-resolved TEM image of CuMnAs

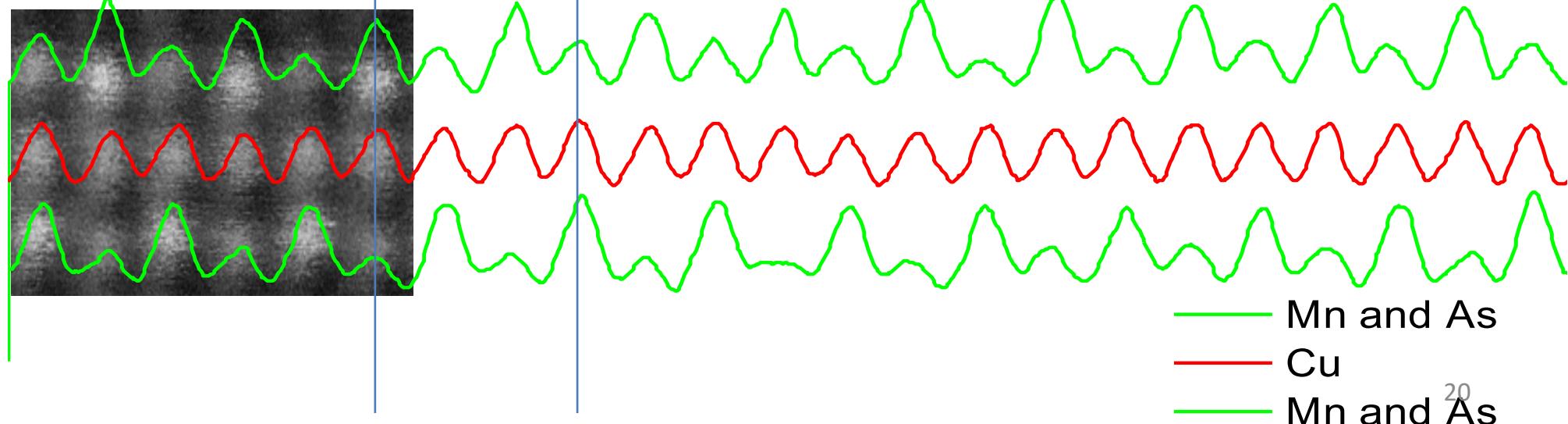


**Remember samples
~100 nm thick.**

CuMnAs TEM- Comparison to model

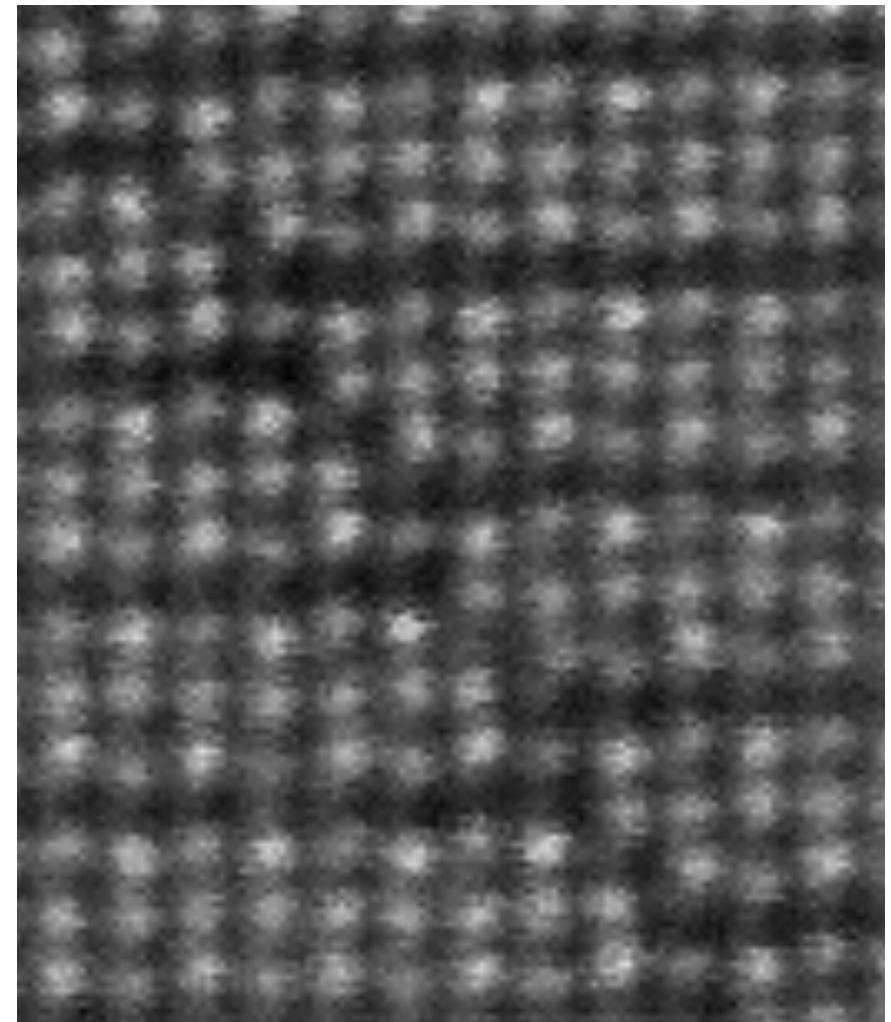
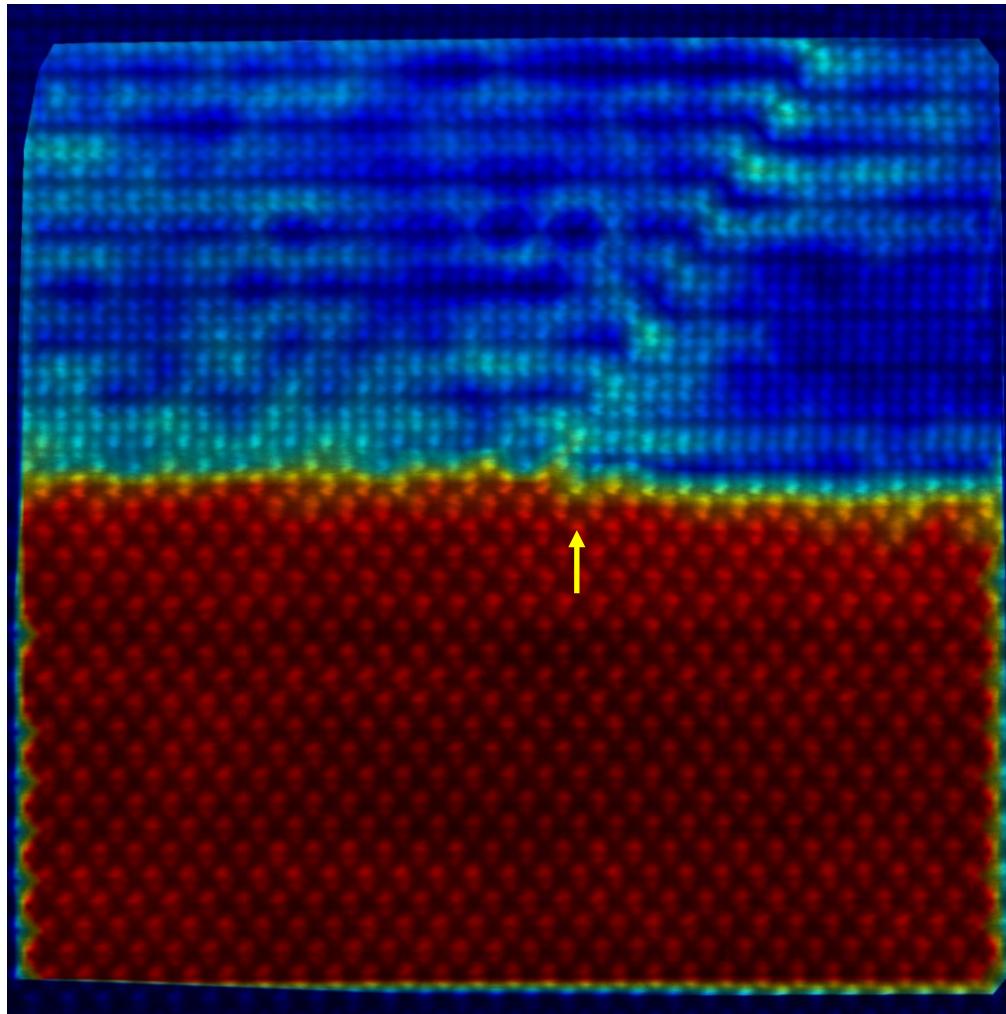


Intensity along rows



CuMnAs: Anti-phase boundary defects

Arise from steps in the substrate



3.1 Magnetic Characterisation: Neutron Scattering

- **Elastic** neutron scattering can reveal magnetic structure.
- **Inelastic** neutron scattering can measure dispersion of phonons and magnons.
- Want neutron wavevector, $k_n \approx$ reciprocal lattice vector, $G \approx 10^{10} \text{ m}^{-1}$.
- For $k_n = 3 \times 10^{10} \text{ m}^{-1}$ neutron energy is $\epsilon_n = 25 \text{ meV} \approx kT @ T \approx 300 \text{ K}$ (Thermal neutrons).
- Thermal neutrons have energies \approx phonons and magnons.

X-ray Scattering

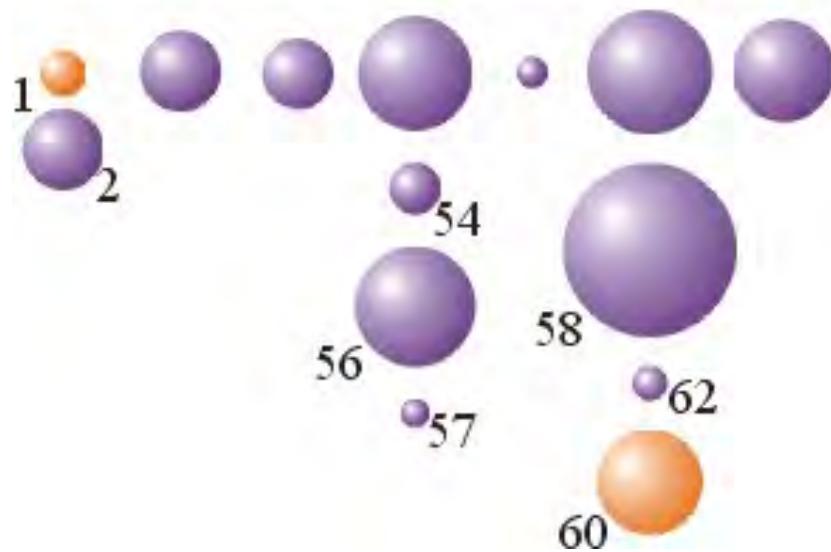
- X-rays scattered by electrons of atoms. Atomic structure factor $f(Q) \sim Z$, atomic number.
- Size of atoms \approx neutron wavelength: scattering factor $f(Q)$ decreases as $|Q|$ increases.



X-rays: Scattered intensity $\sim Z^2$. Hard to 'see' light elements.

Neutron Nuclear Scattering

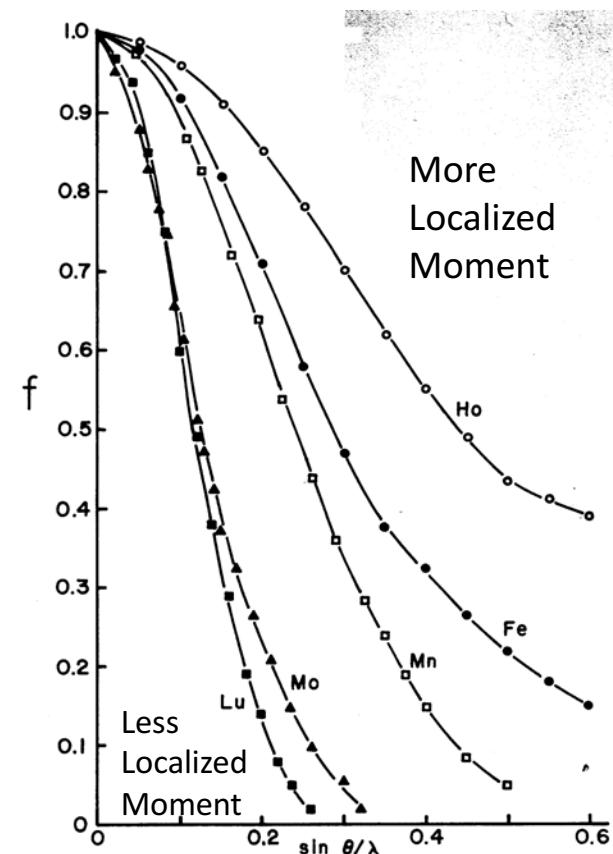
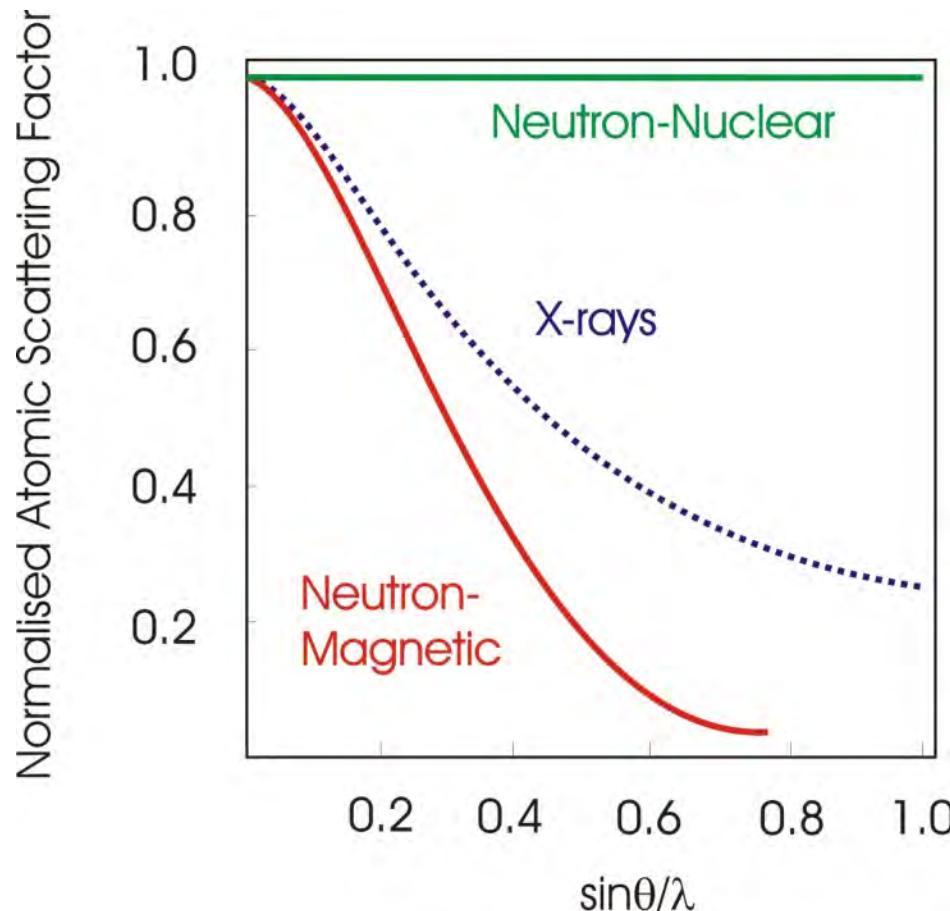
- Neutrons scattered by nuclei (strong force).
- Size of nuclei $\approx 10^{-5} \times$ neutron wavelength.
- Nuclear scattering factor (b) for neutrons is \approx independent of scattering wavevector q . Irregular variation with atomic number and isotope.



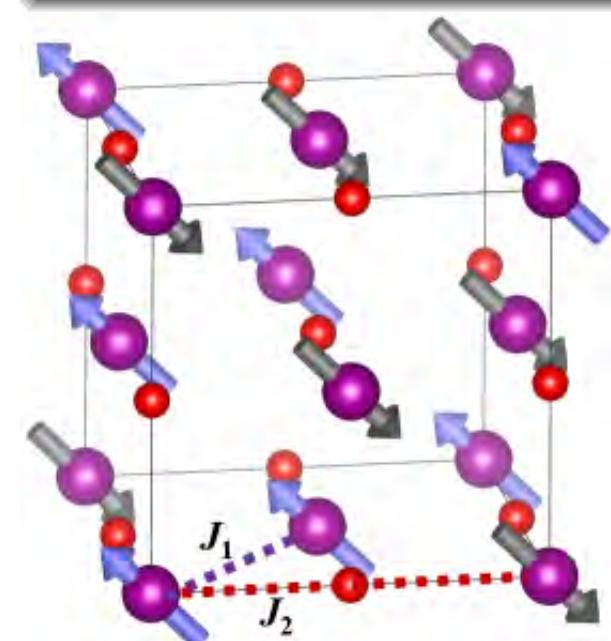
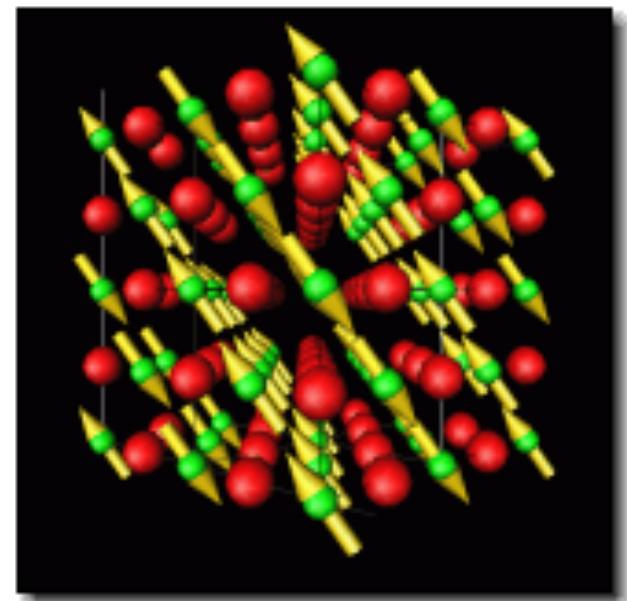
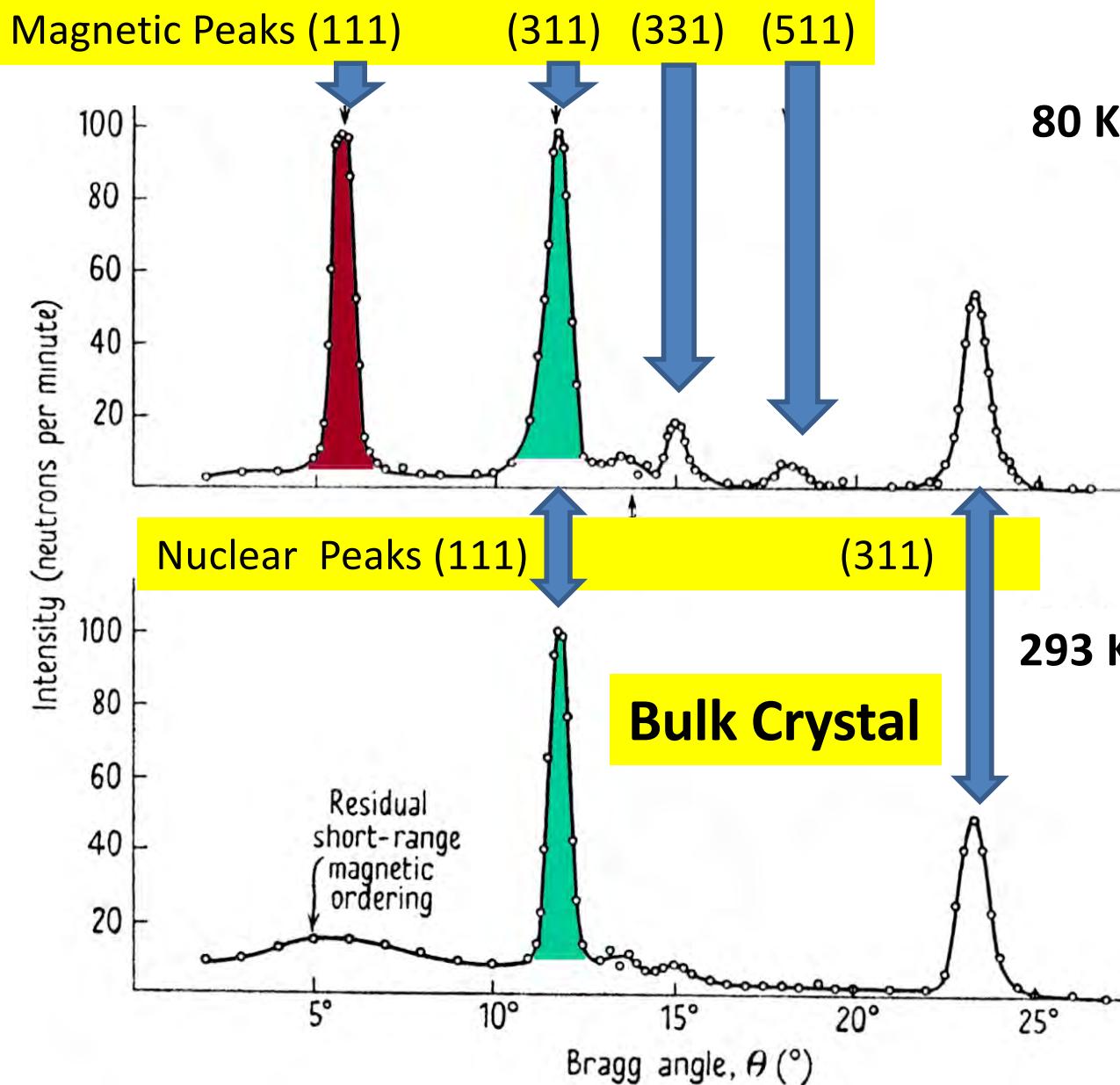
Neutrons: Sensitive to light elements. Isotope dependent: can vary contrast by isotope substitution.

Neutron Magnetic Scattering

- Neutrons scattered by magnetic moments.
- Spin and orbital moments of atom associated with outer electrons.
- Magnetic scattering factor for neutrons is strongly dependent on scattering wavevector Q.

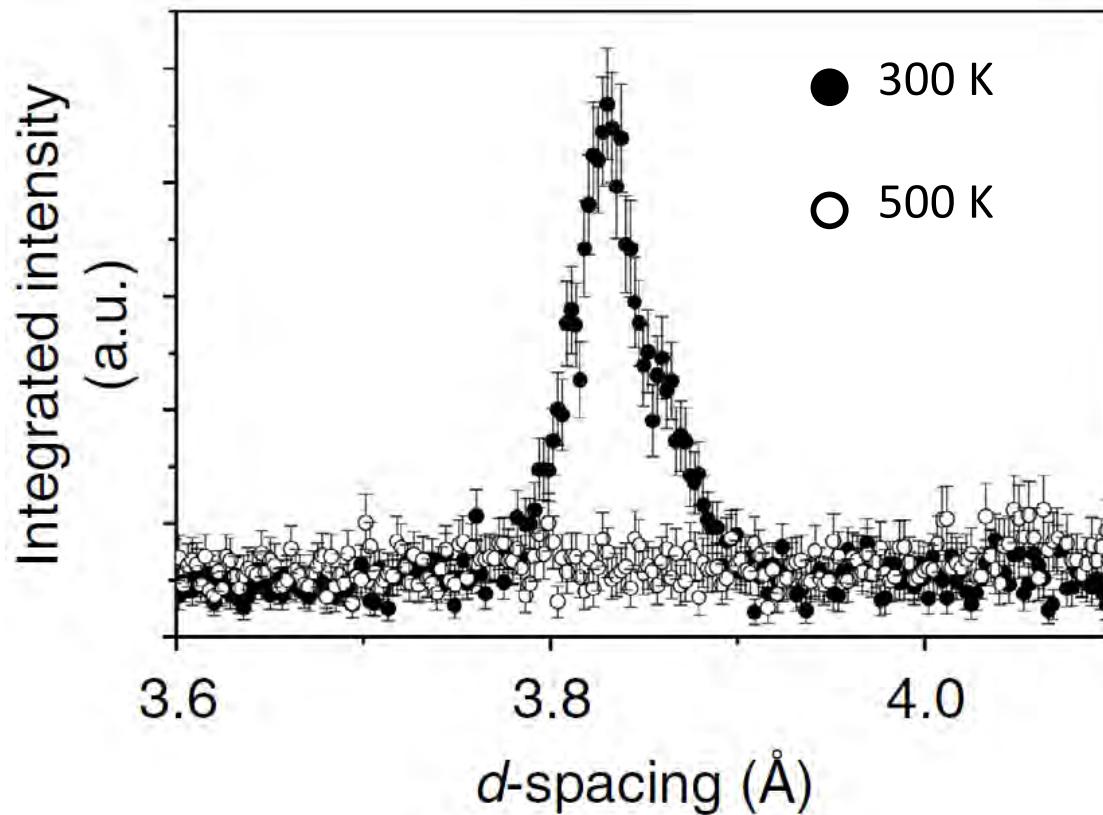


Neutron Diffraction: MnO ($T_N = 116$ K)



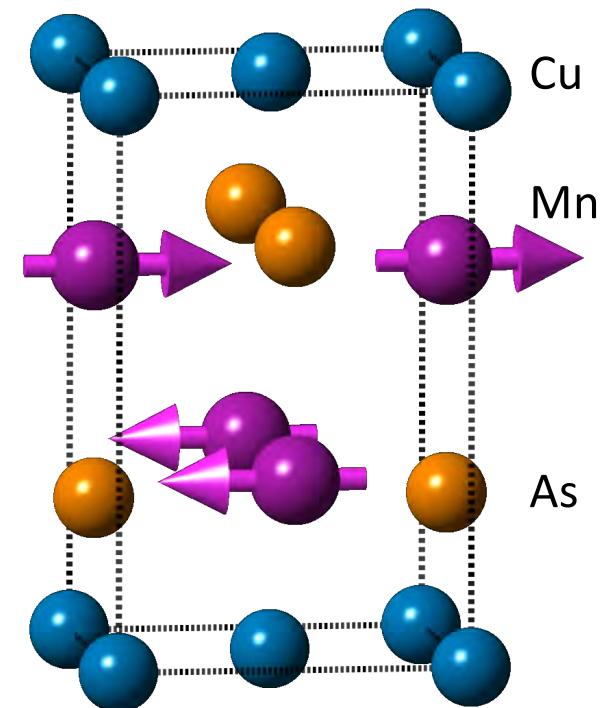
Neutron Diffraction: CuMnAs

Structurally forbidden in plane (100) diffraction peak.



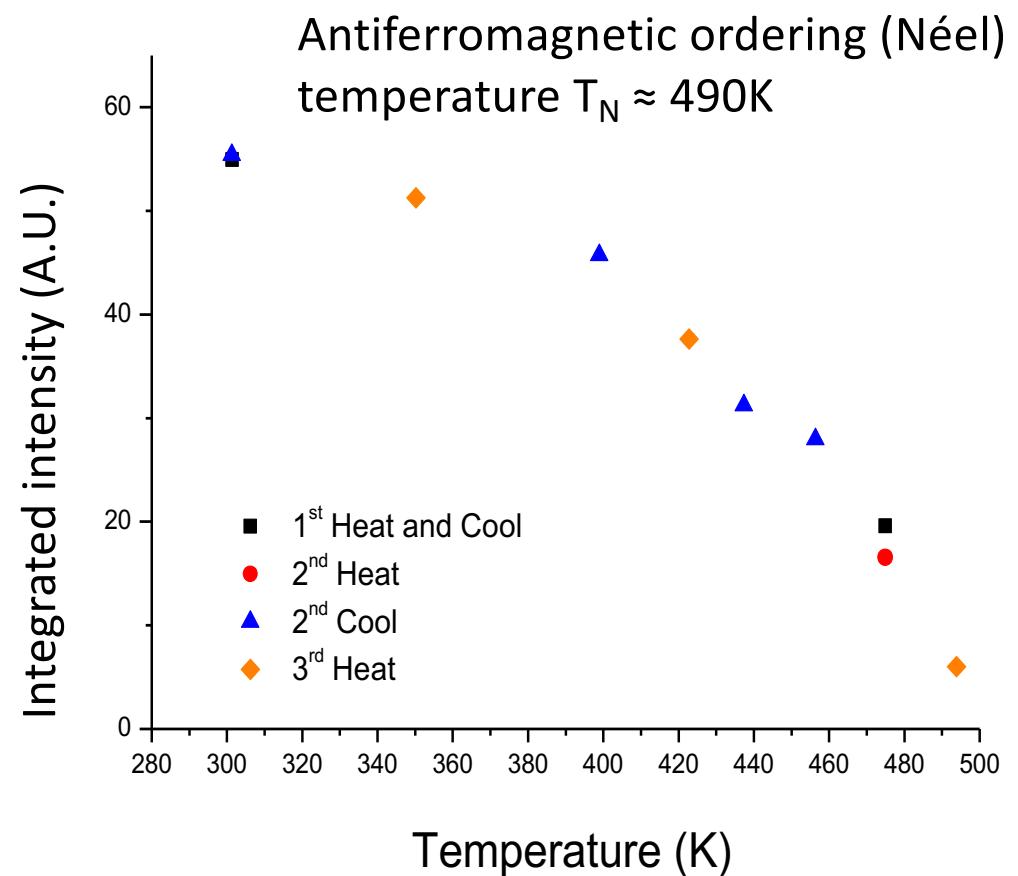
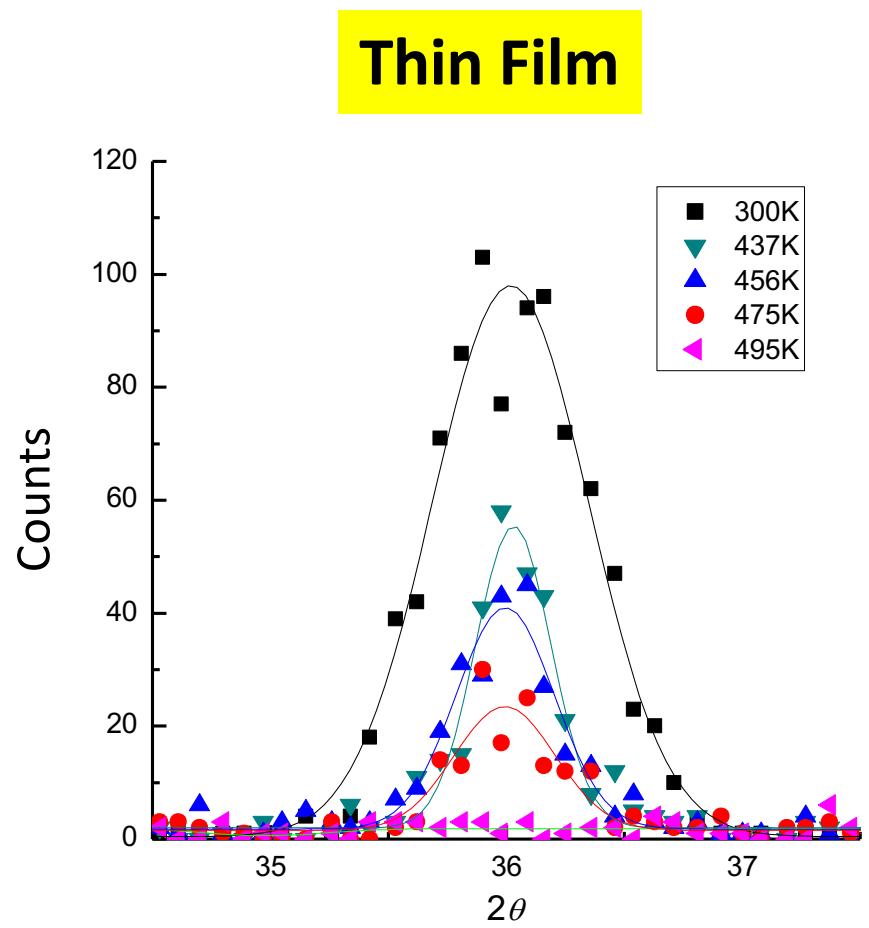
Thin Film

Structural peaks such as (002) as strong at 500K



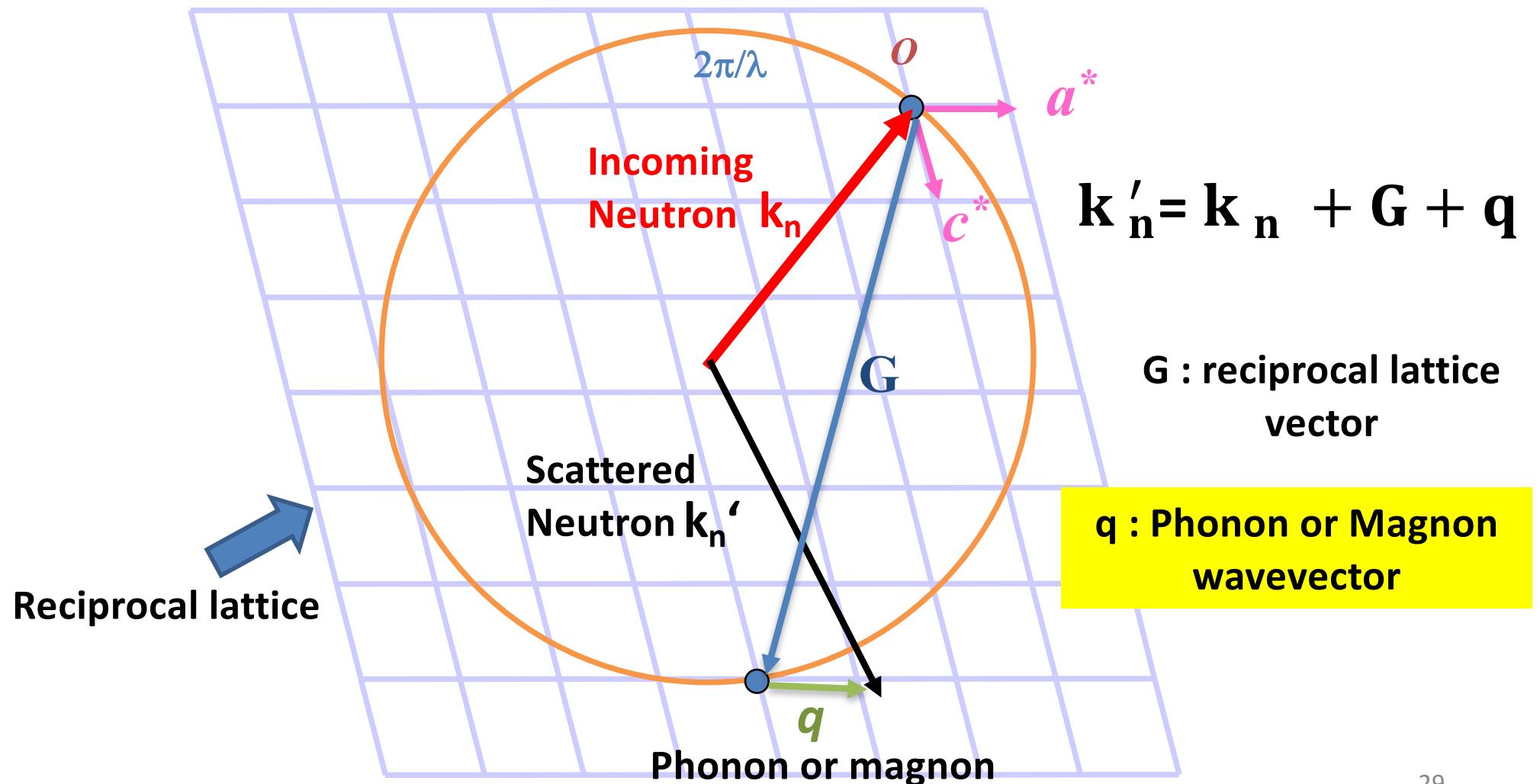
- AFM ordering with same dimensions as the structural unit cell
- Collinear layered antiferromagnet with spin axis in the ab plane
- Mn moment $\approx 3.6\mu_B$

Neutron Diffraction: CuMnAs

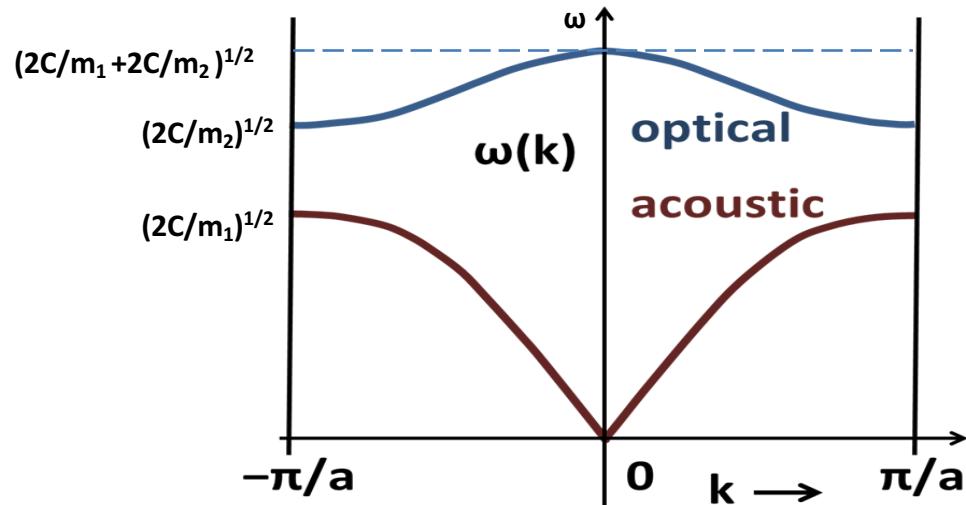
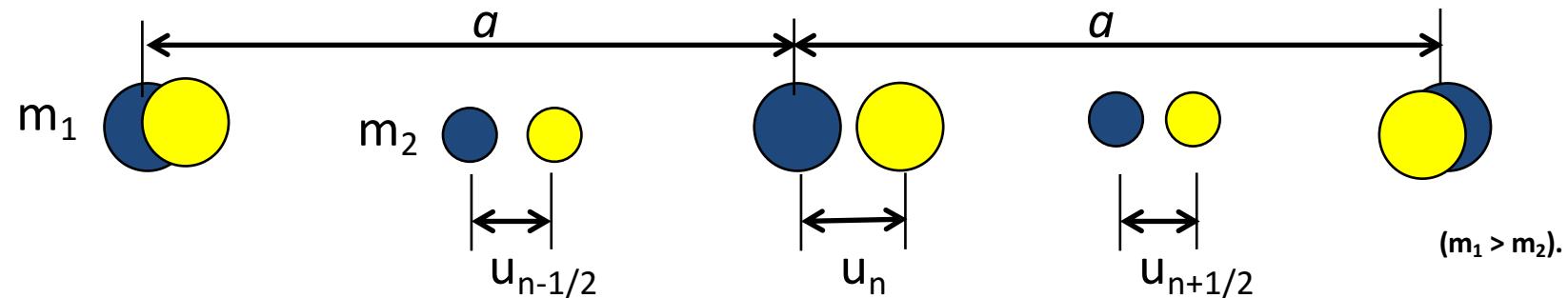


3.2 Magnetic Characterisation: Magnons

Inelastic Neutron Scattering

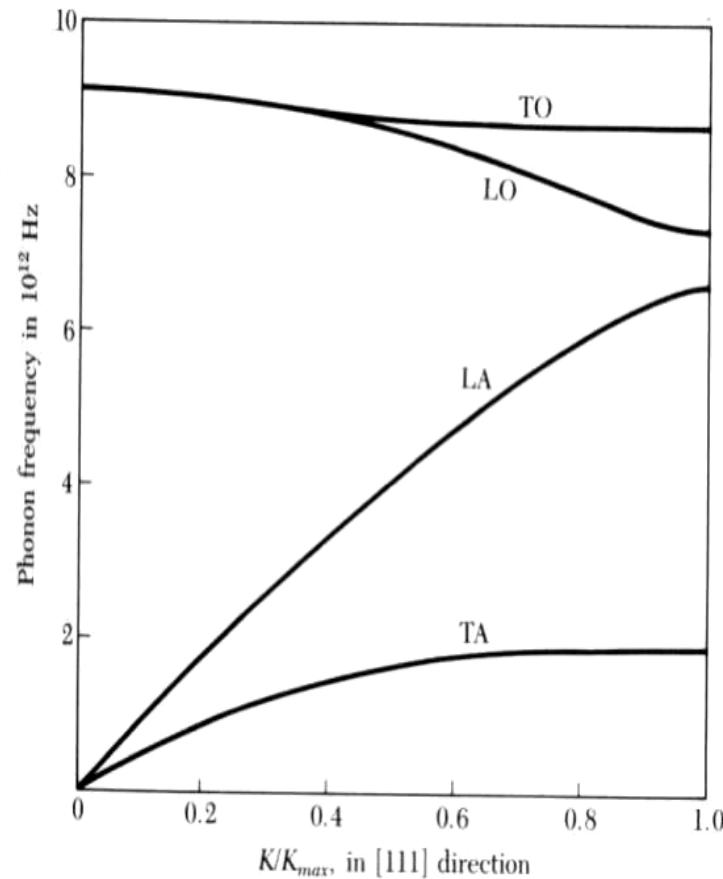


Phonons: 1D Harmonic Diatomic Chain

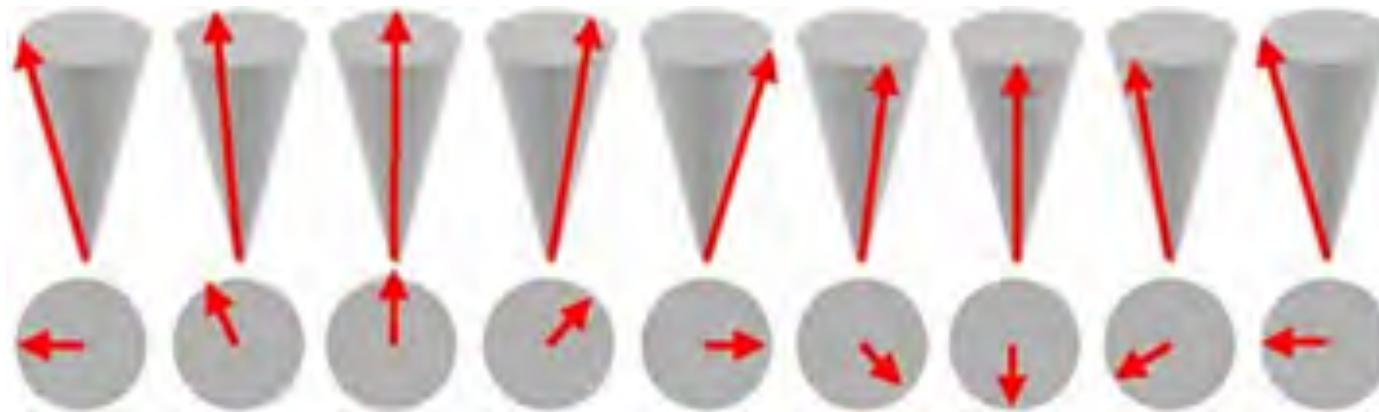


Antiferromagnetic material

Dispersion relations for Ge (Kittle)



Ferromagnetic Magnons / Spin Waves



1D linear response, nearest neighbour exchange.

Without anisotropy

$$\hbar\omega = 2E_{ex}(1 - \cos(ka))$$

$$\hbar\omega \approx E_{ex}a^2k^2 = Dk^2 \text{ for small } k$$

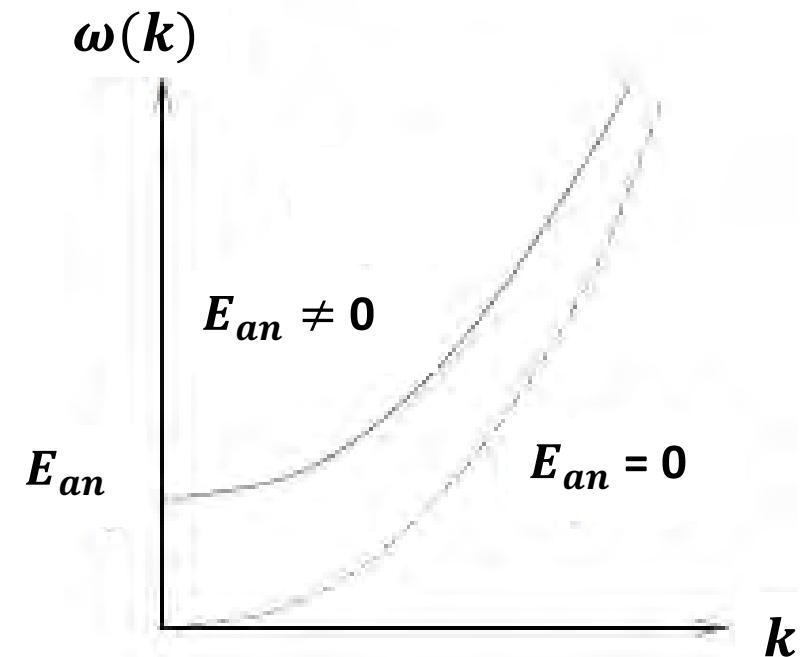
Energies per moment. D: spin wave stiffness

With uniaxial anisotropy

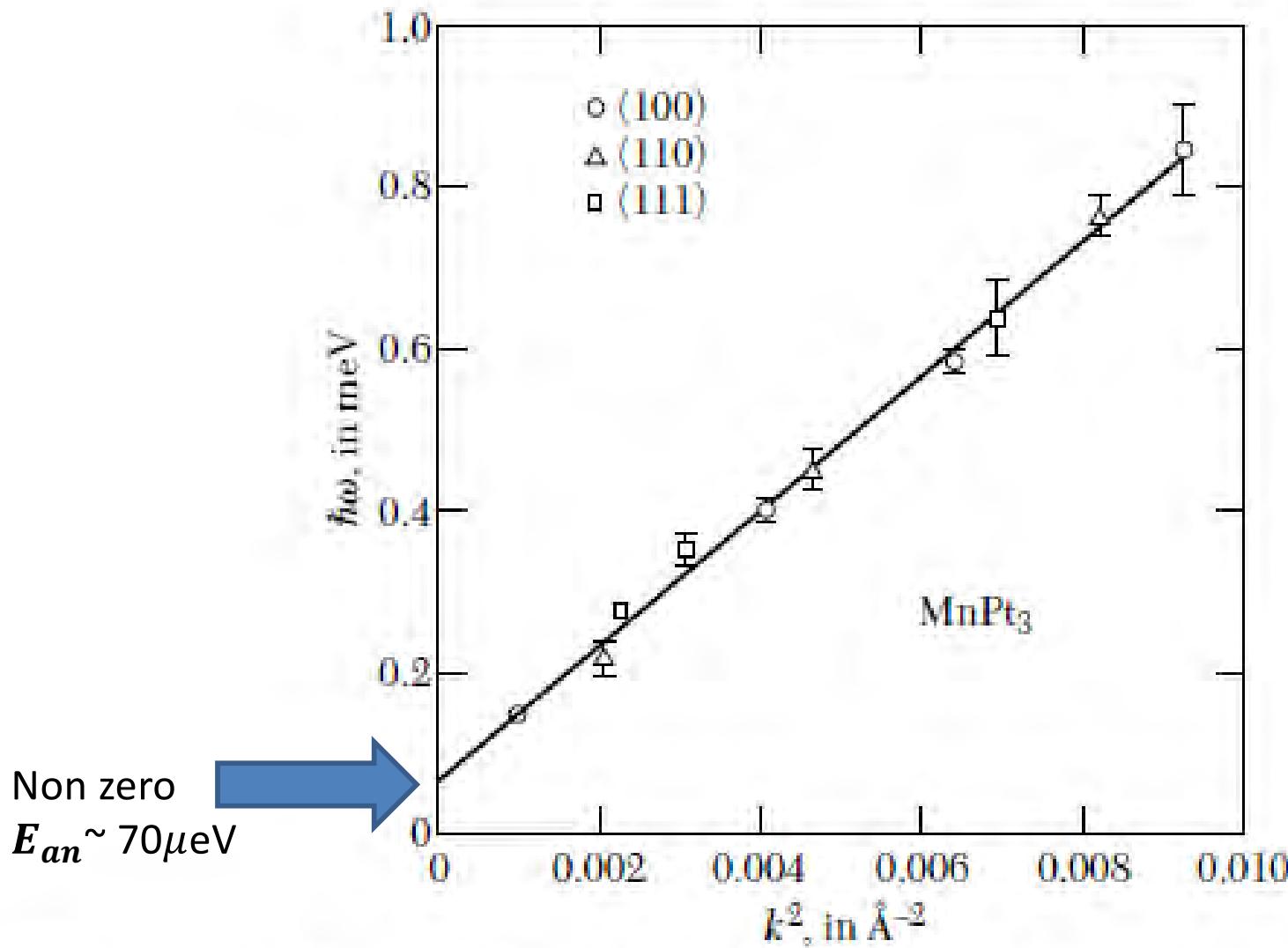
$$\hbar\omega \approx E_{an} + Dk^2 \text{ for small } k$$

For $k = 0$

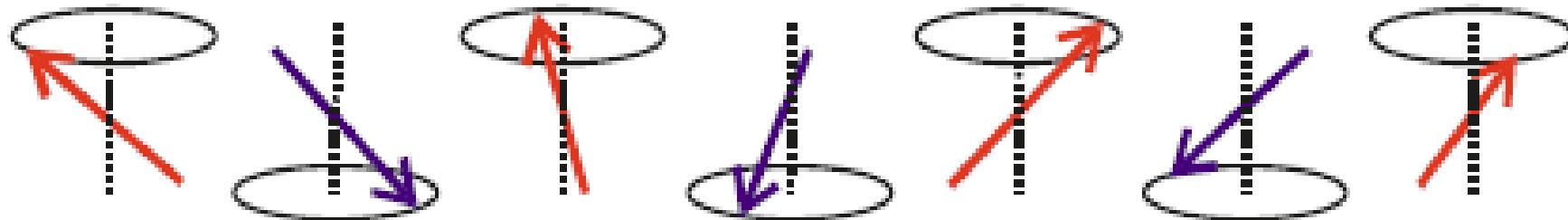
$$\hbar\omega \approx E_{an}$$



Ferromagnetic Magnons



Antiferromagnetic Magnons / Spin Waves



1D linear response, nearest neighbour exchange.

Without anisotropy

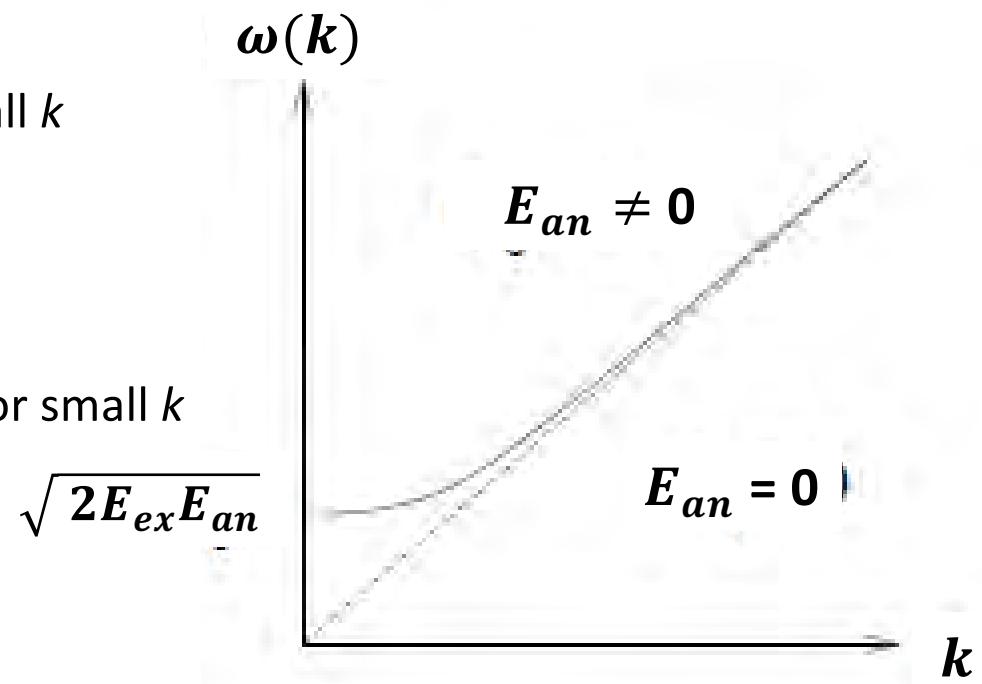
$$\hbar\omega = E_{ex}|\sin(ka)| \approx E_{ex}ak \text{ for small } k$$

With uniaxial anisotropy

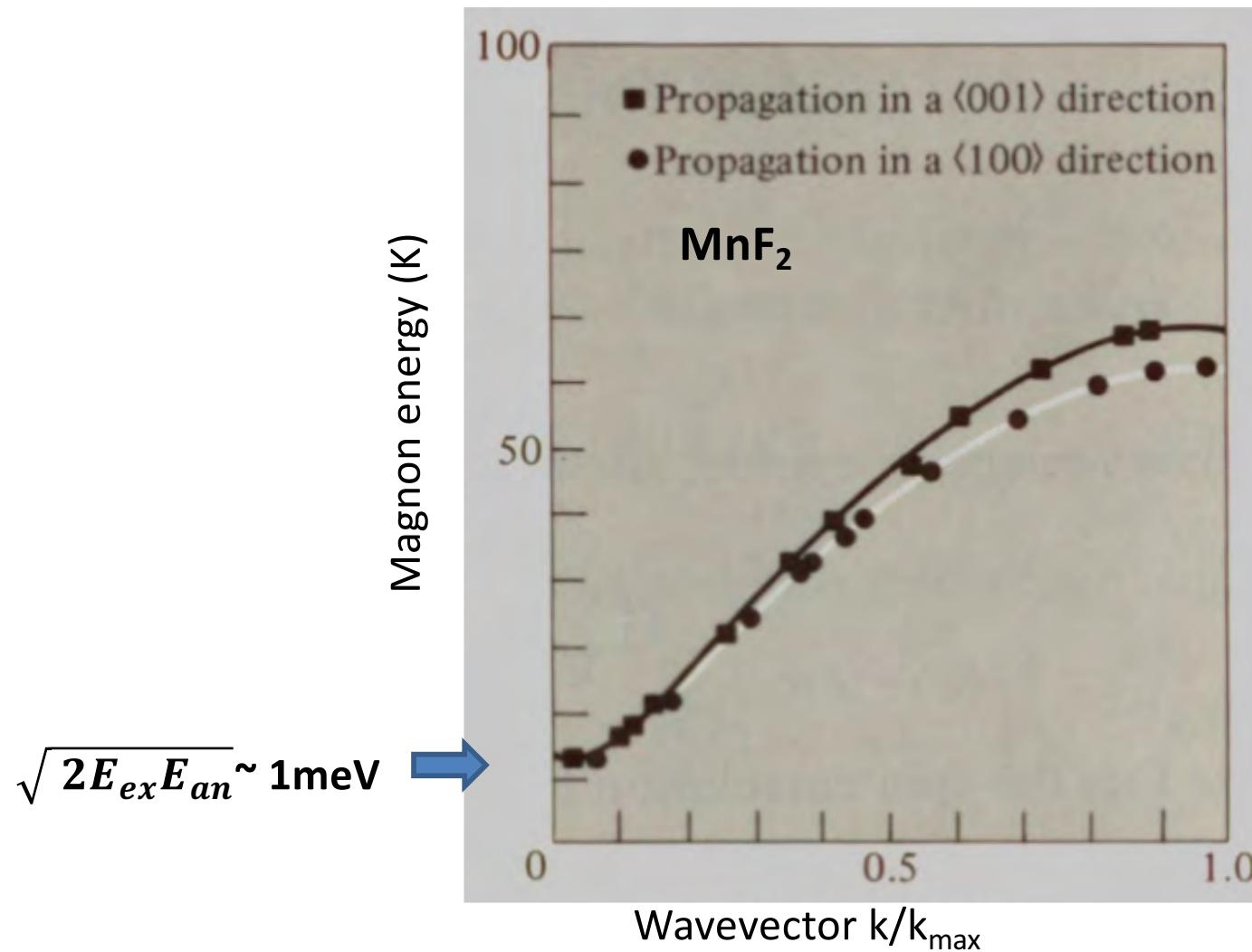
$$\hbar\omega \approx \sqrt{2E_{ex}E_{an} + E_{an}^2 + (E_{ex}ak)^2} \text{ for small } k$$

For $k = 0$

$$\hbar\omega \approx \sqrt{2E_{ex}E_{an} + E_{an}^2} \approx \sqrt{2E_{ex}E_{an}}$$

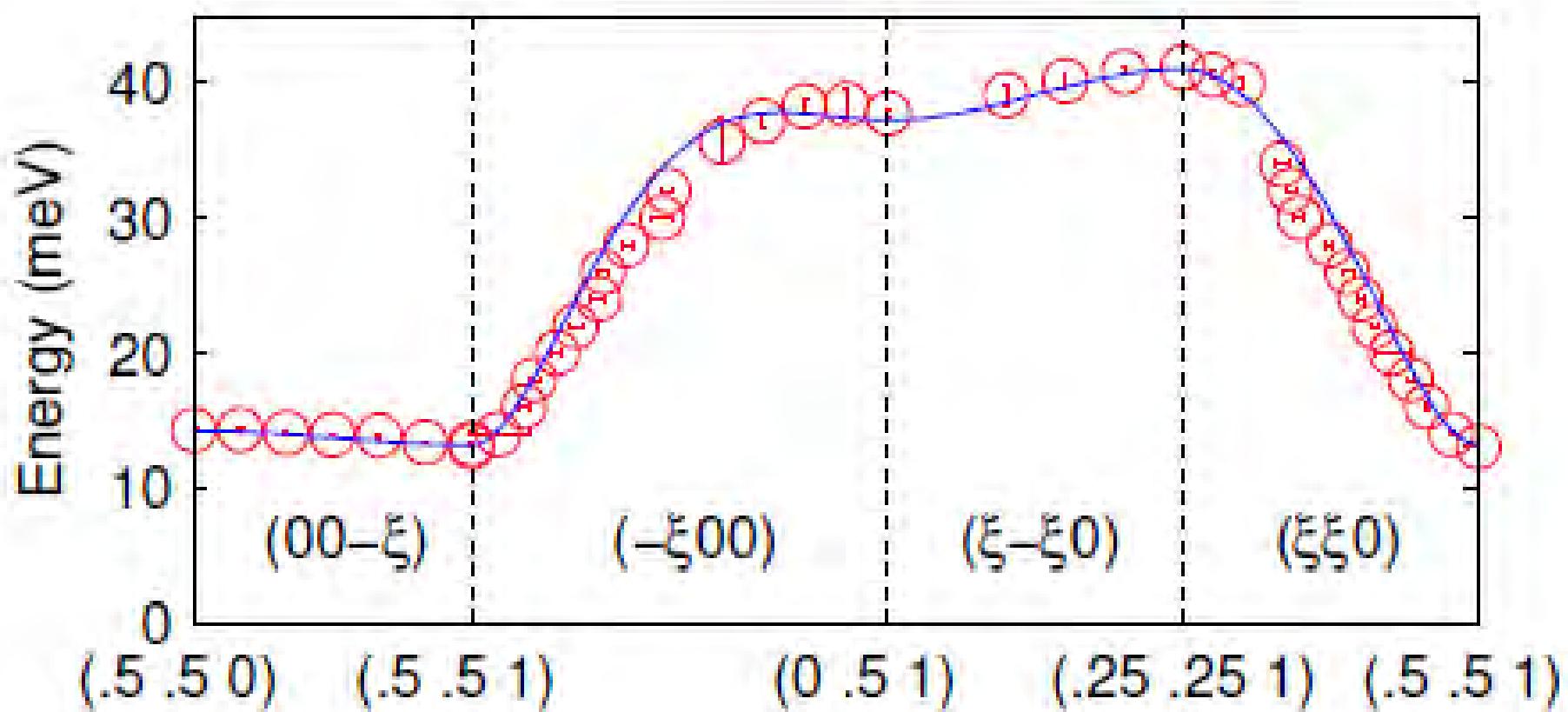


Antiferromagnetic Magnons



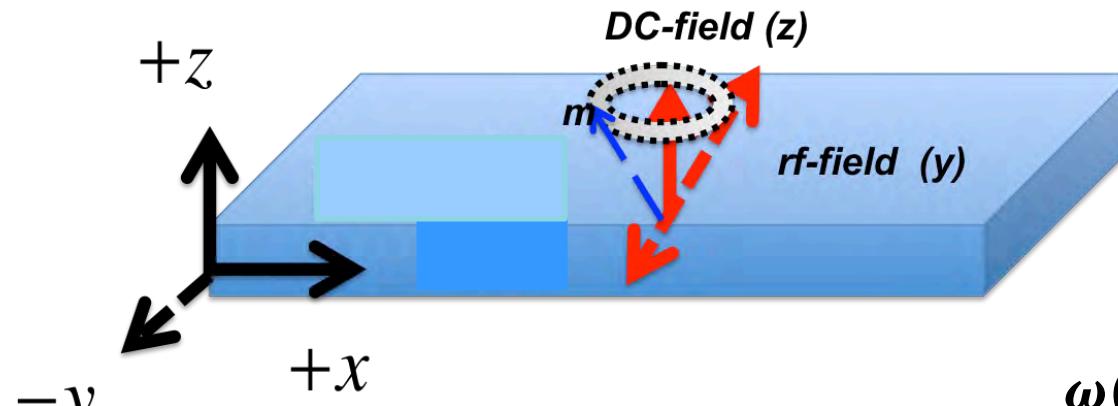
Antiferromagnetic Magnons

Highly anisotropic magnon dispersion in Ca_2RuO_4 : evidence for strong spin orbit coupling



4 Magnetic Characterisation: Magnetic Resonance

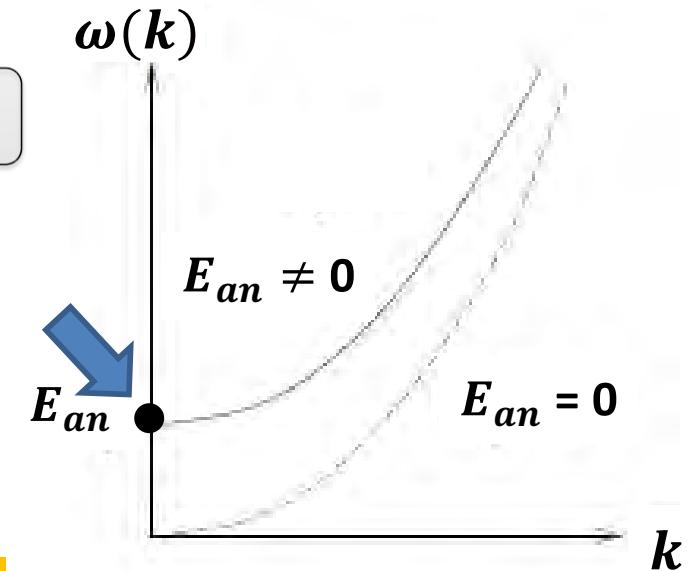
Ferromagnetic Resonance (FMR)



$$\omega_{Kittel} = \sqrt{(\omega_K + \omega_0)(\omega_K + \omega_0 + \omega_D)}$$

$$\omega_0 = \gamma H_{ext} \quad \omega_K = \gamma H_K \quad \omega_D = \gamma H_{demag.} = 0$$

Applied field Anisotropy field Demagnetising field



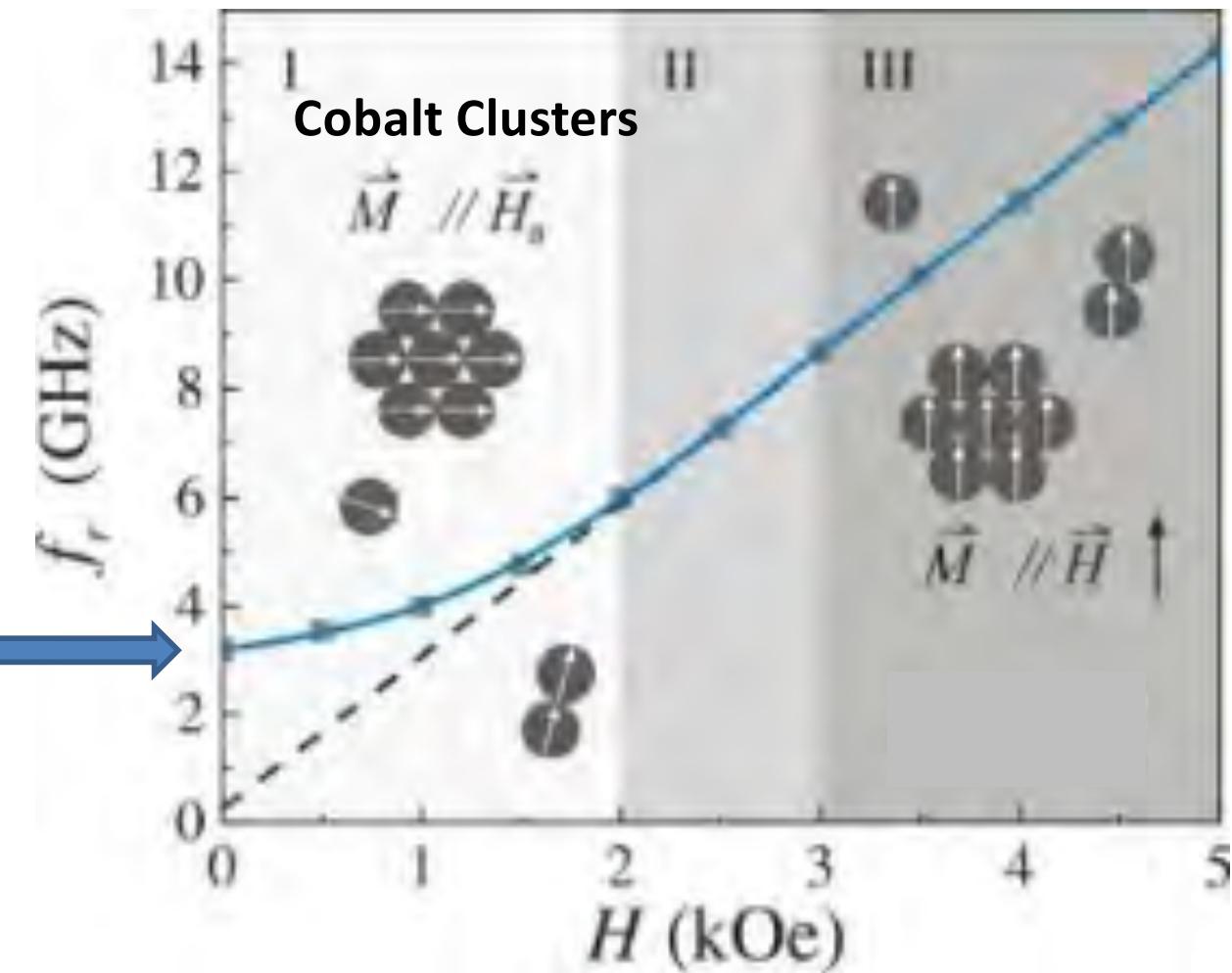
$H_{ext} \rightarrow 0$ Excited modes are magnons with $k \sim 0$

Resonant frequencies $\sim 1 - 10$ GHz. Natural speed of dynamics

Ferromagnetic Resonance (FMR): Anisotropy Energies

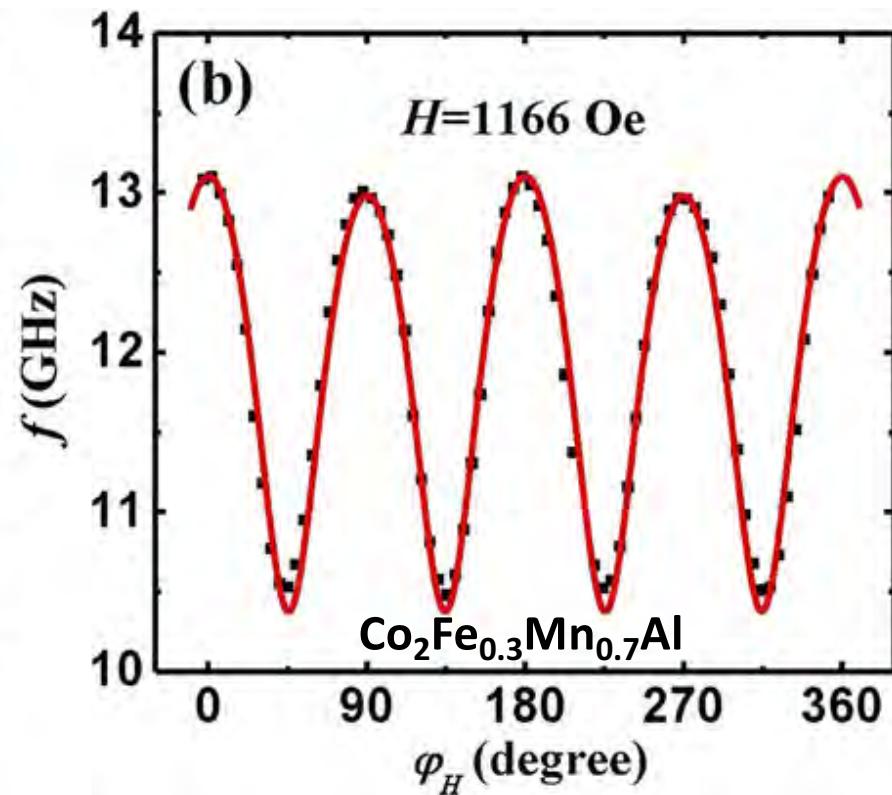
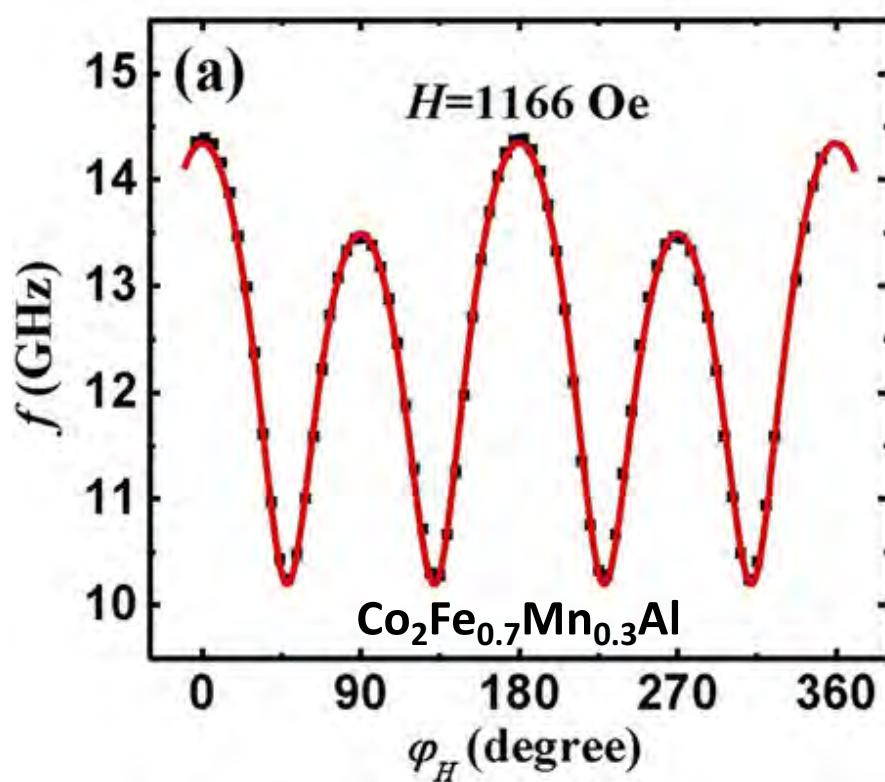
Zero applied field resonance due to anisotropy.

$$E_{an} \sim 15 \mu\text{eV}$$



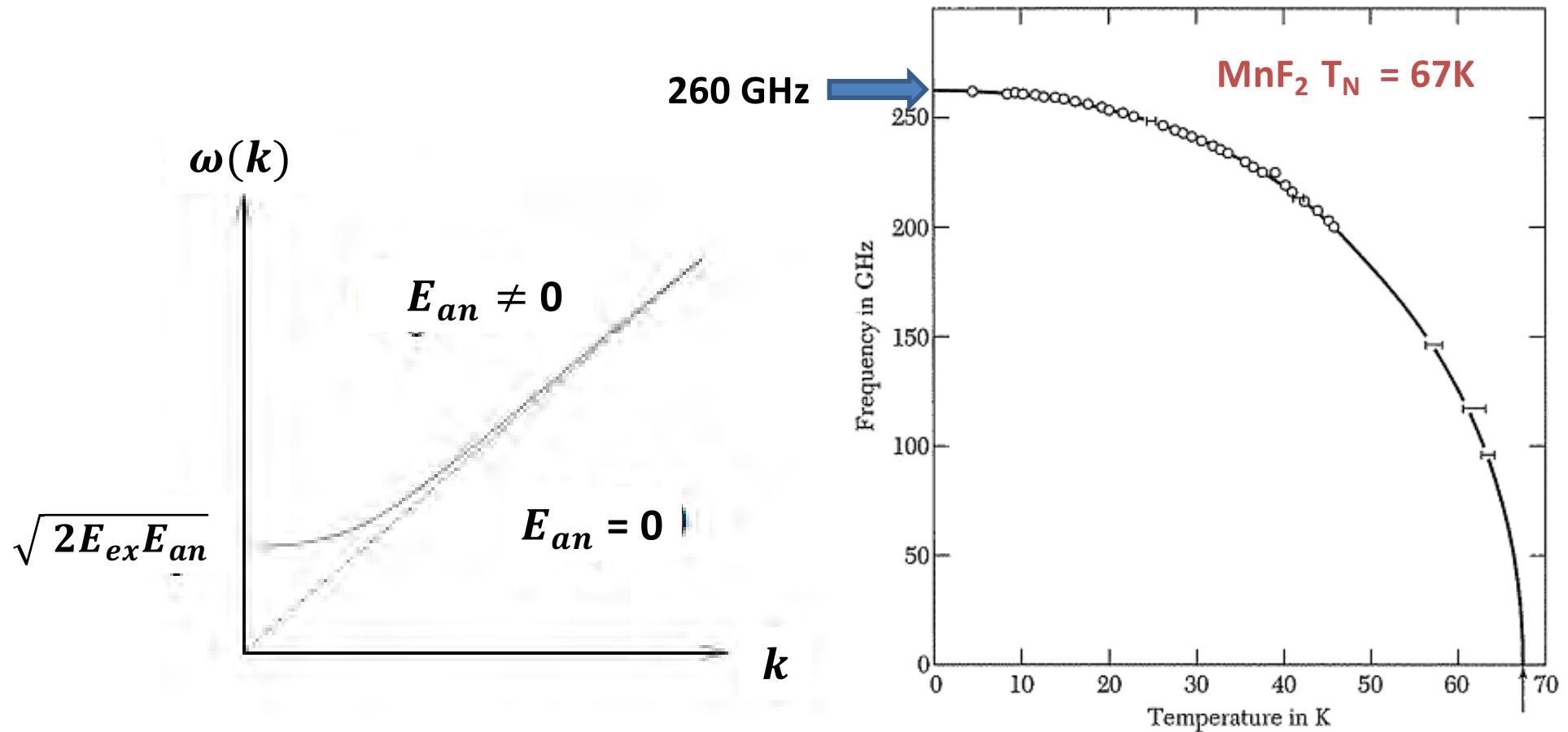
Ferromagnetic Resonance: Anisotropy

Angular dependence of ferromagnetic resonance frequency with a constant magnitude Applied magnetic field rotated in-plane.



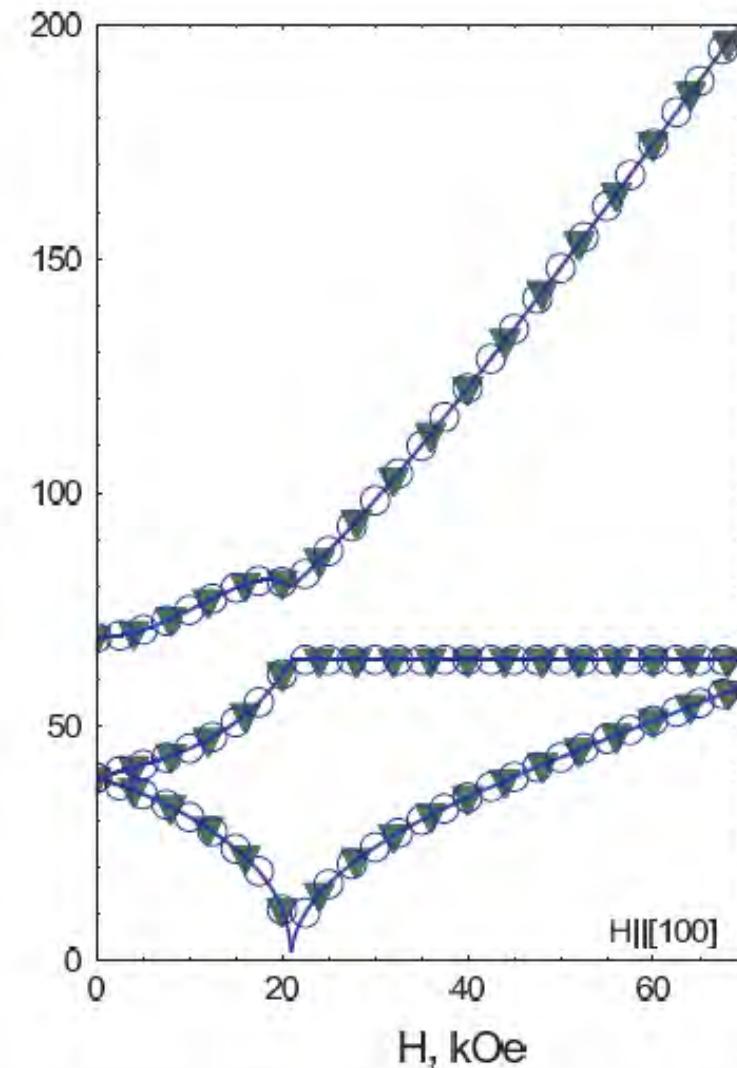
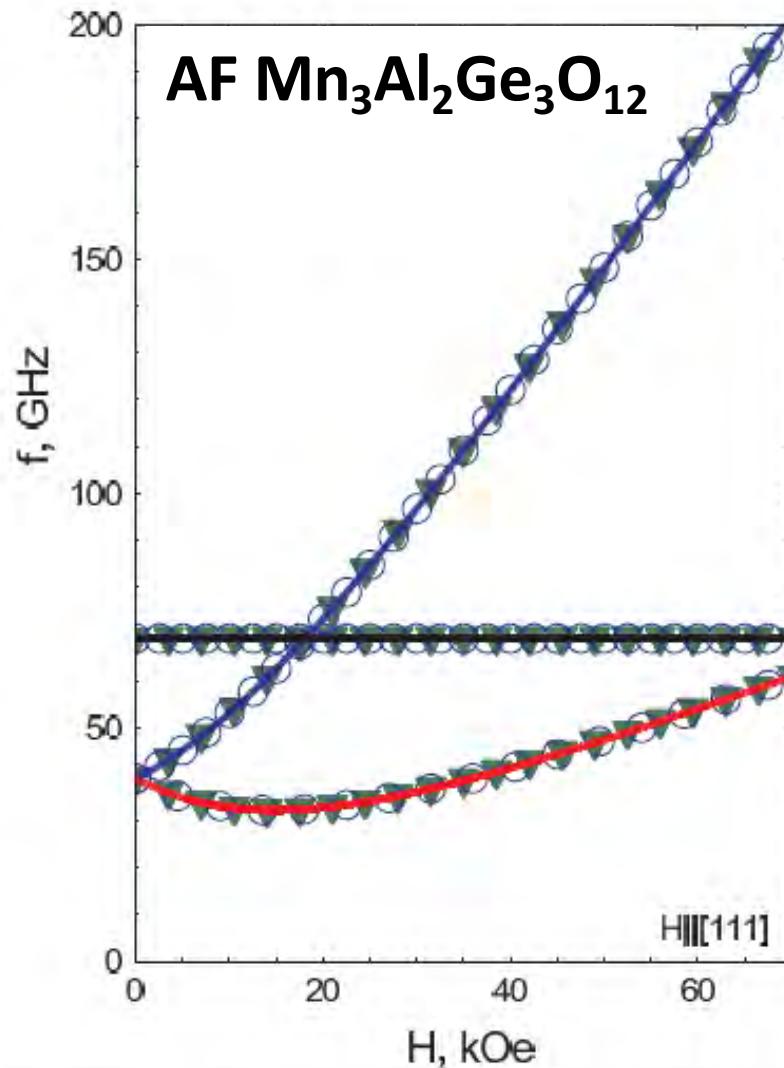
Antiferromagnetic Resonance (AFMR)

$H_{\text{ext}} \rightarrow 0$ Excited modes are magnons with $k \sim 0$



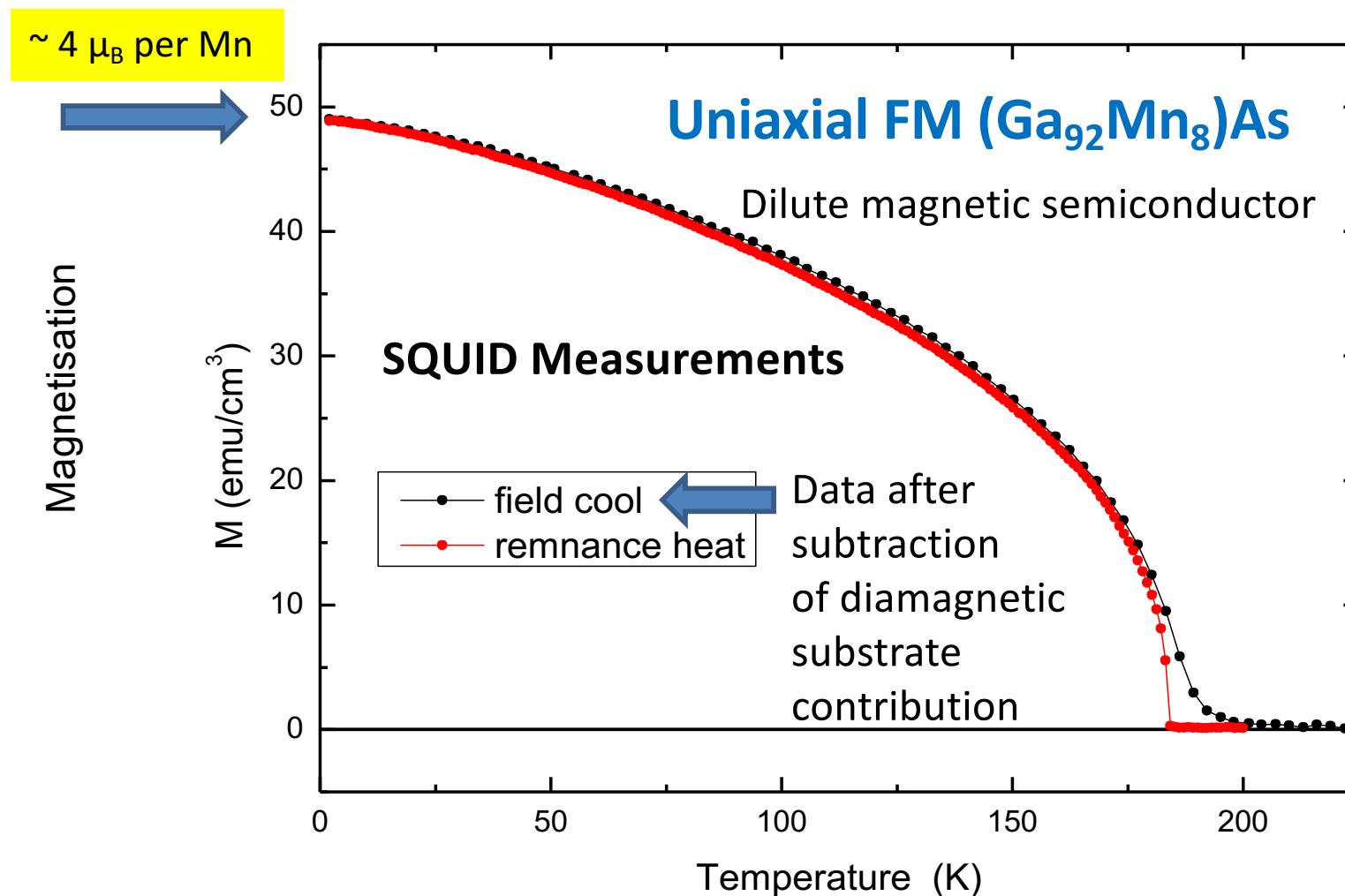
Resonant frequencies $\sim 100 - 1000$ GHz. Natural speed of dynamics

Antiferromagnetic Resonance



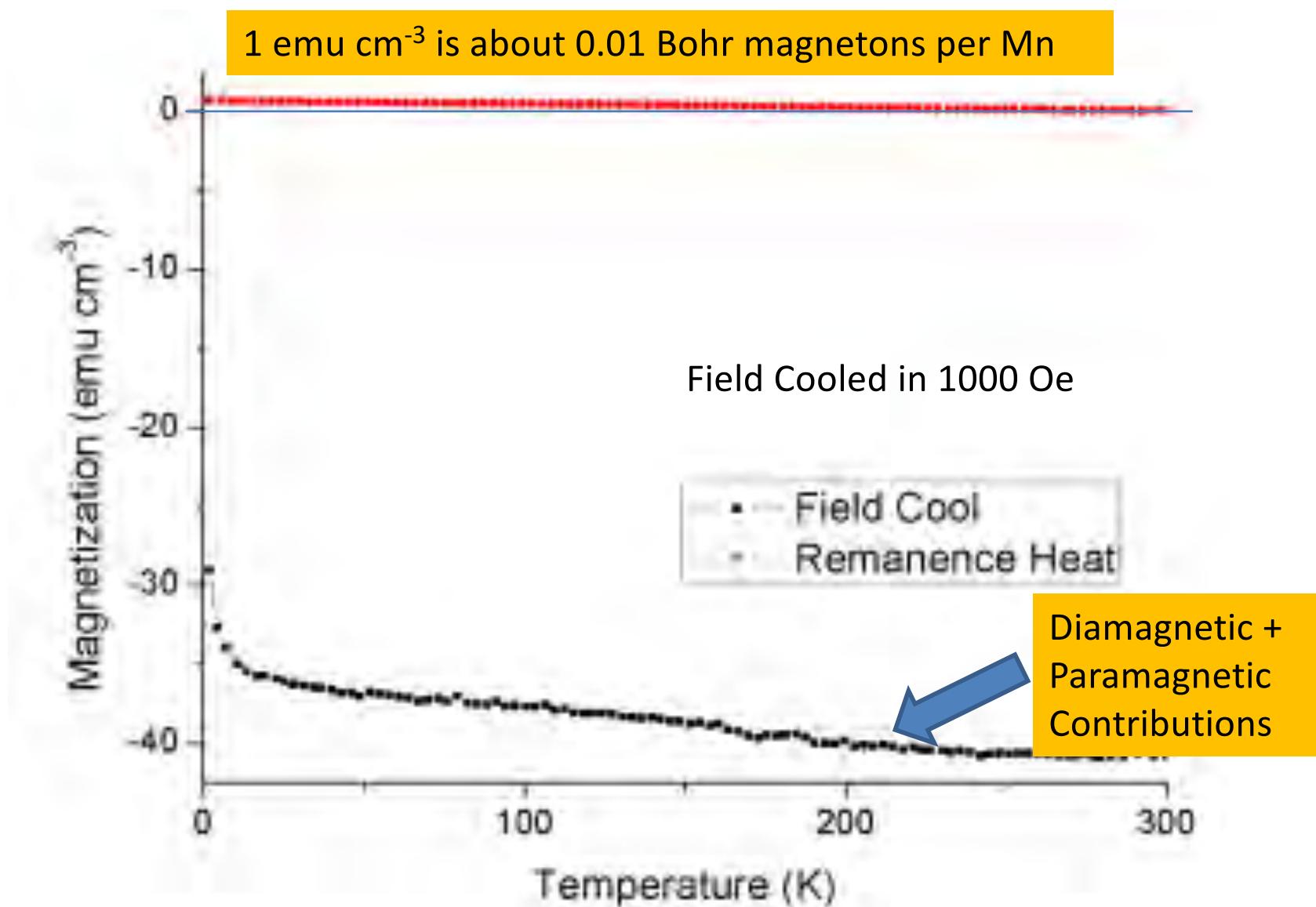
More than one type of Mn site : both “acoustic” and “optical” magnons

3.4 Magnetic Characterisation: Remanence

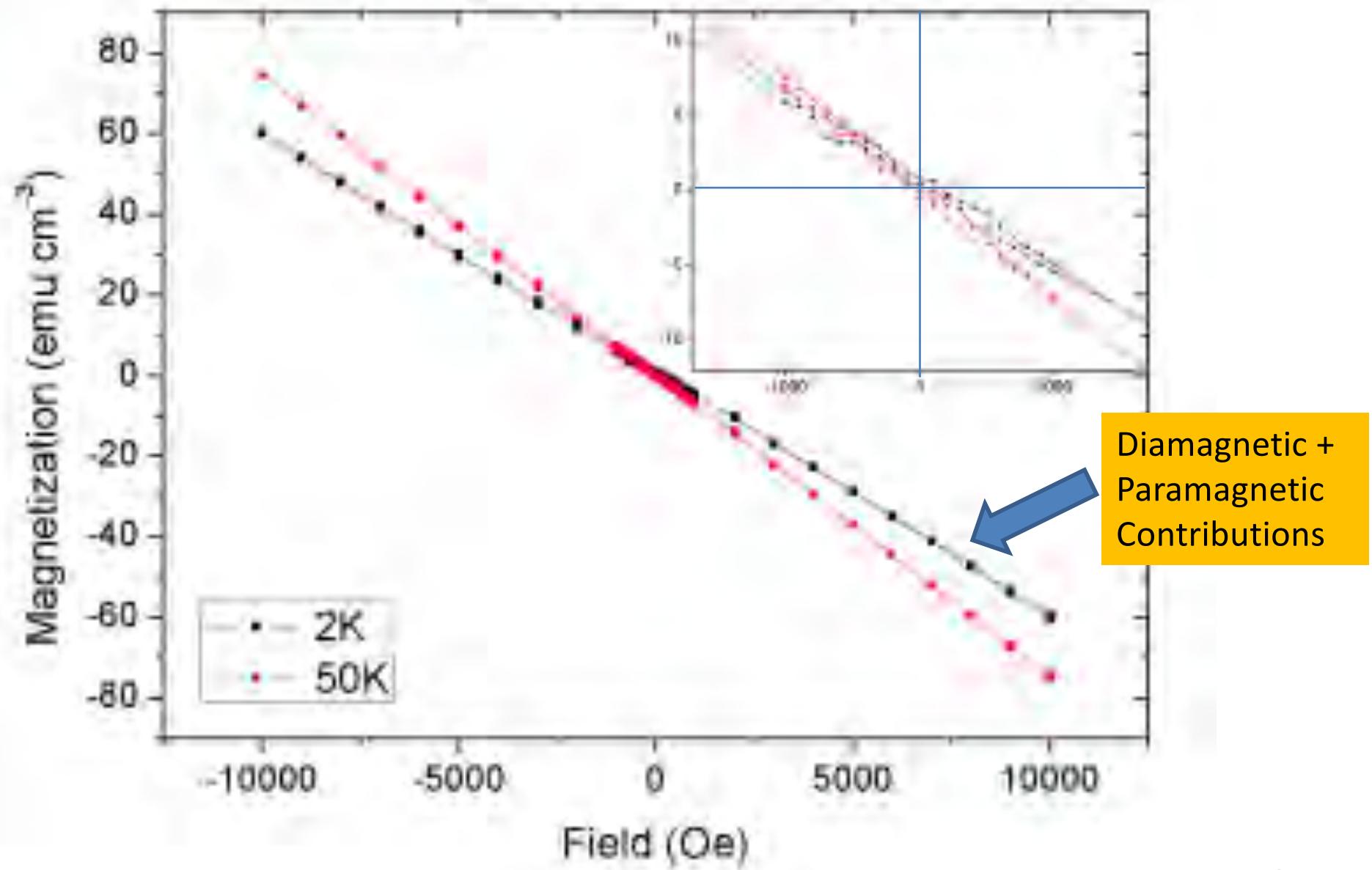


Measuring Remanence. Cool in field (in 300mT). Warm in zero field

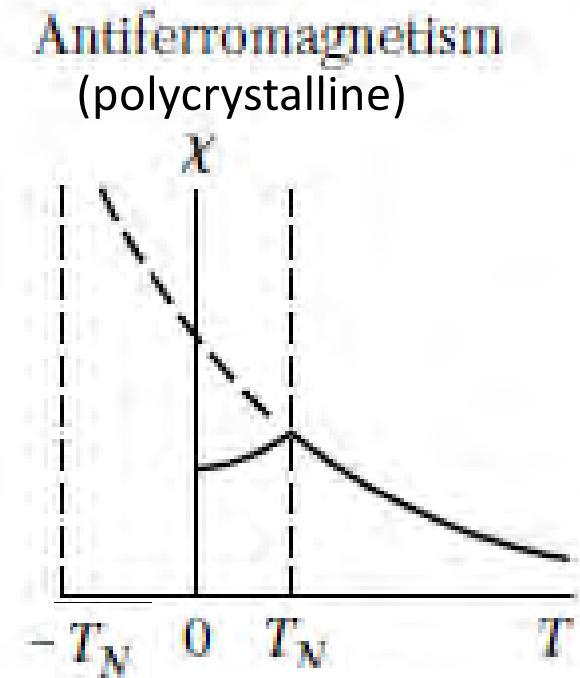
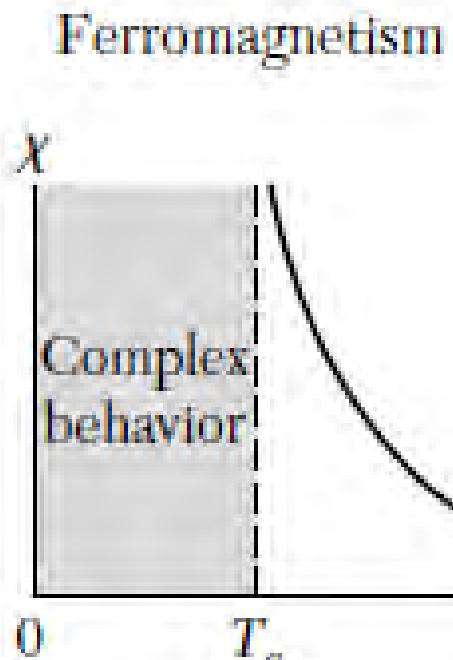
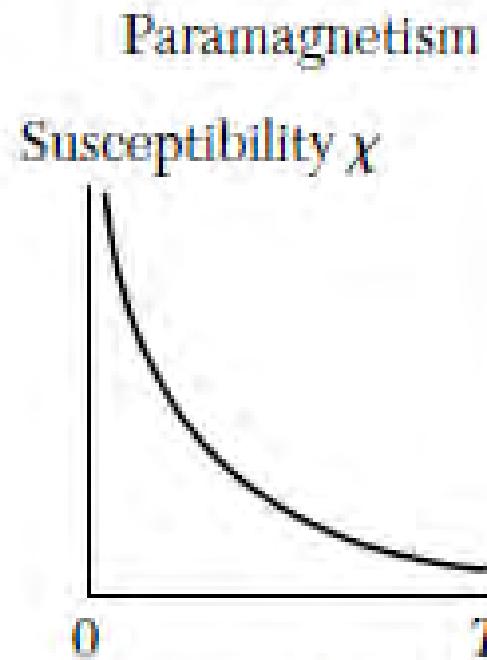
Magnetisation: AF CuMnAs



Magnetisation: AF CuMnAs



Susceptibility



$$\chi = \frac{C}{T}$$

Curie Law

$$T > T_c$$

$$\chi = \frac{C}{T - T_c}$$

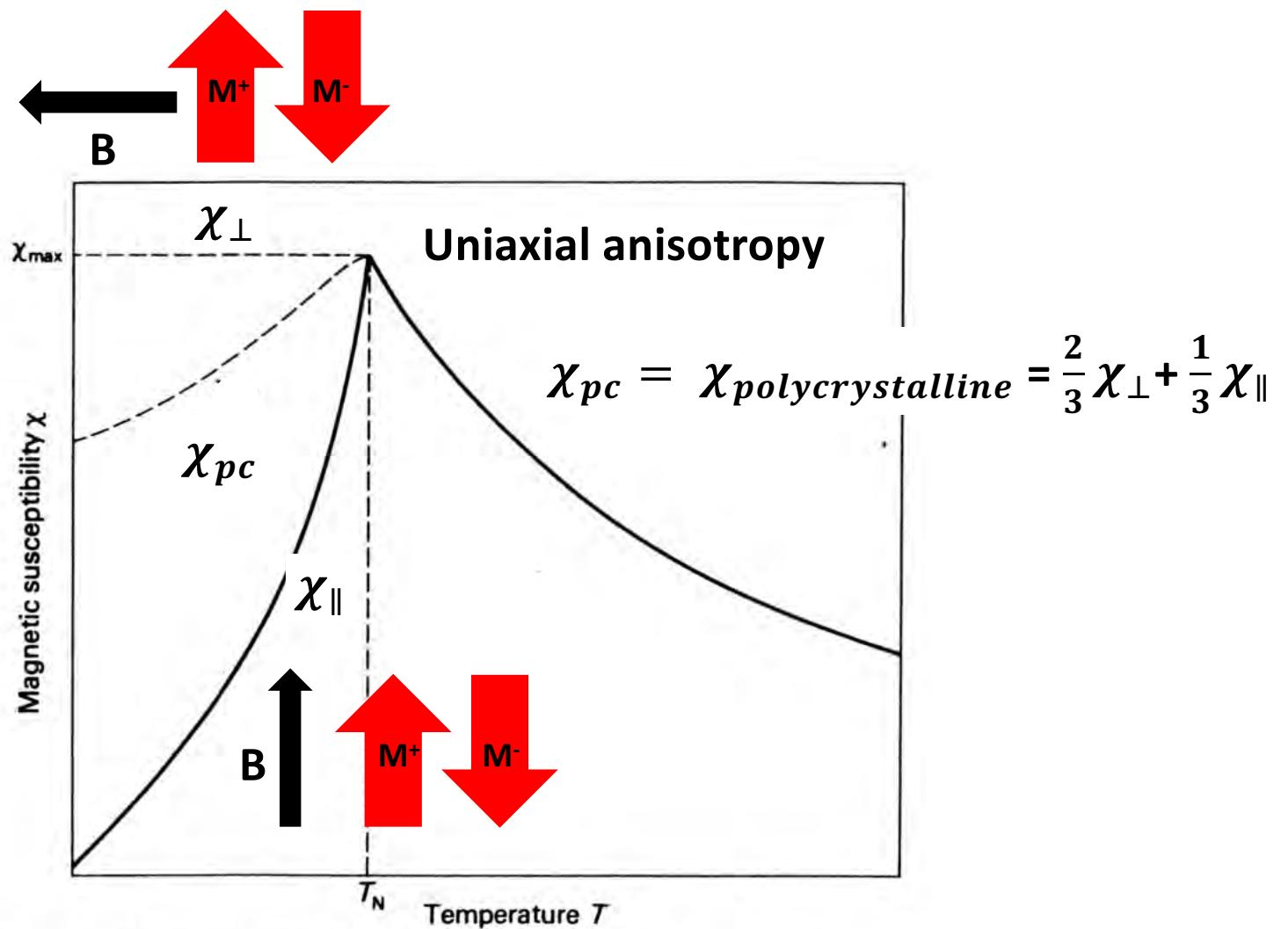
Curie - Weiss

$$T > T_N$$

$$\chi = \frac{C}{T + T_N}$$

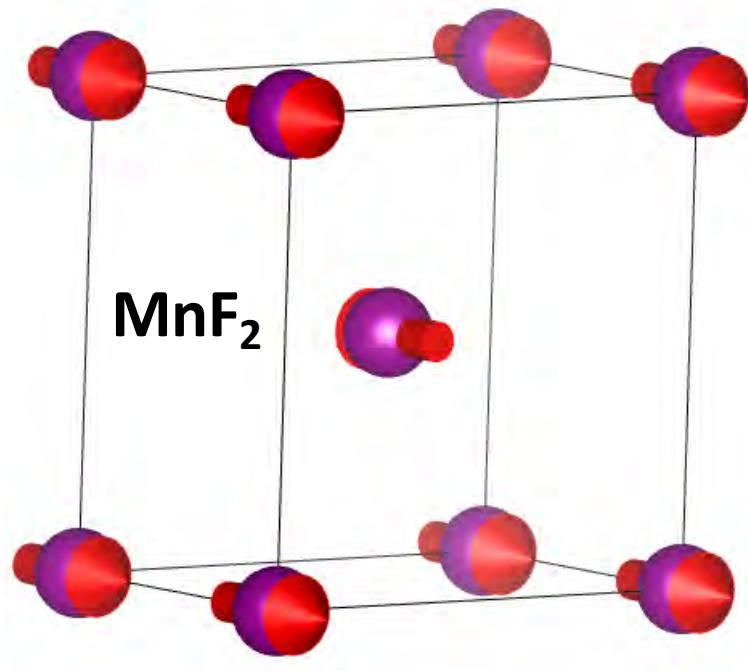
From Kittl
44

Antiferromagnetic Susceptibility

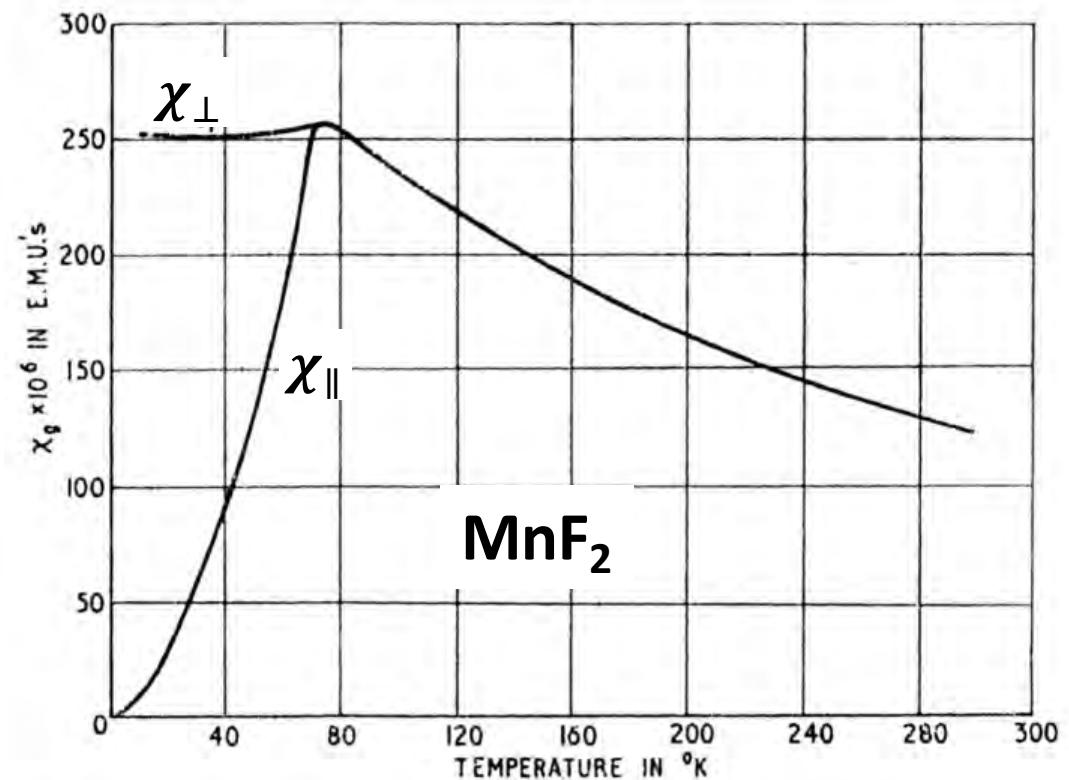


Peak susceptibility gives Neel Temperature

Antiferromagnetic Susceptibility



Positions of Mn atoms

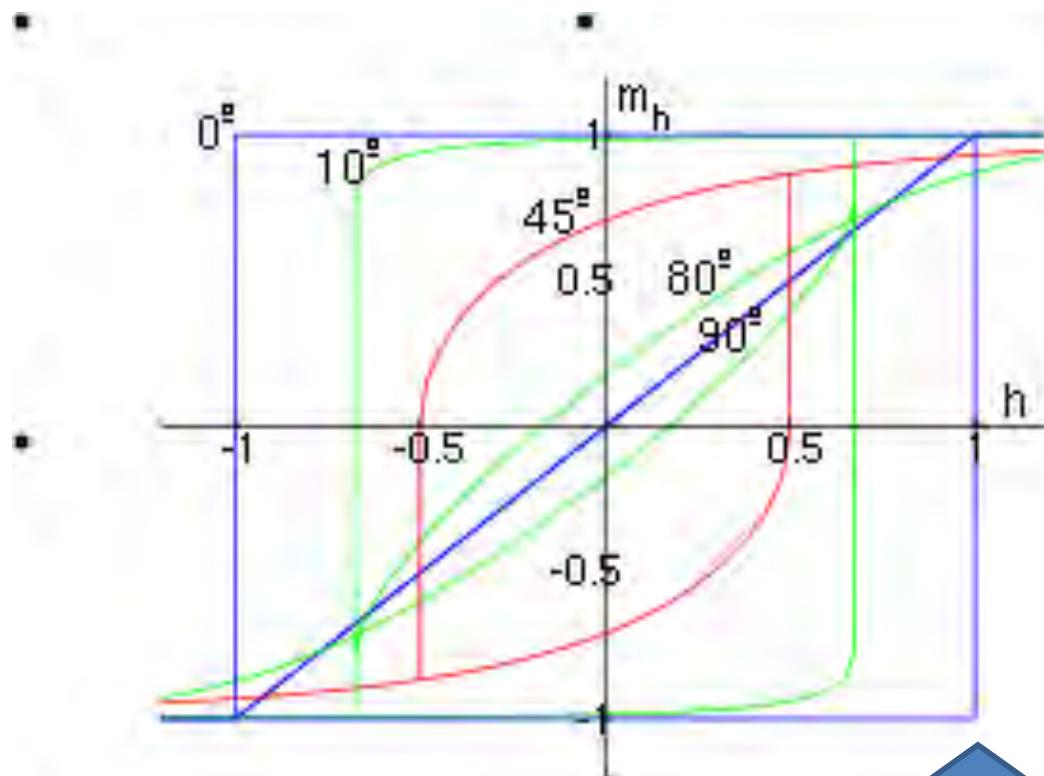


Stop / Jump ?

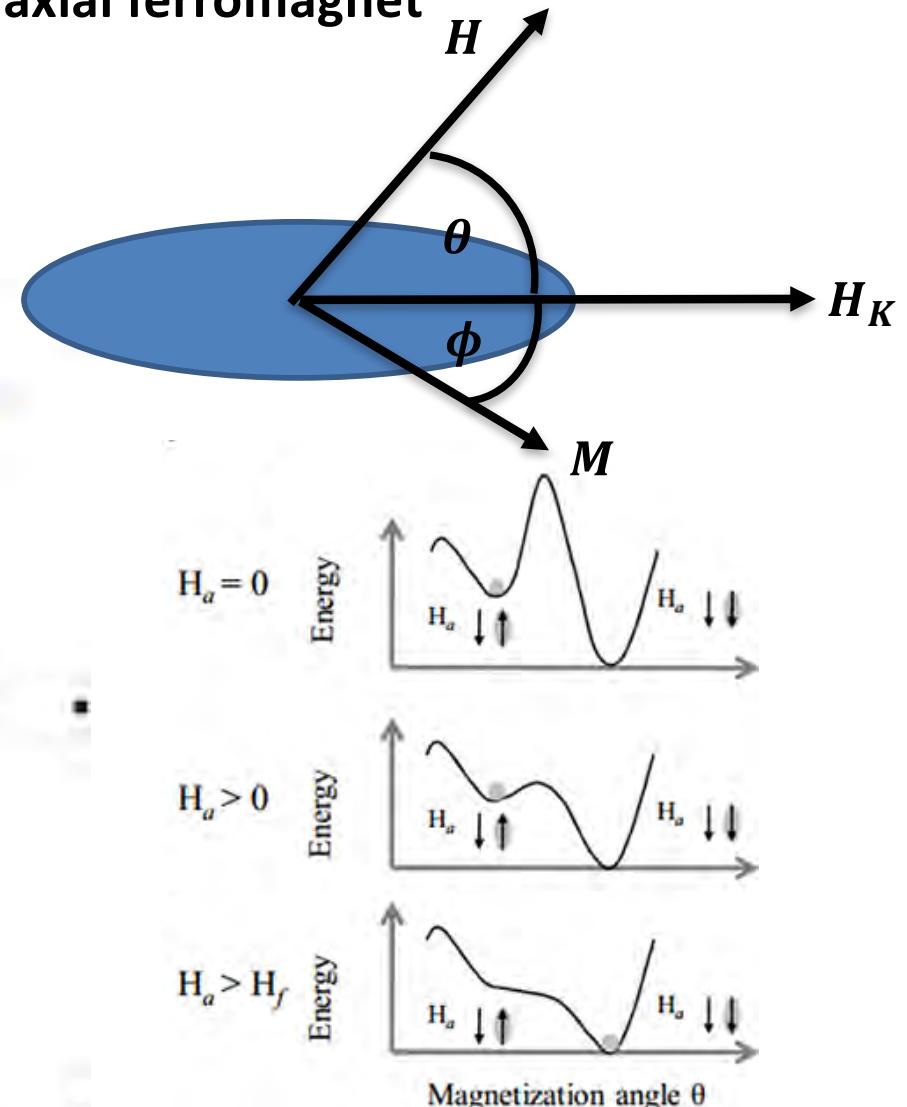
Magnetic Behaviour in “Large” Fields

Stoner–Wohlfarth model : single domain uniaxial ferromagnet

$$\epsilon = -K_u \cos^2 \phi - \mu_0 H M \cos(\phi + \theta)$$

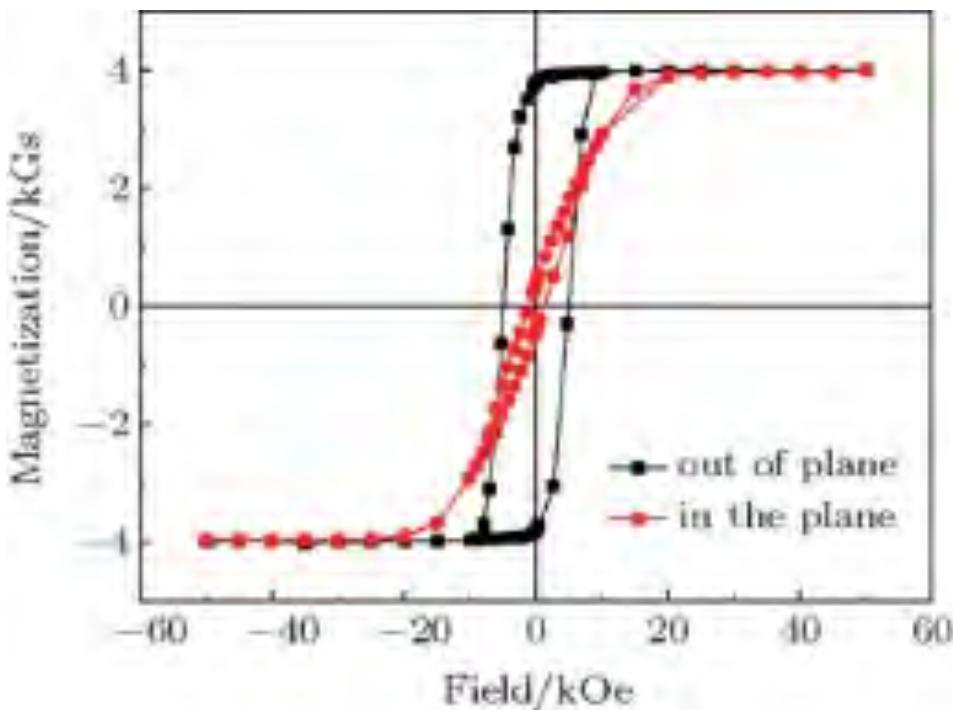


Spin Flip



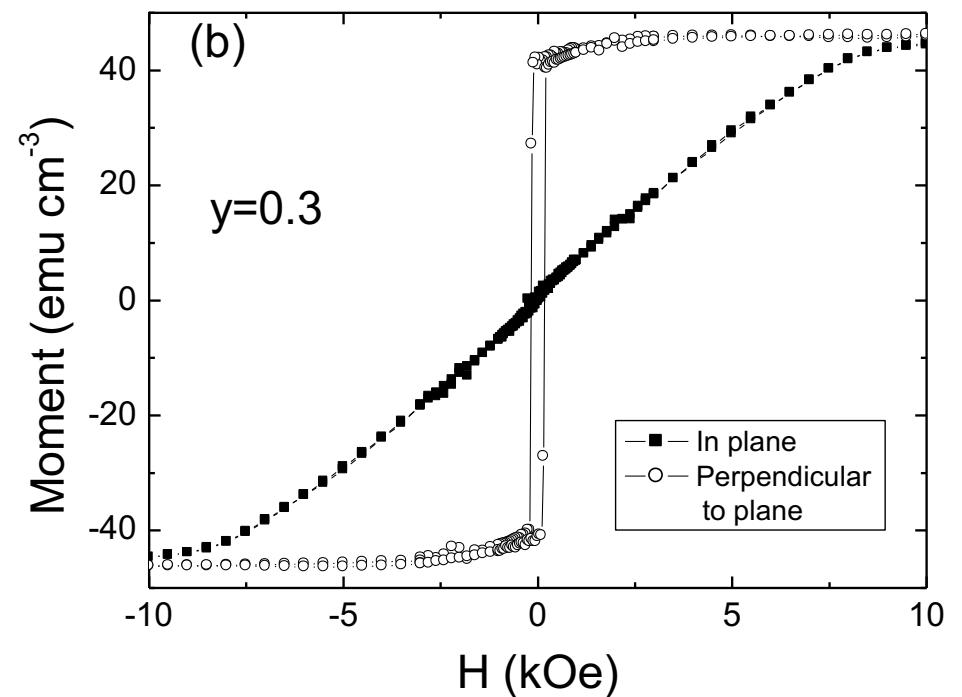
Magnetic Behaviour in “Large” Fields

FM Perpendicularly barium ferrite thin film



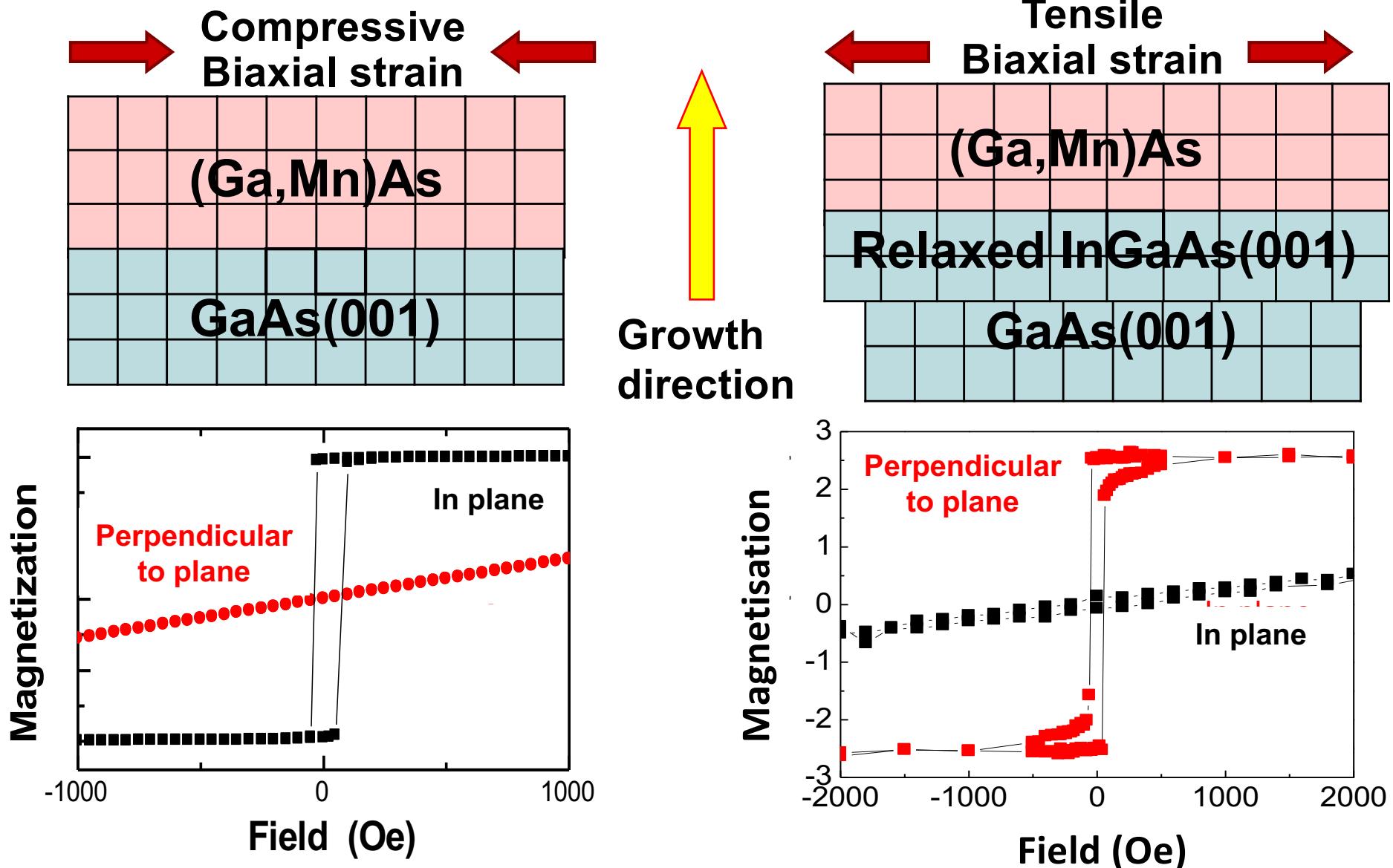
Da-Ming et al *Chinese Physics B*, 2016, 25(6): 068403

FM $(\text{Ga,Mn})(\text{As}_{70} \text{P}_{30})$



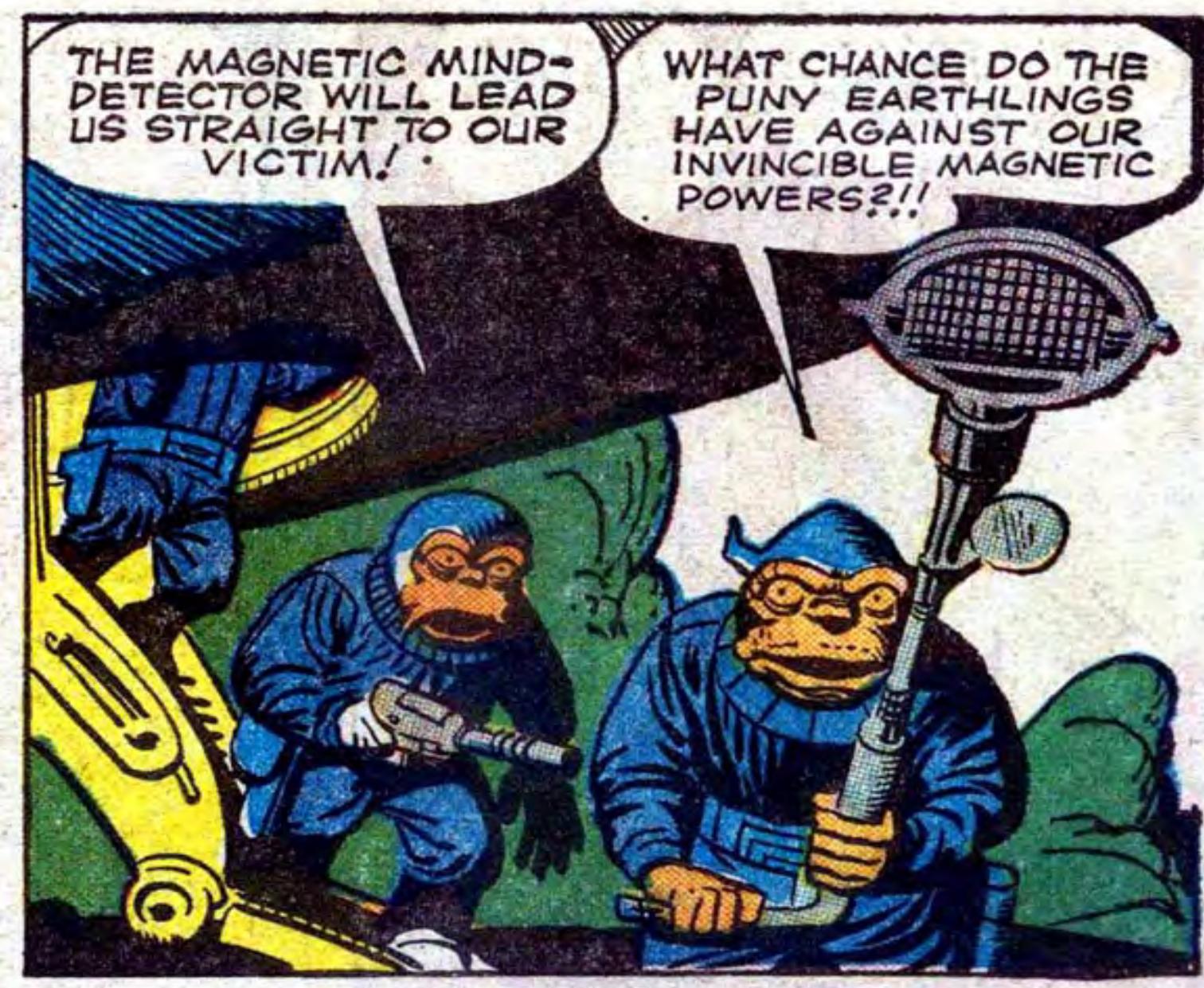
Normally don't see ideal behaviour because of domain formation

FM: Strain-induced Anisotropy



Dominant magneto-crystalline anisotropy in-plane for compressive strain and or out-of-plane for tensile strain strain

End Part One



The importance of avoiding stray fields

Welcome back



The human magnetism of Miroslaw Magola
Der Menschliche Magnetismus des Miroslaw Magola
Fenómenos de la magnetismo animal y terapéutico / Fenômenos de la Biofísica

www.magola.com

Erleuchtung

Red Bull

Miroslaw Magola

WIN WIN CD

KINESIS

... at magnetism among other strange effects.

... of his hand upon an upturned metal pot, keys or even coins and lift them up into the air. Through the same concentration he can

... gravity, which according to the known and generally accepted laws

... by people who have witnessed Magola, is absolutely possible.

... the same effects can also be seen in others!!

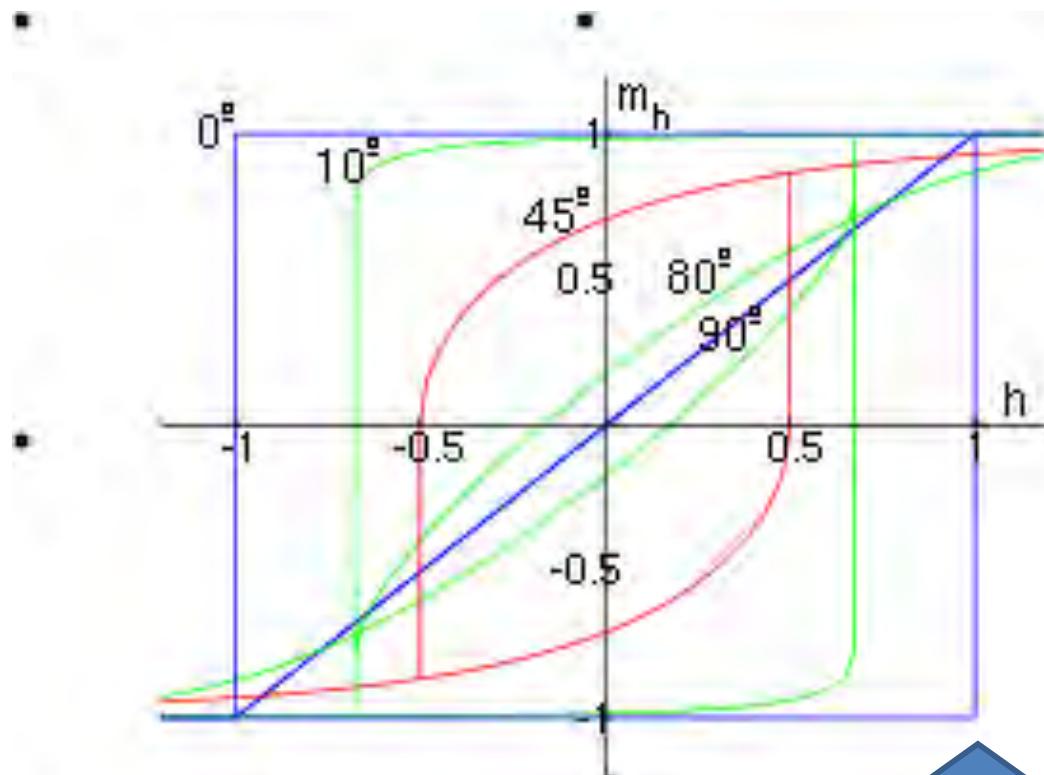
... to the interested public and explains his ideas of what lies behind them.

"Der Menschliche Magnetismus" wurde vor ca. 150 Jahren, zuerst von Franz Förschner (Hector Durville, Albert de Rochas u.a.) geprägt. Vielmehr ist es eine konzentrierte Gedankenkraft, die

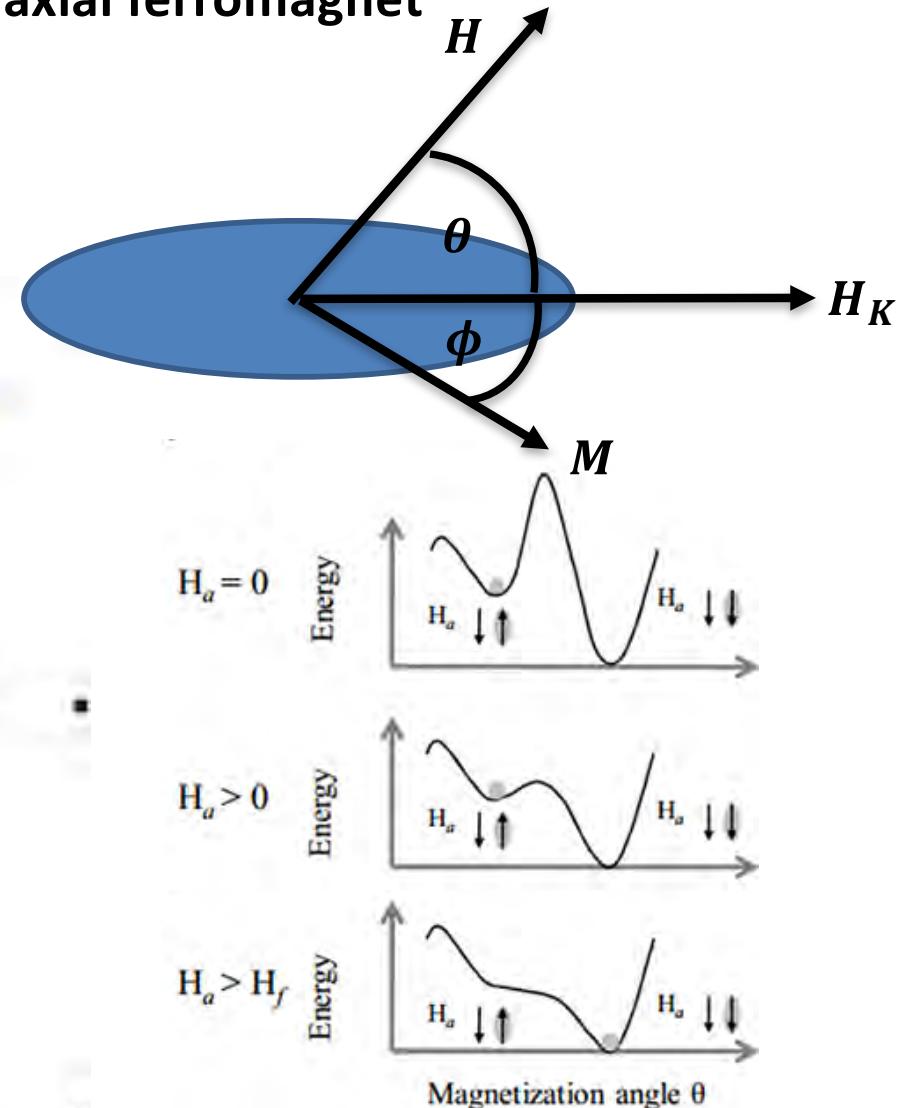
Magnetic Behaviour in “Large” Fields

Stoner–Wohlfarth model : single domain uniaxial ferromagnet

$$\epsilon = -K_u \cos^2 \phi - \mu_0 H M \cos(\phi + \theta)$$



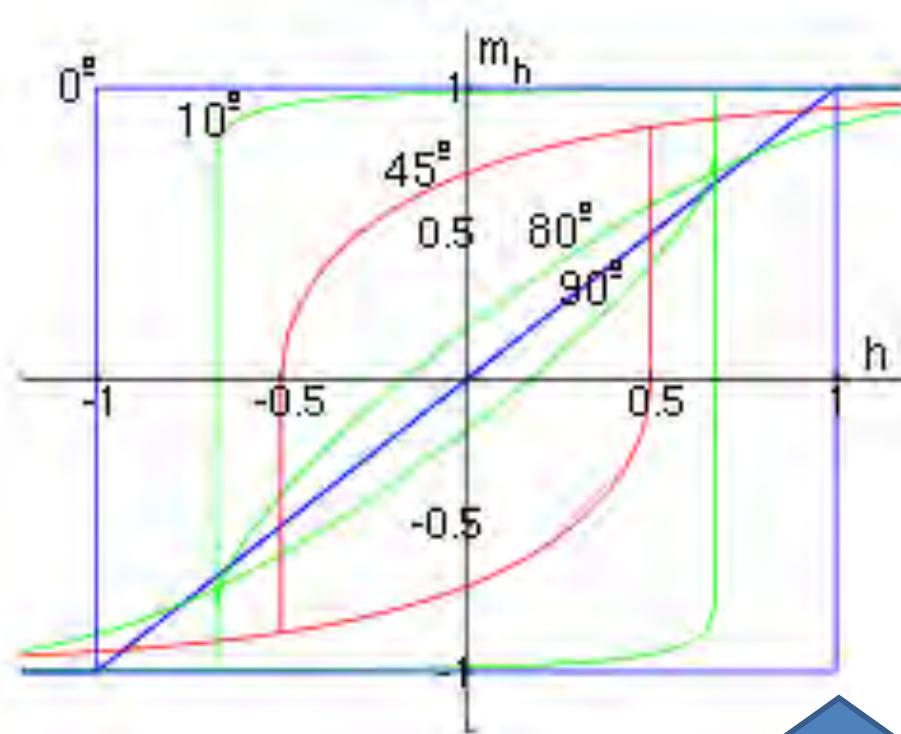
Spin Flip



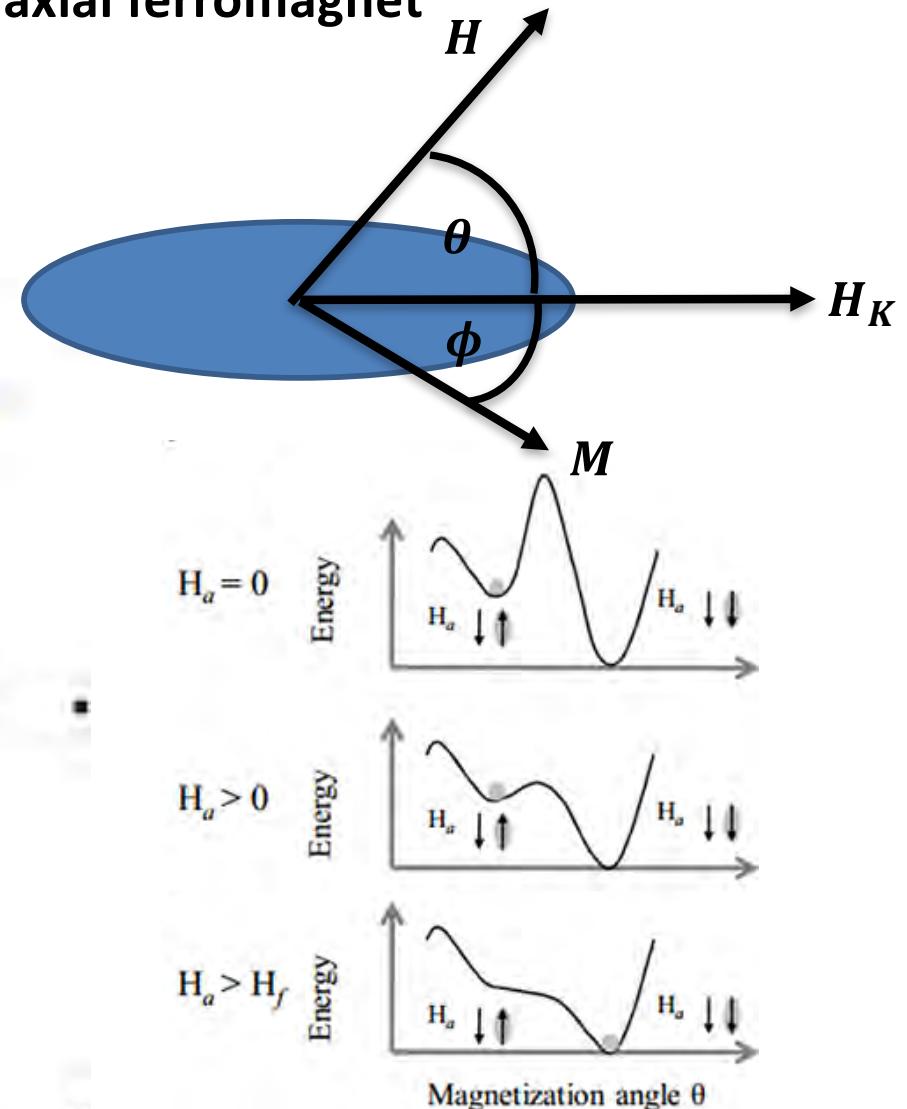
Magnetic Behaviour in “Large” Fields

Stoner–Wohlfarth model : single domain uniaxial ferromagnet

$$\epsilon = -K_u \cos^2 \phi - \mu_0 H M \cos(\phi + \theta)$$

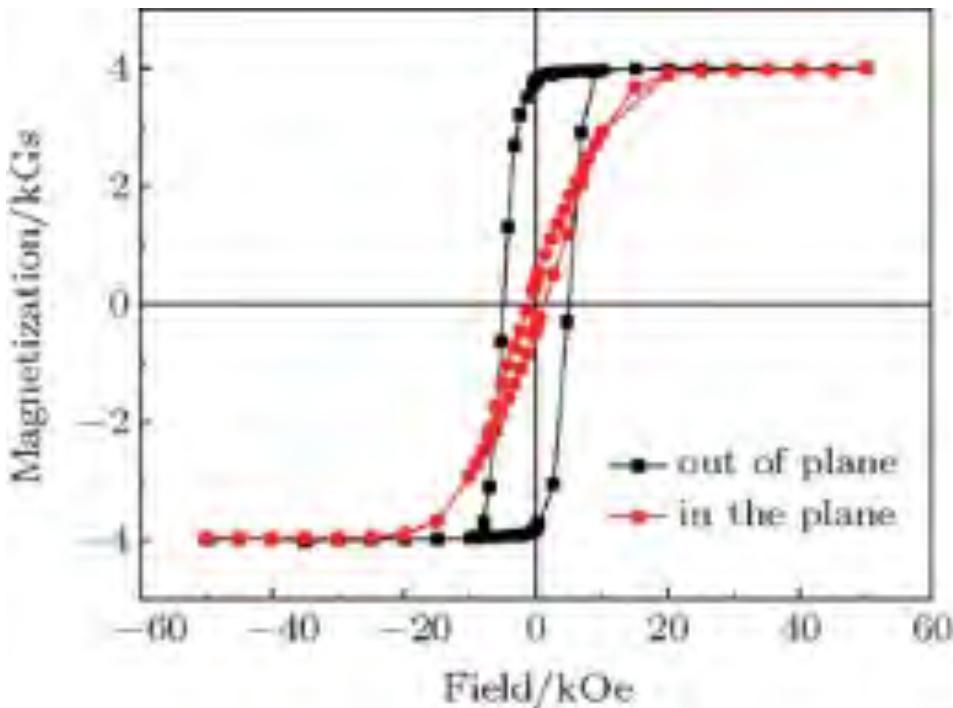


Spin Flip



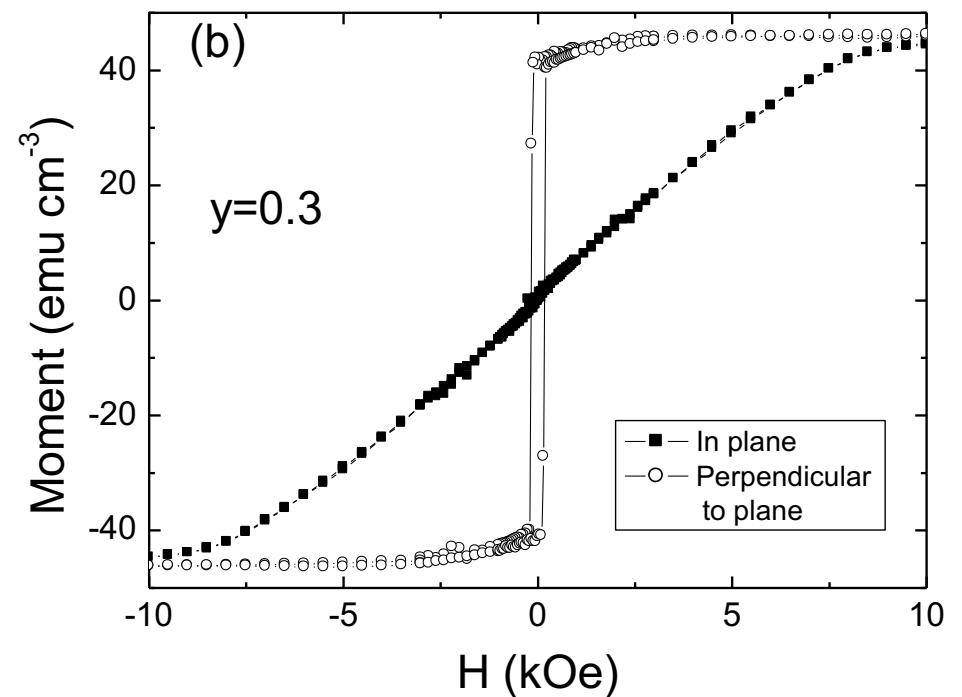
Magnetic Behaviour in “Large” Fields

FM Perpendicularly barium ferrite thin film



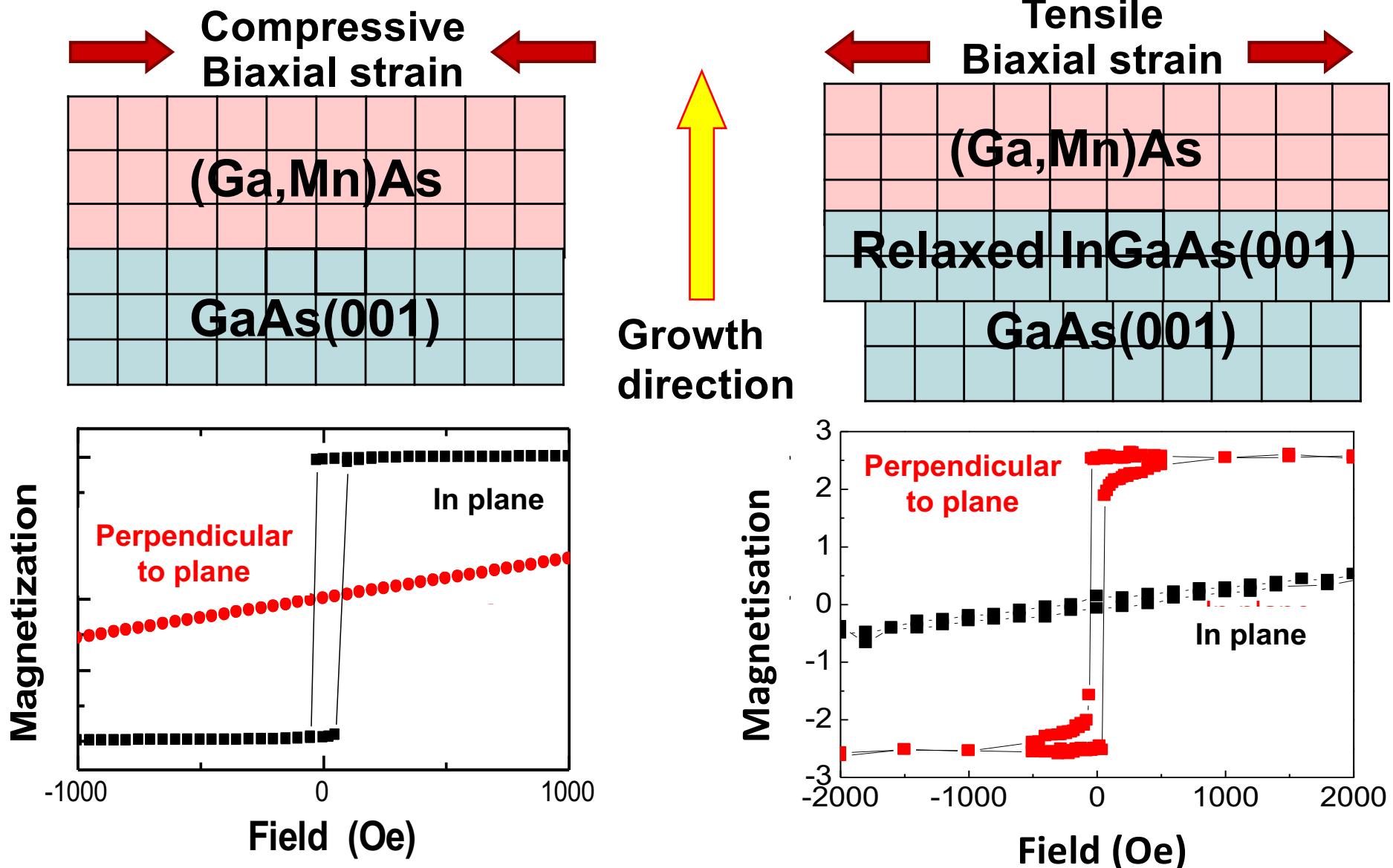
Da-Ming et al *Chinese Physics B*, 2016, 25(6): 068403

FM $(\text{Ga,Mn})(\text{As}_{70}\text{P}_{30})$



Normally don't see ideal behaviour because of domain formation

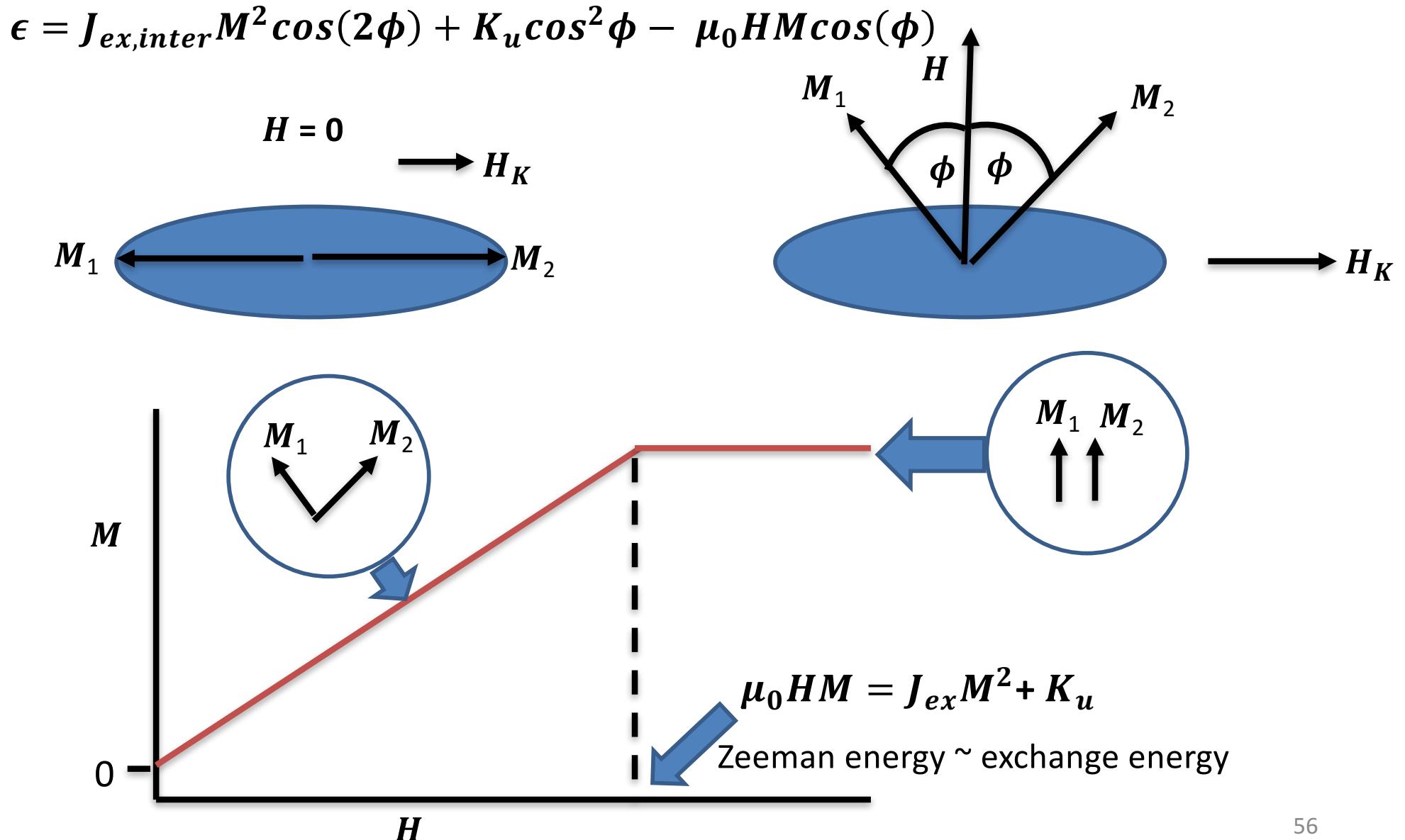
FM: Strain-induced Anisotropy



Dominant magneto-crystalline anisotropy in-plane for compressive strain and or out-of-plane for tensile strain strain

Antiferromagnets: Spin Flop Transitions

Single domain uniaxial antiferromagnet H perpendicular to easy axis

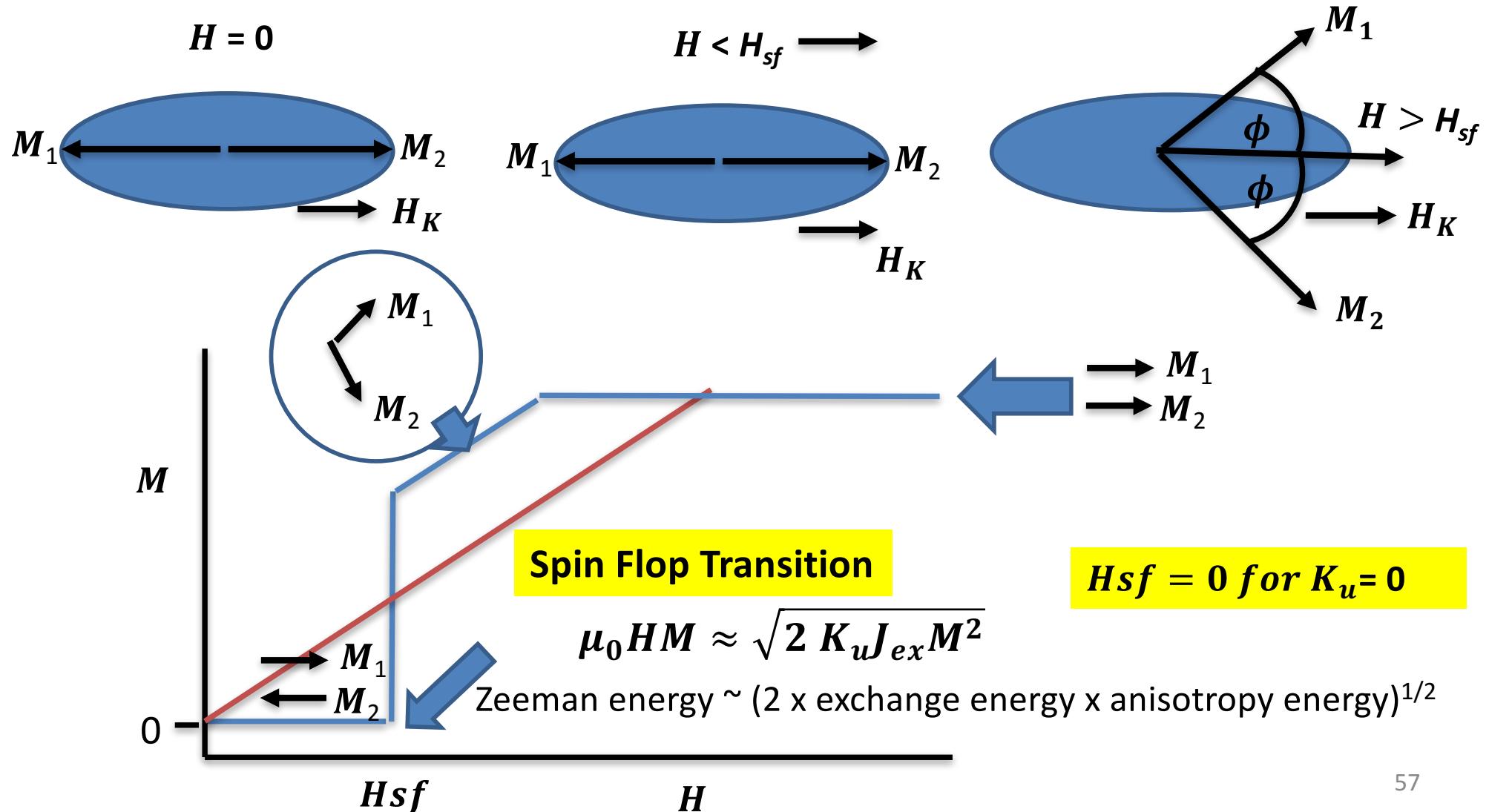


Antiferromagnets: Spin Flop Transitions

Single domain uniaxial antiferromagnet. H parallel to easy axis.

$$\epsilon = J_{ex,int} M^2 \cos(2\phi) - K_u \cos^2 \phi - \mu_0 H M \cos(\phi)$$

Small K_u

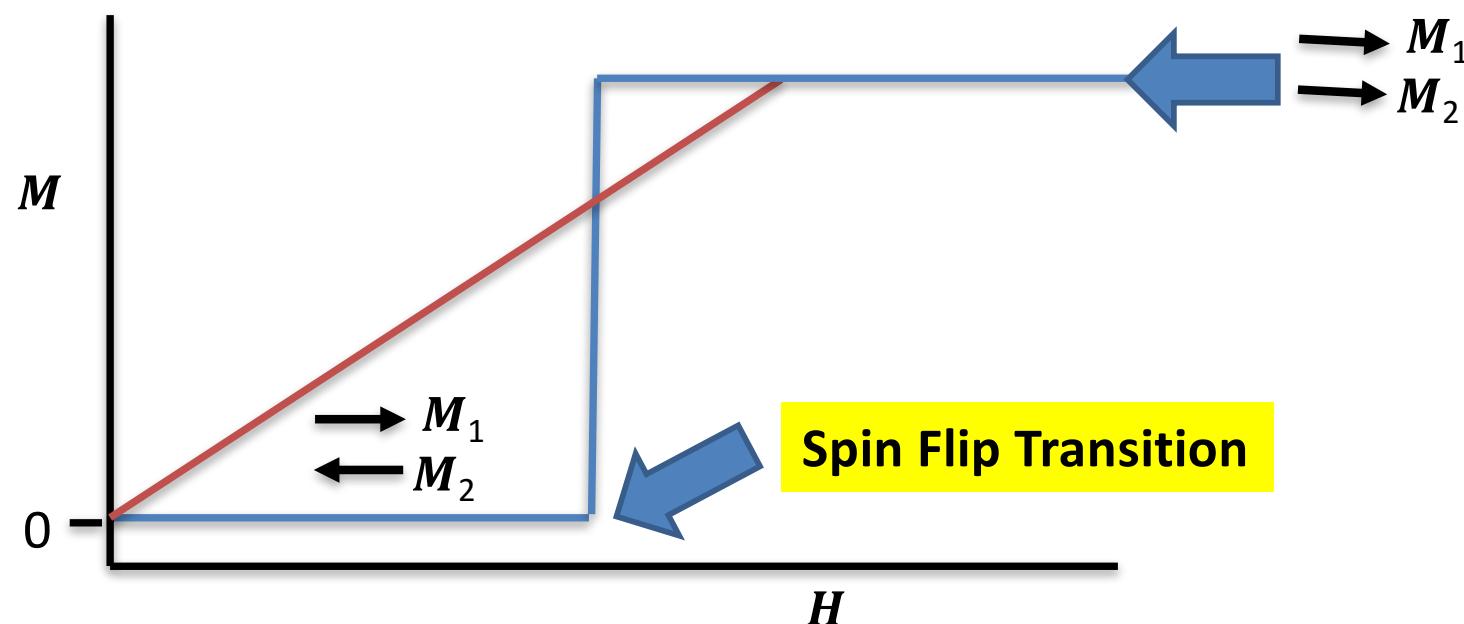
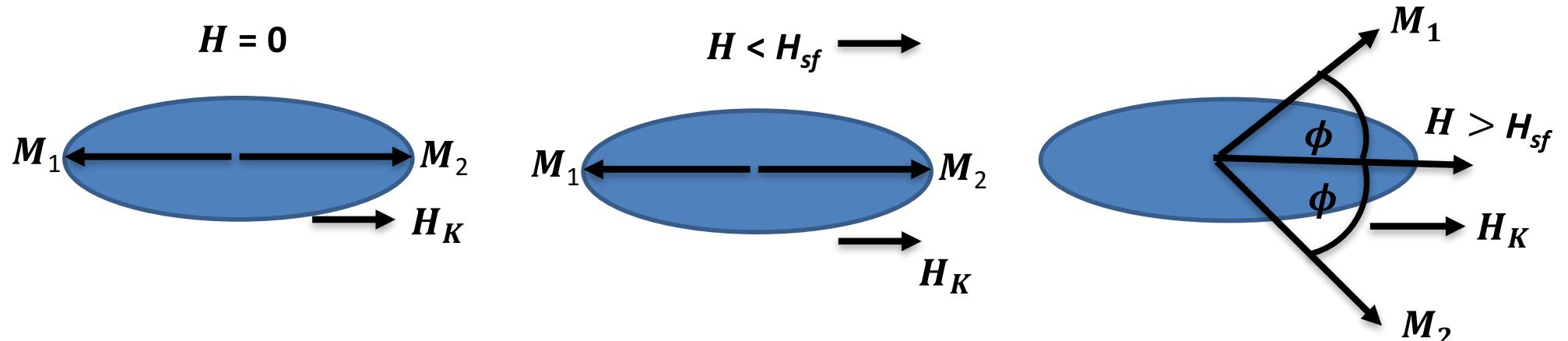


Antiferromagnets: Spin Flip Transitions

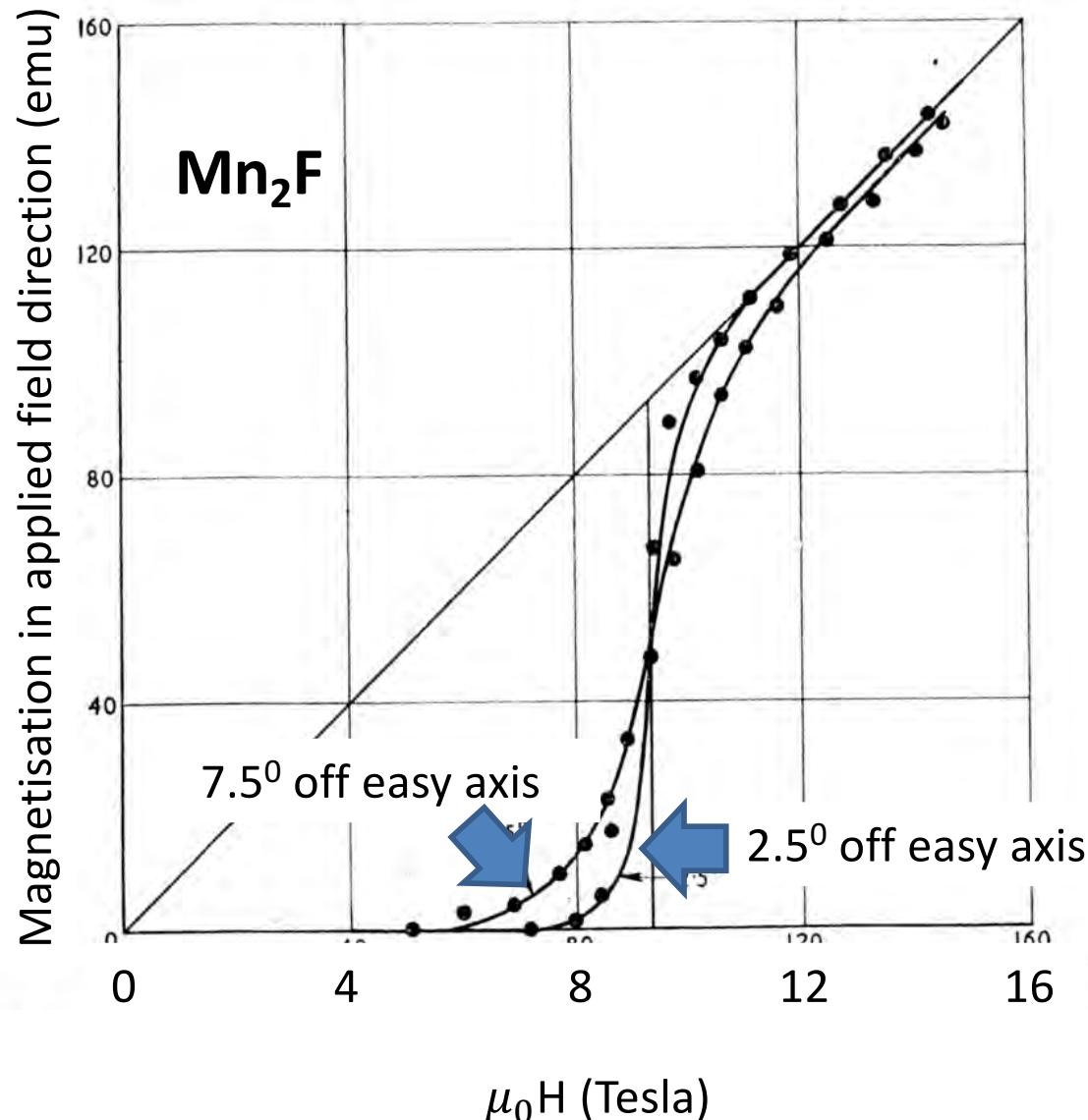
Single domain uniaxial antiferromagnet. H parallel to easy axis.

$$\epsilon = J_{ex,int} M^2 \cos(2\phi) - K_u \cos^2 \phi - \mu_0 H M \cos(\phi)$$

Large K_u



Antiferromagnets: Spin Flop Transitions



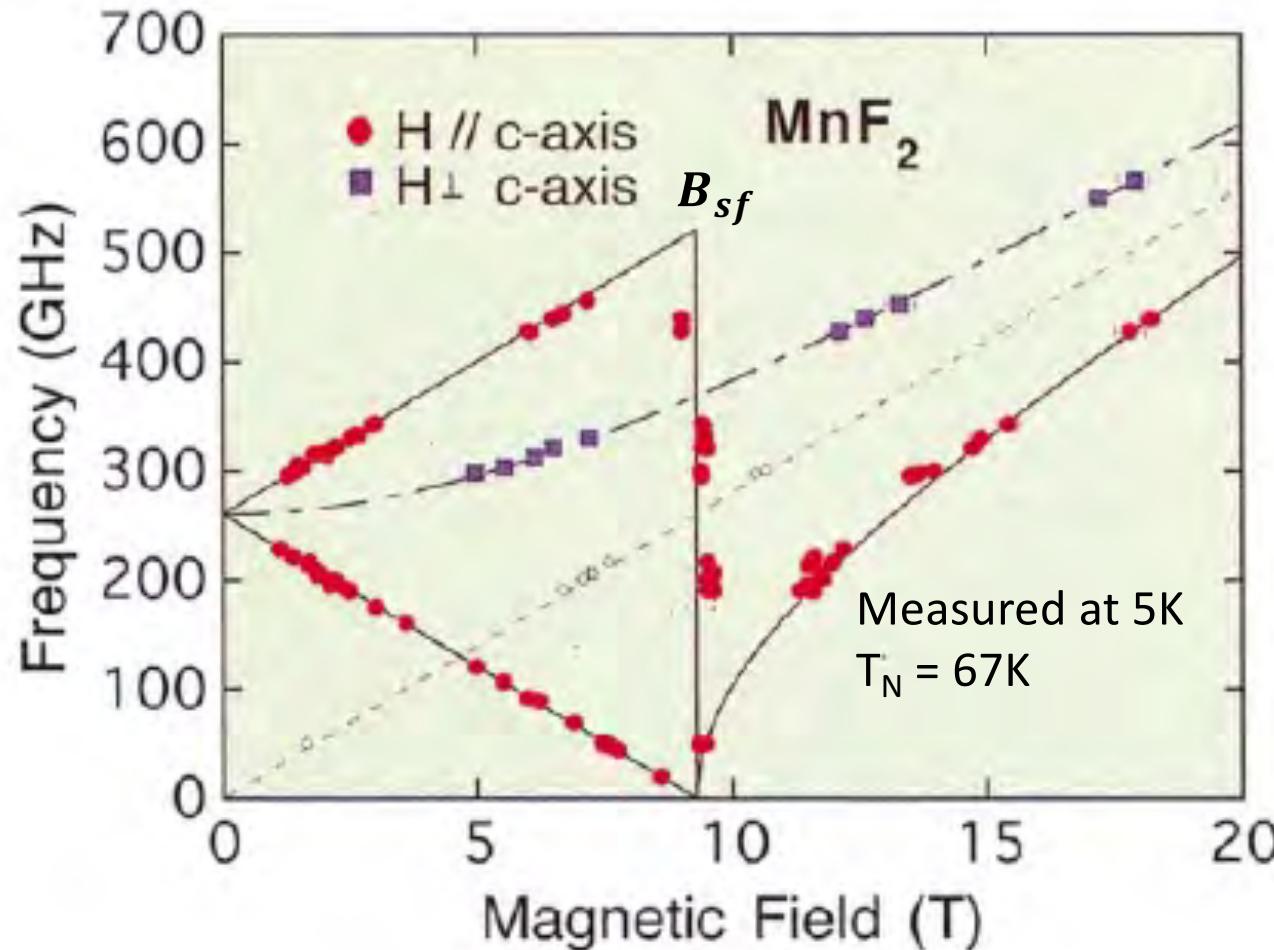
Spin Flop Transition seen in AFMR

If Exchange energy >> Anisotropy energy and $T \ll T_N$

$$H_{app} \perp c - axis \quad \hbar\omega = g\mu_B(B_{sf}^2 + B_{app}^2)^{1/2}$$

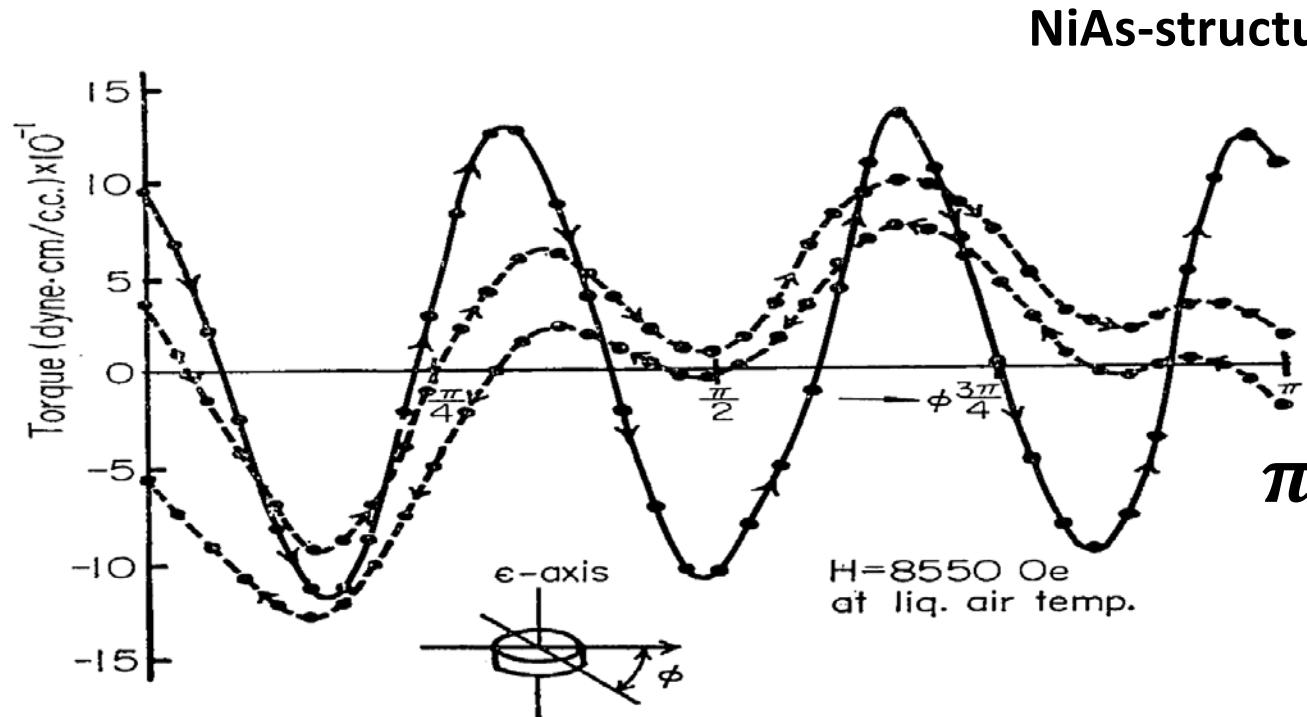
$H_{app} \parallel c - axis$ and $B_{app} < B_{sf}$ $\hbar\omega = g\mu_B(B_{sf} \pm B_{app})$ $B_{sf} = spin flop field$

$$H_{app} \parallel c - axis$$
 and $B_{app} > B_{sf}$ $\hbar\omega = g\mu_B(B_{sf}^2 - B_{app}^2)^{1/2}$



Torque Magnetometry: α -MnTe

Can give symmetry and strength of anisotropy



1 cm diameter single crystal disk.
Field rotated in c-plane
Solid line first measurements.
Others after sample deterioration

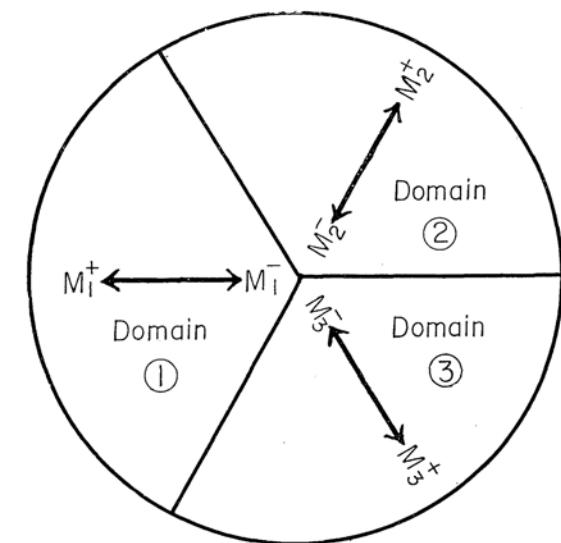
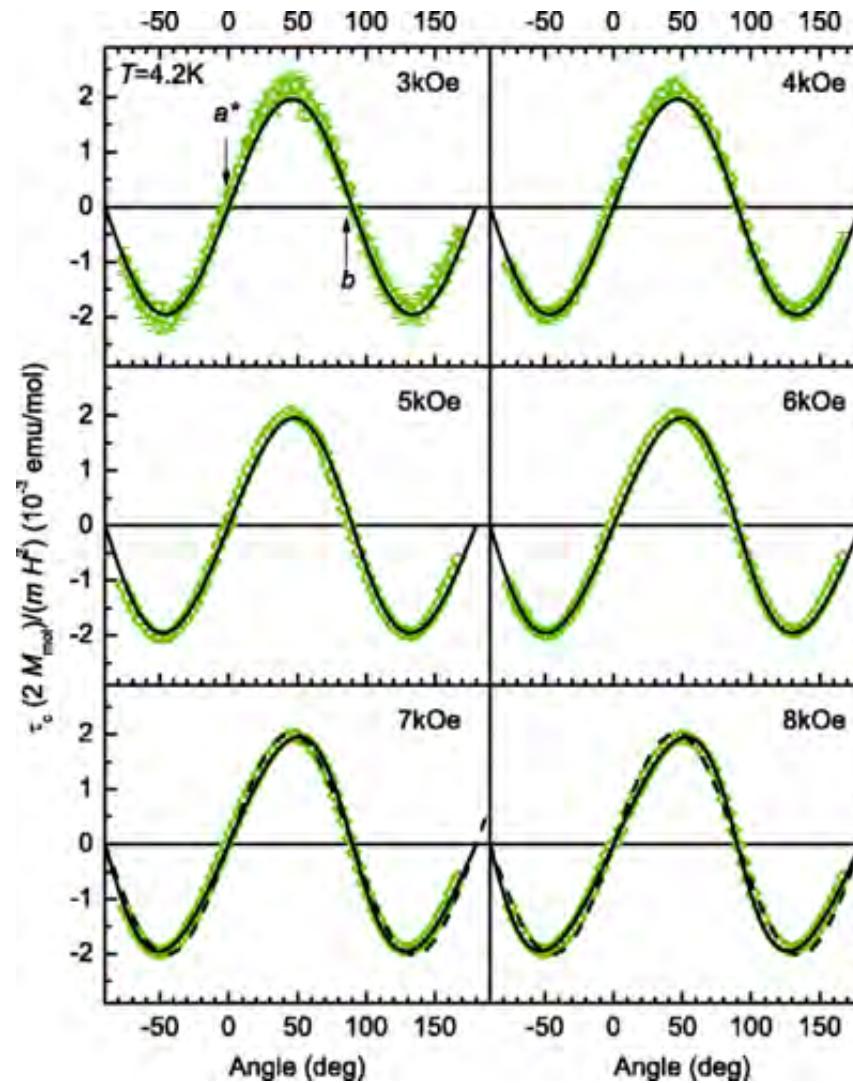


Fig. 6. The three-domain structure in the c -planes. The arrows in each domain indicate a pair of the antiferromagnetic spins, the common axis of spin pairs make angle of $2\pi/3$ with each other.

Torque Magnetometry

monoclinic quasi-one-dimensional $S = 1/2$ Heisenberg antiferromagnet CuSb_2O_6

Bulk Sample
Quite small magnetic fields

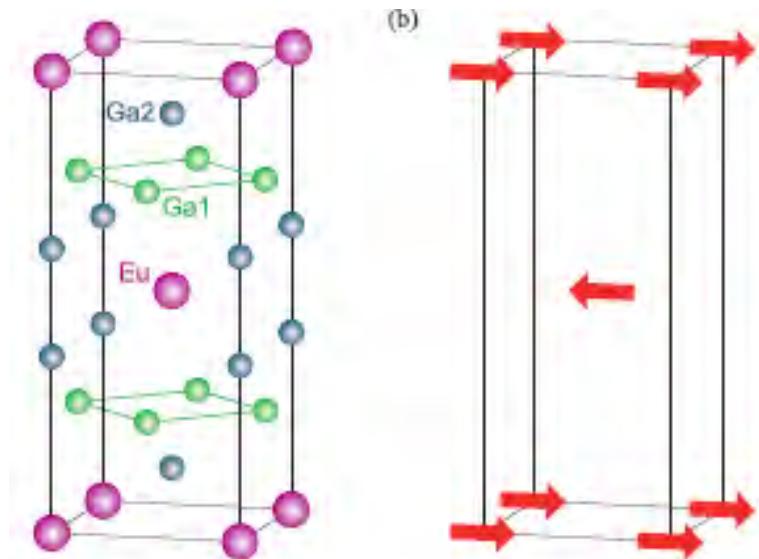
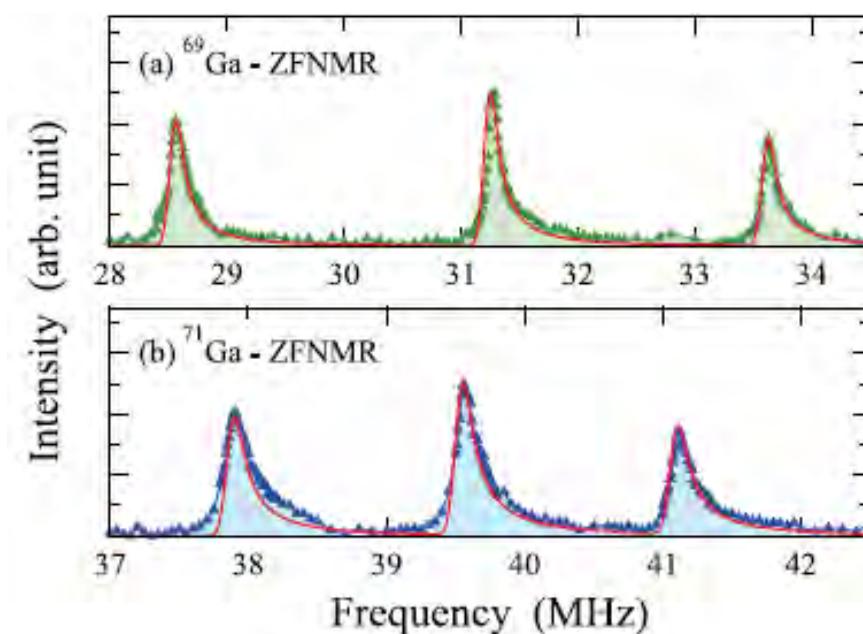
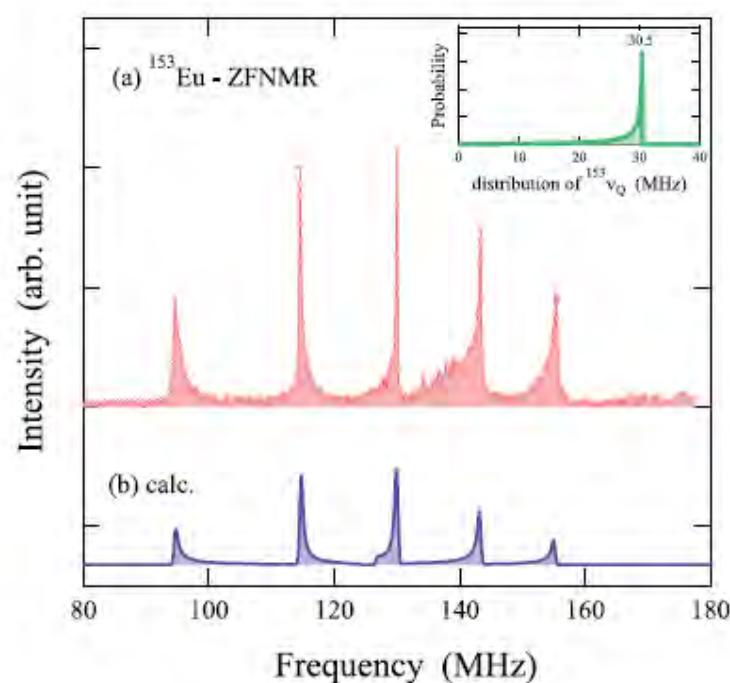


Zero Field NMR in Antiferromagnets

^{153}Eu and $^{69,71}\text{Ga}$ Zero-Field NMR Study of Antiferromagnetic State in EuGa_4

Mamoru Yogi^{1*}, Saori NAKAMURA¹, Nonoka HIGA¹, Haruo NIKI¹, Yusuke HIROSE², Yoshichika ŌNUKI², and Hisatomo HARIMA³

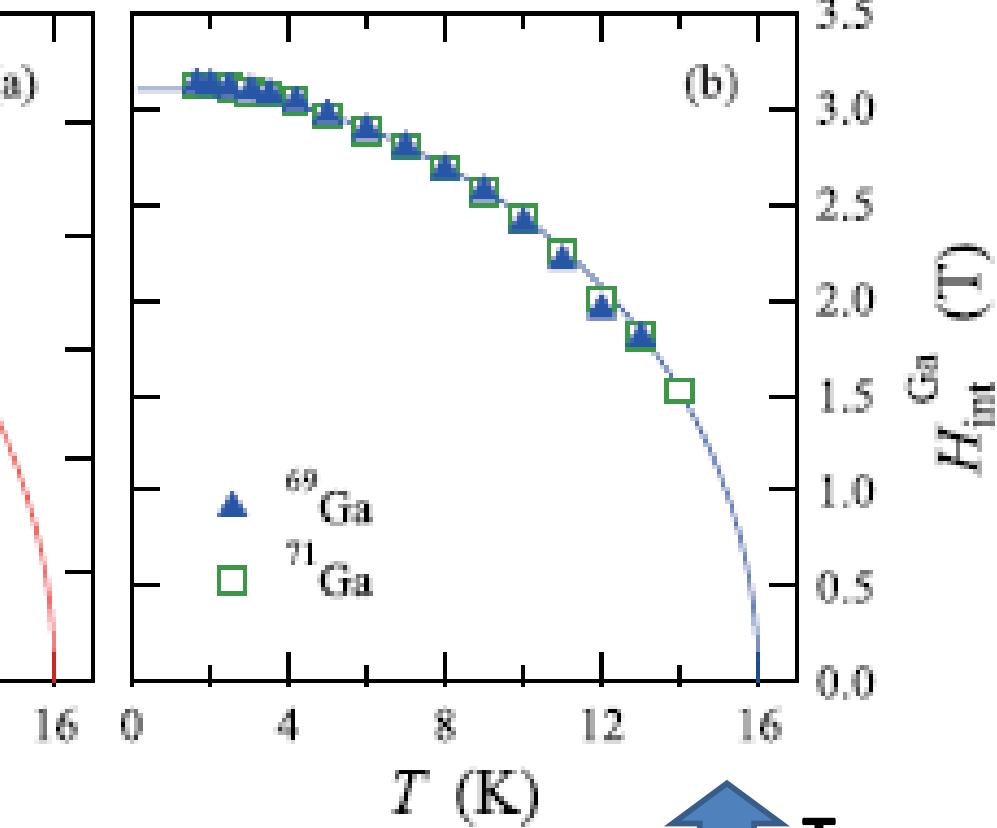
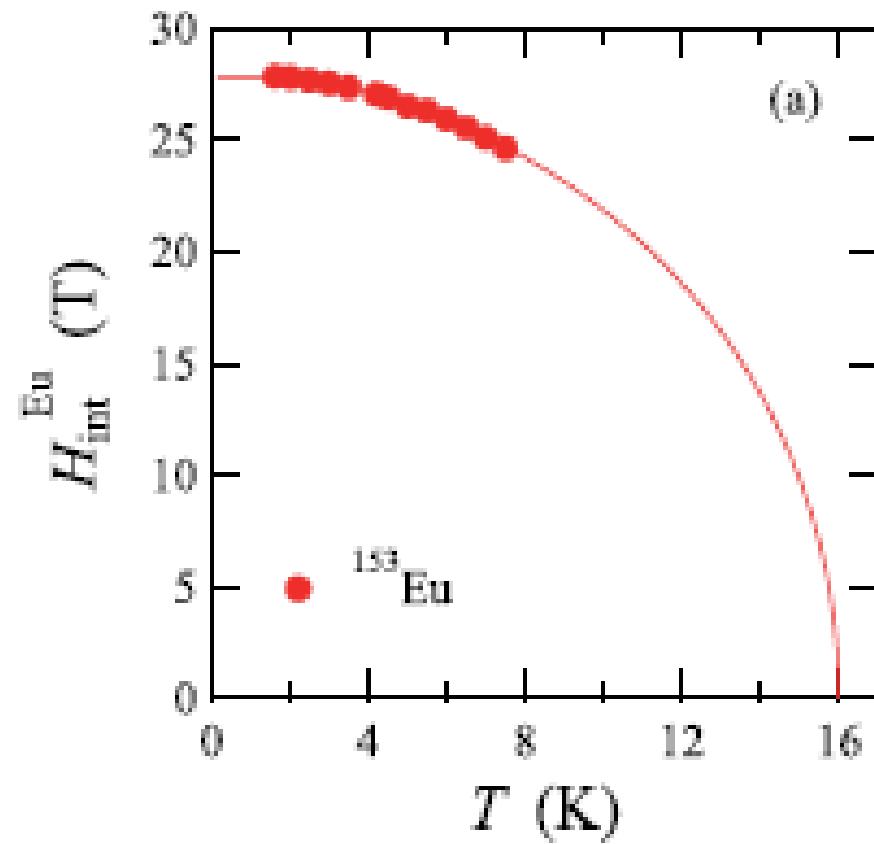
In ordered AF state the magnetic field is the same at each equivalent nuclear site \rightarrow NMR signals for zero applied field.



Antiferromagnetism : ZFNMR

^{153}Eu and $^{69,71}\text{Ga}$ Zero-Field NMR Study of Antiferromagnetic State in EuGa_4

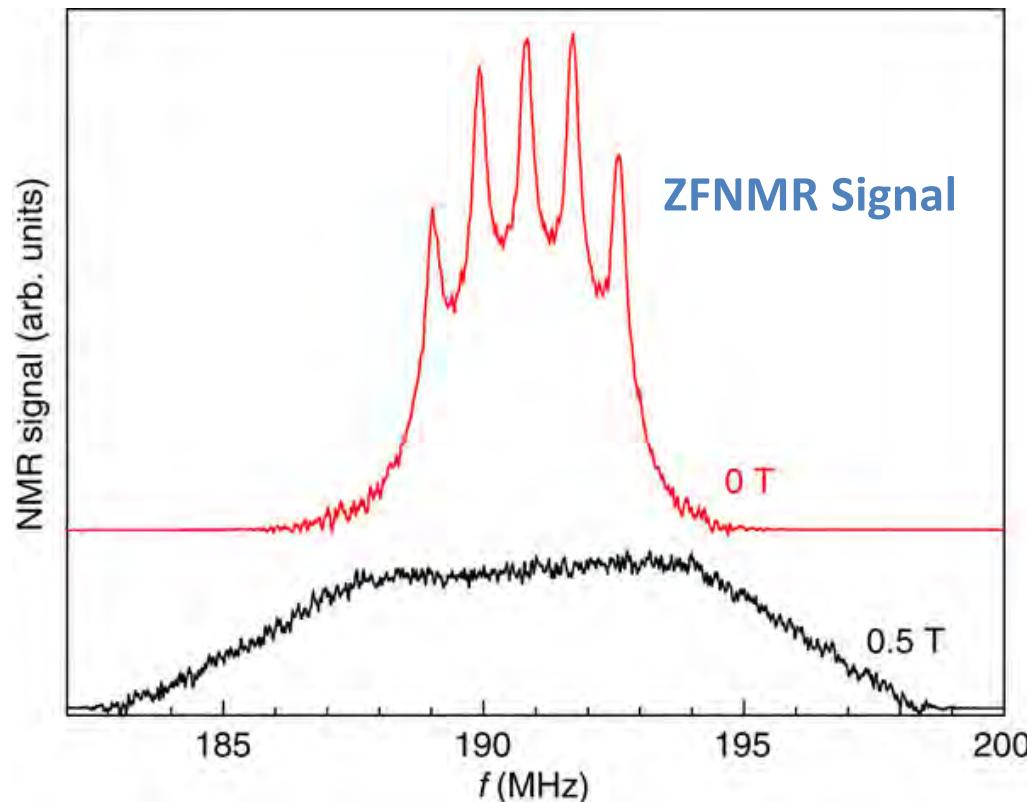
Mamoru Yogi^{1*}, Saori NAKAMURA¹, Nonoka HIGA¹, Haruo NIKI¹, Yusuke HIROSE², Yoshichika ŌNUKI², and Hisatomo HARIMA³



But need to find resonances!

Antiferromagnetism : ZFNMR

Polycrystalline Mn_2Au

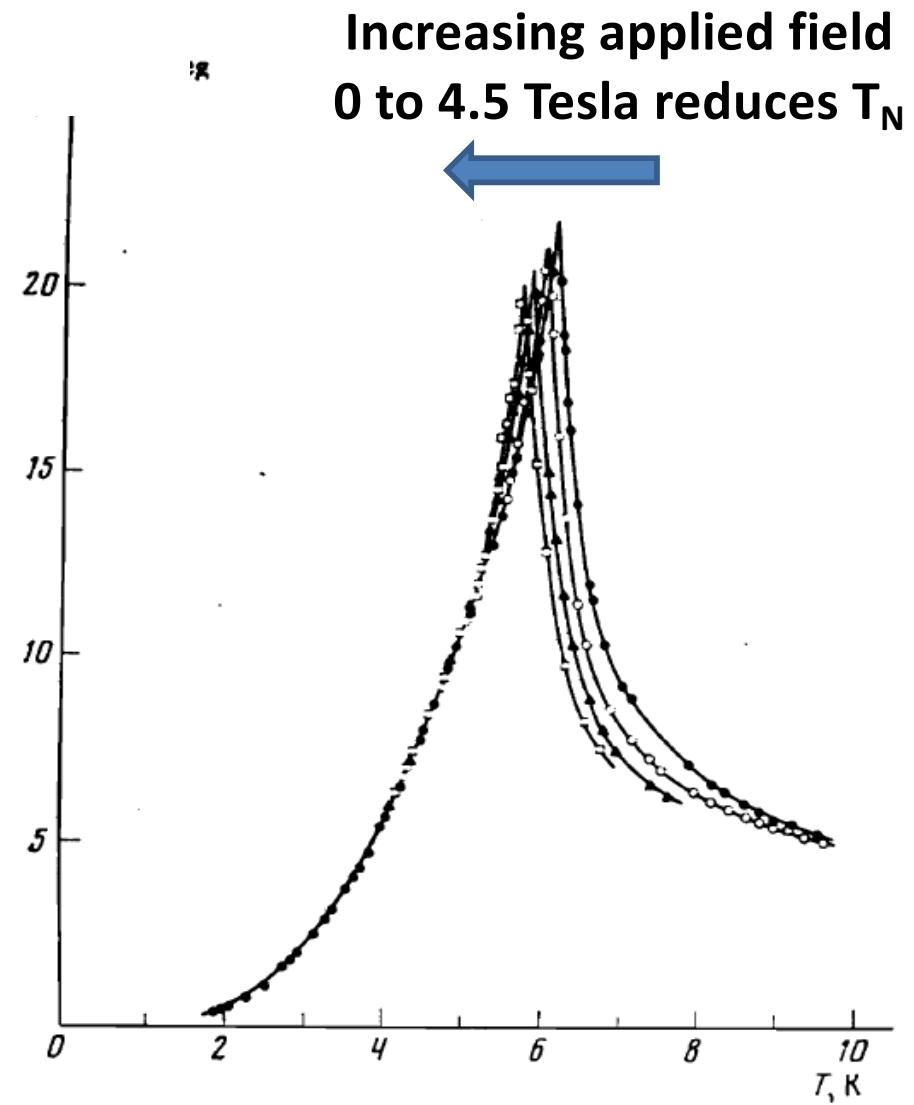
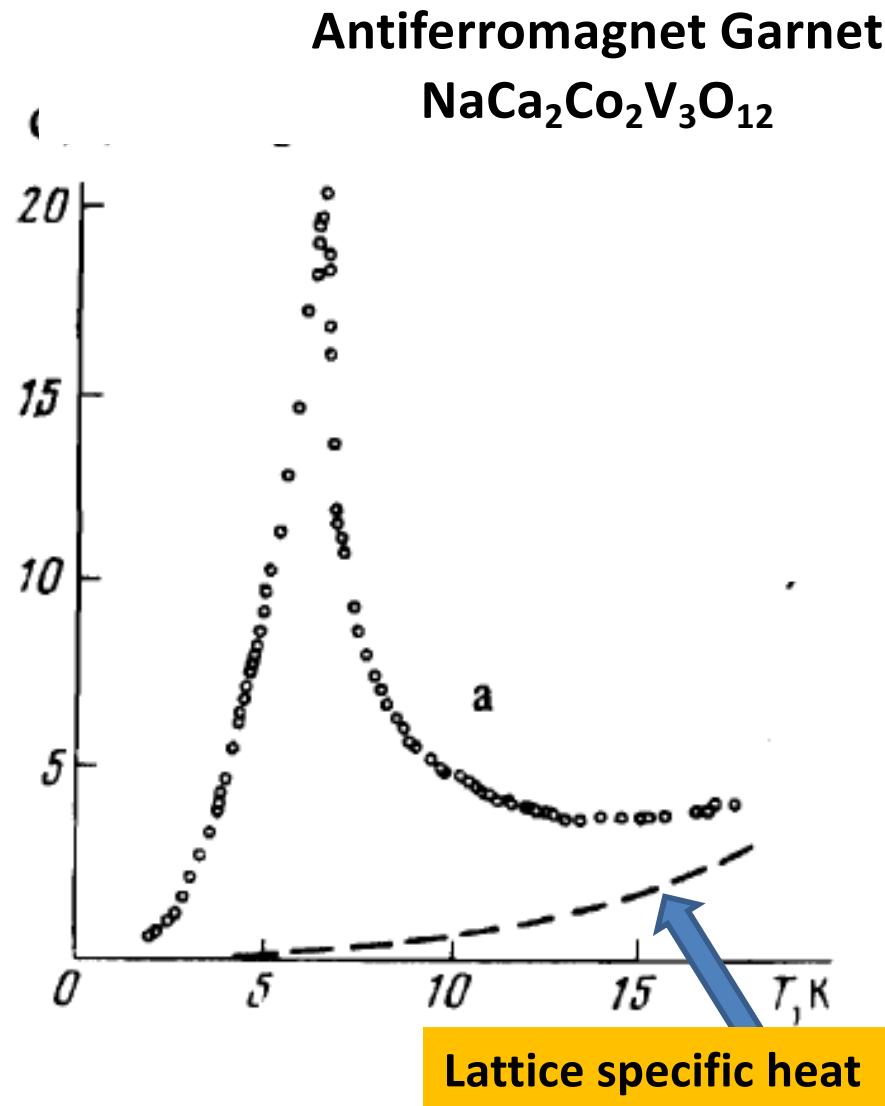


Applying a magnetic field of $\mu_0 H_{\text{app}} = 0.5 \text{ T}$ changes the five-peak ZFNMR spectrum into an unresolved broad spectrum extending.

In the applied field equivalent sites in the constituent grains experience different fields because of the distribution of angles between internal and applied fields. The total field experienced by ^{55}Mn nuclei takes all values between $H_{\text{hyp}} + H_{\text{app}}$ and $-H_{\text{hyp}} + H_{\text{app}}$.

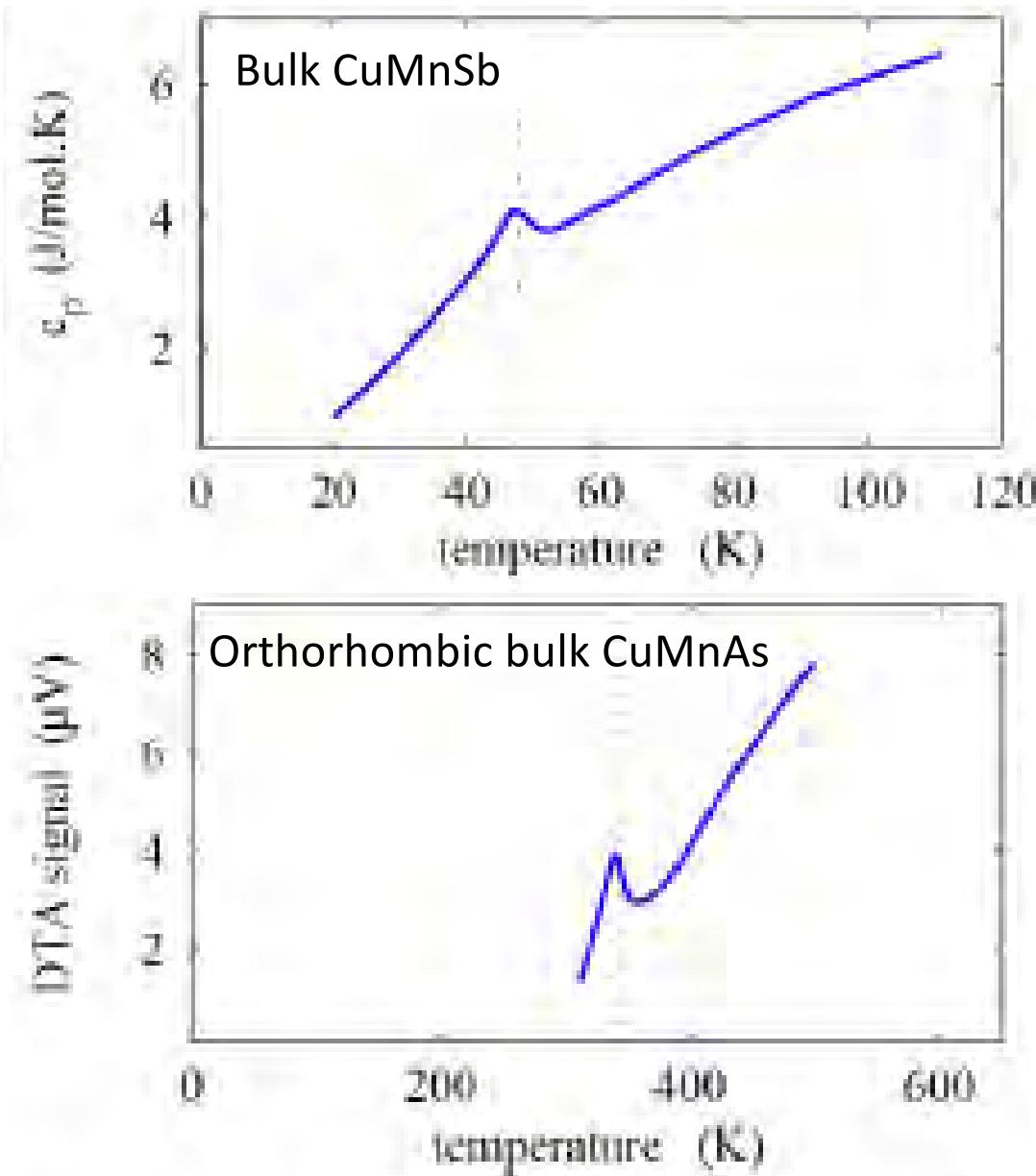
4.1 Critical behaviour: Specific Heat

Second order phase transition. Specific heat diverges at critical



Antiferromagnets : Specific Heat

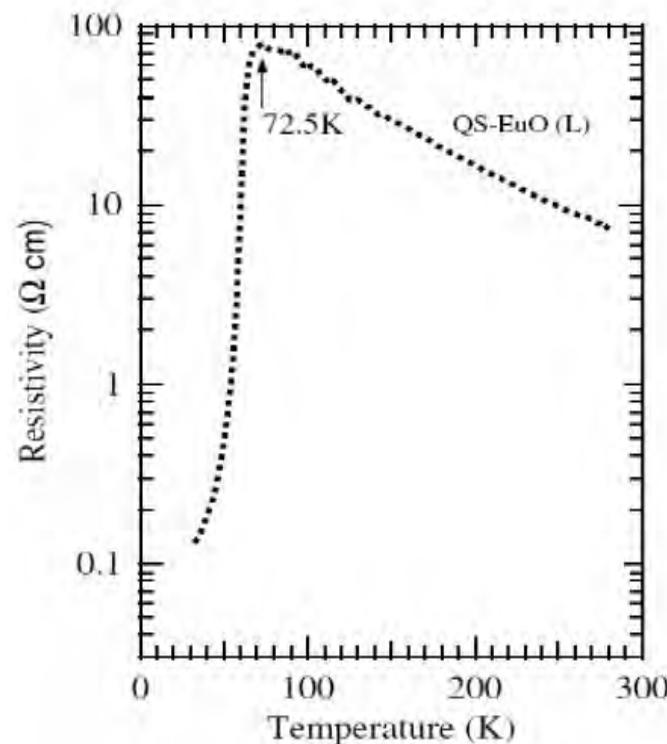
Measurement much more difficult at higher temperatures



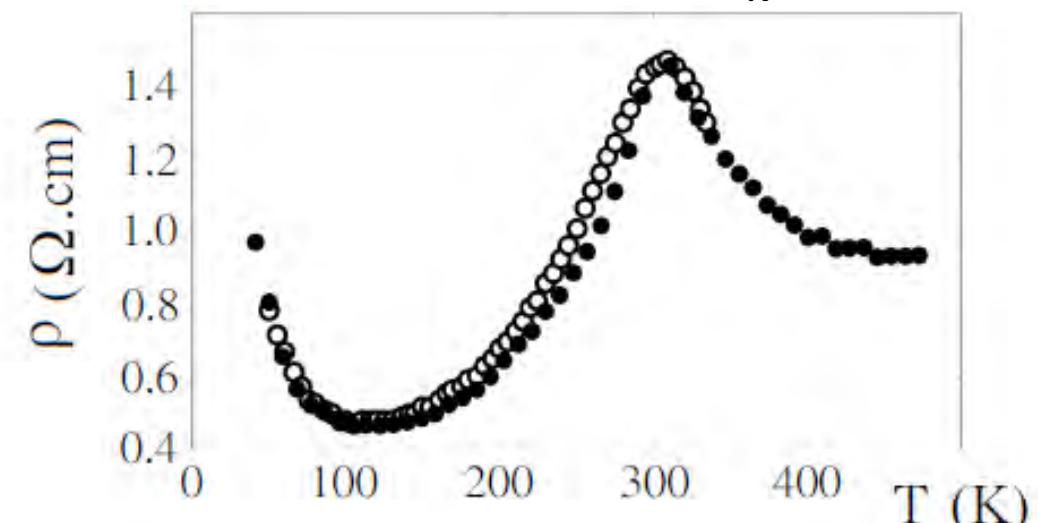
4.2 Critical Behaviour: Resistivity

For dense moment magnetic semiconductors the moment density \gg carrier density. Only long wavelength spin fluctuations scatter carriers effectively [1] and the **mobility** has a peak at T_c / T_N with the same form as the susceptibility [2].

FM semiconductor EuO $T_N = 73\text{K}$



AF semiconductor MnTe $T_N = 310\text{K}$



Y. Magnin and H. T. Diep Cond - Mat

Carrier density \ll moment density
See clear peak in resistivity at $\sim T_N$

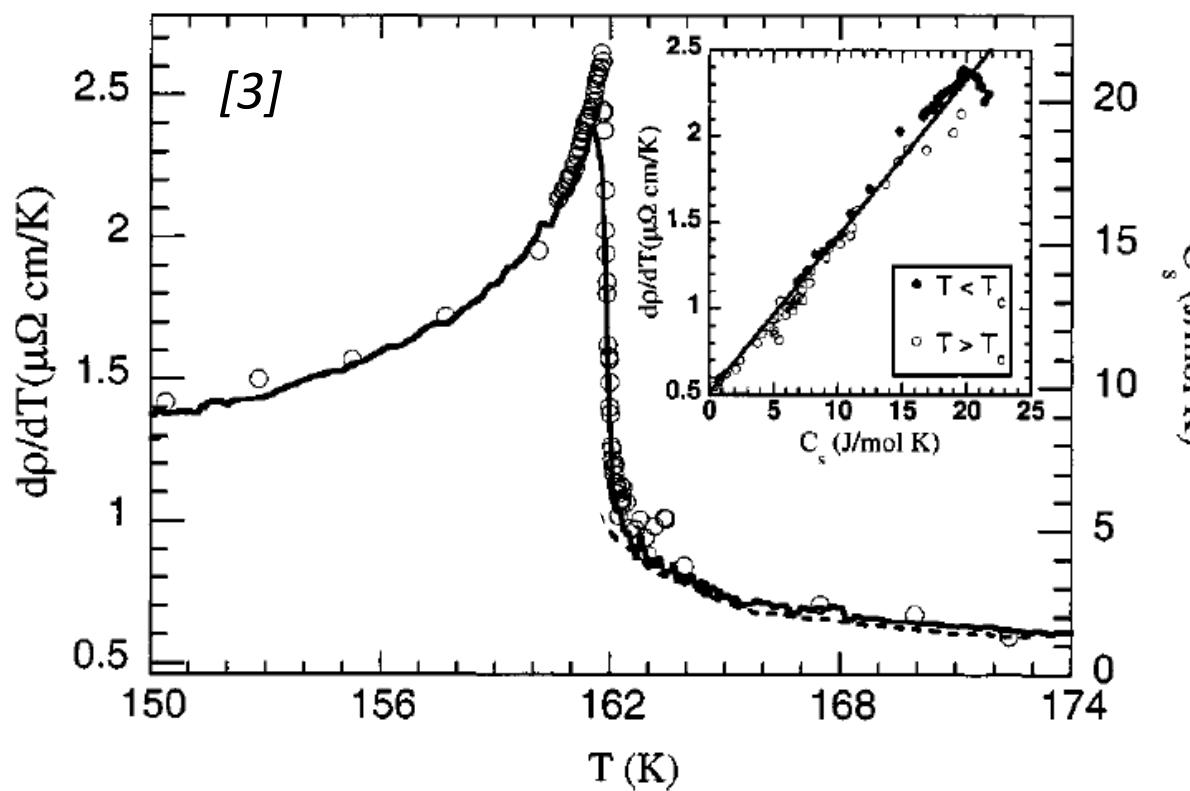
[1] de Gennes and Friedel, *J. Phys. Chem. Solids* **4**, 71 (1958)

[2] Haas, *Crit. Rev. Solid State Sci.* **1**, 47 (1970)

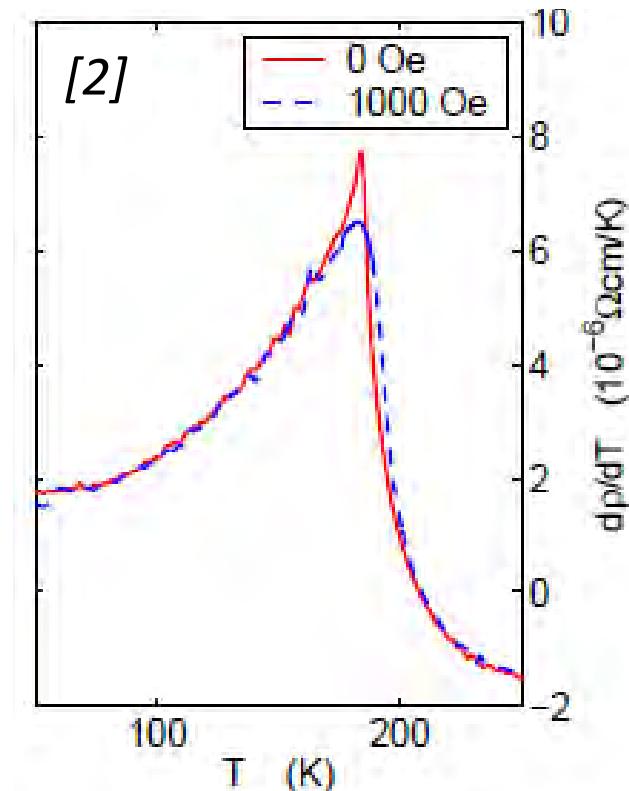
4.2 Critical Behaviour: Resistivity

In ferromagnetic metals [1] and dilute ferromagnetic semiconductors [2] carrier density \sim moment density. Critical fluctuations produce a peak in $d\rho_{xx}/dT$ at T_c with same form as specific heat [3].

FM Metal SrRuO_3



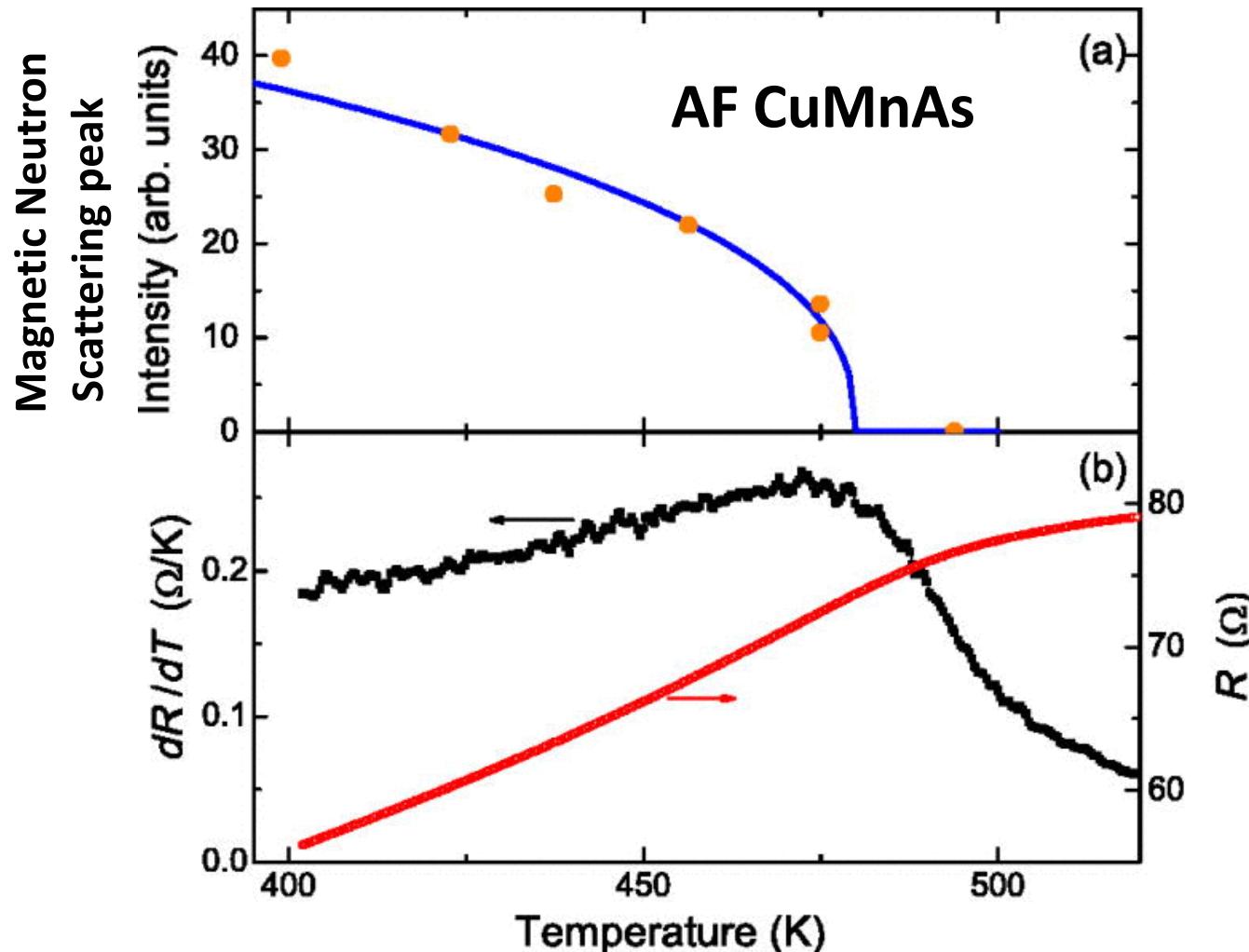
Dilute FM semiconductor
 $(\text{Ga}_{88}\text{Mn}_{12})\text{As}$



[1] Craig et al *Phys. Rev. Lett.* **19**, 1334 (1967); Shacklette , et al , *Phys. Rev. B* **9**, 3789 (1974)

[2] Novák, et al *Phys. Rev. Lett.* **101**, 077201 (2008) [3] Kim, et al *Phys. Rev. B* **67**, 100406(R) (2003)

Critical Behaviour: Resistivity



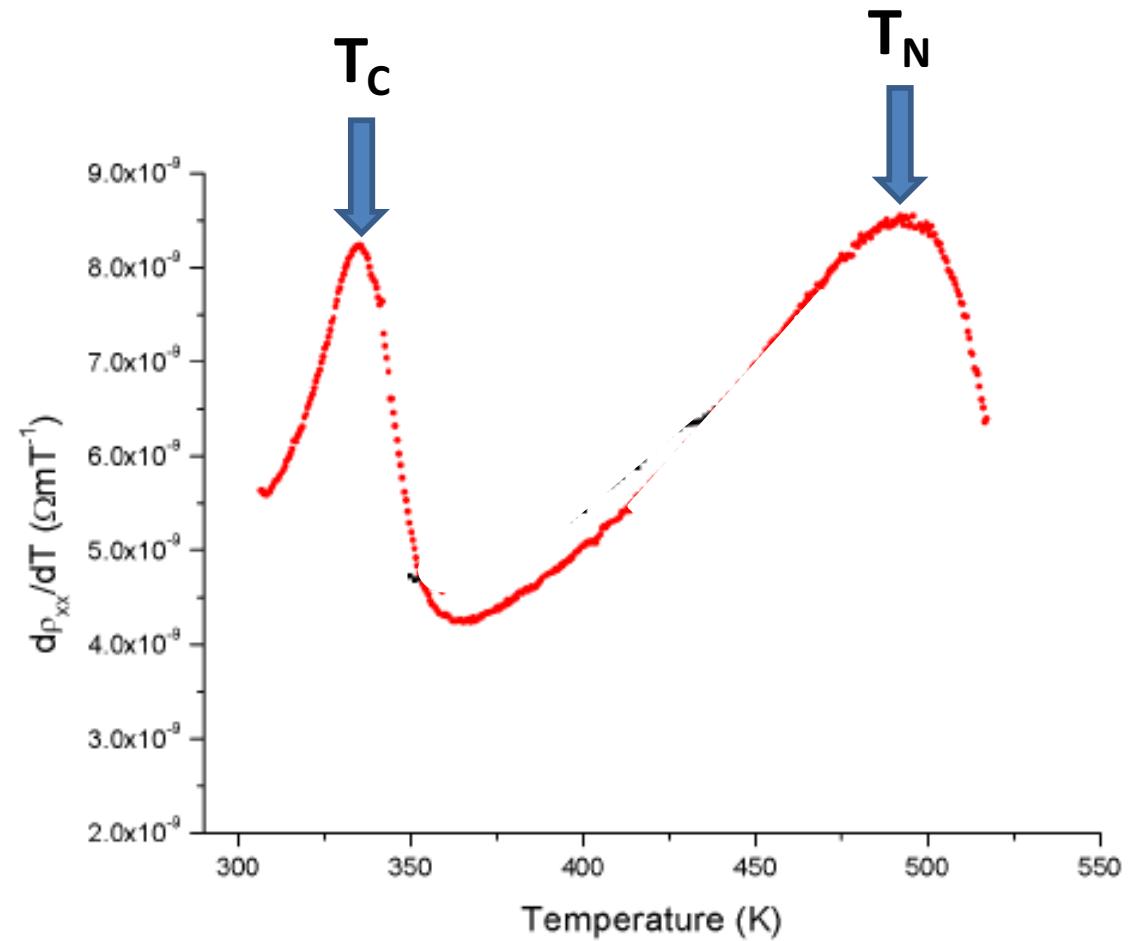
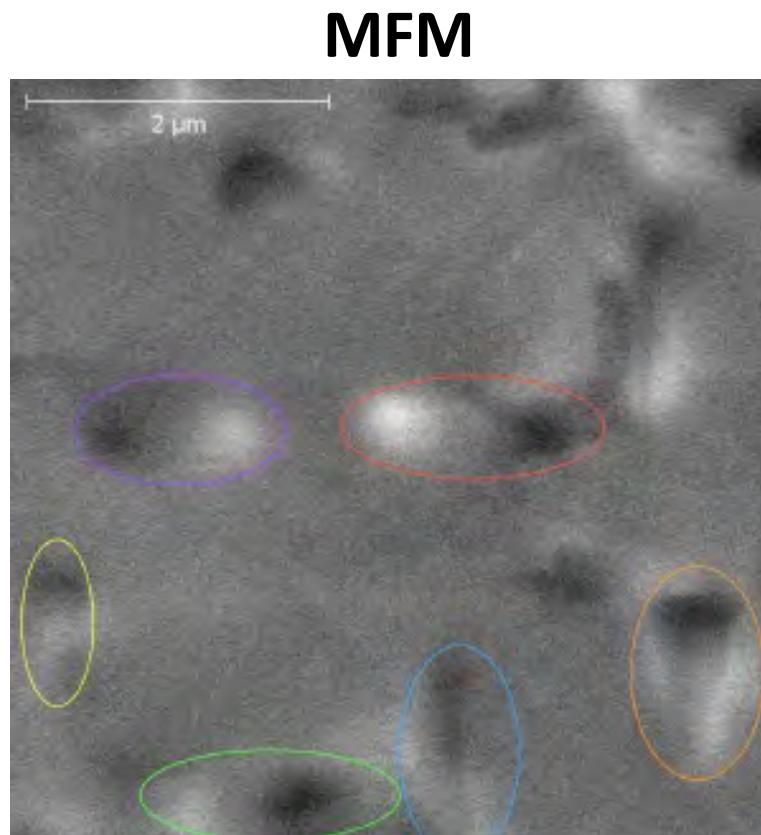
CuMnAs AF semimetal (?)

Carrier density \sim moment density

See clear peak in dR/dT at $\sim T_N$

AF CuMnAs with FM MnAs inclusions

CuMnAs grown with excess Mn producing
~15% by volume FM MnAs in CuMnAs matrix



4.3 Critical Exponents

Close to a second order phase transition power law relationships between thermodynamic variables exist.

Ferromagnet **M** is order parameter

$$M \propto t^\beta \text{ where } t = (1-T/T_c) \text{ for } T < T_c$$

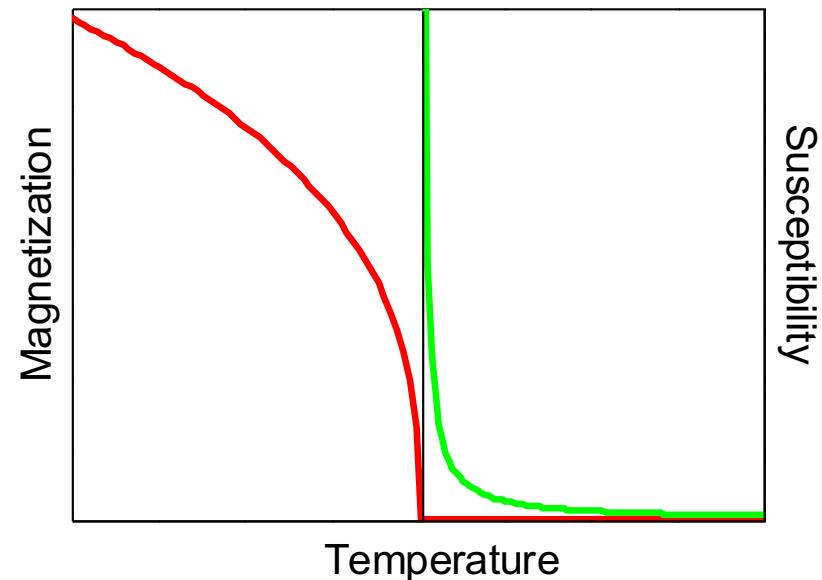
$$\chi \propto (-t)^{-\gamma} \text{ for } T > T_c$$

$$C_V \propto |t|^{-\alpha}$$

Antiferromagnet **L** = $M_1 - M_2$ is order parameter

$$L \propto t^\beta \text{ where } t = (1-T/T_N) \text{ for } T < T_N$$

$$C_V \propto |t|^{-|\alpha|}$$



Only C_V is easy to measured accurately in an antiferromagnet.

Critical Exponents

The values of the exponents close to the critical temperature often depend only on the dimensionality and symmetry of the system and belong to a small number of **universality classes**.

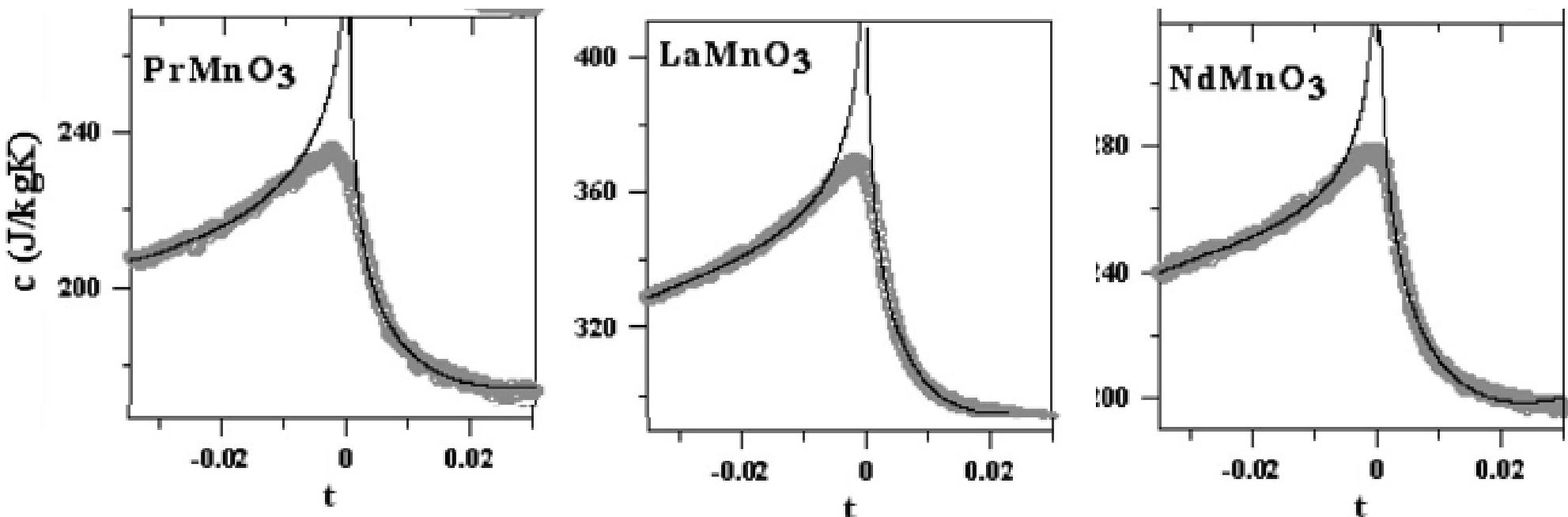
	α	β	γ
Mean Field	0	0.5	1.0
Ising	0.11	0.326	1.237
3D Heisenberg	-0.134	0.369	1.396

Determining critical exponents can give information on dimensionality of moment system and range of exchange interactions.

FM (Ga,Mn)As	??	0.366 +/- 0.003	1.47 +/- 0.05
--------------	----	--------------------	------------------

Three-dimensional Heisenberg critical behaviour in the highly disordered dilute ferromagnetic semiconductor (Ga,Mn)As M. Wang, R. A. Marshall, K. W. Edmonds, A. W. Rushforth, R. P. Campion, and B. L. Gallagher Physical Review B 93, 184417 (2016)

Critical Exponents of Antiferromagnets



Fitting functions used to obtain critical exponent α

$$c = B + C(T - T_N) + A^+|T - T_N|^{-\alpha}(1 + E^+|T - T_N|^{0.5}) \quad T > T_N$$

$$c = B + C(T - T_N) + A^-|T - T_N|^{-\alpha}(1 + E^-|T - T_N|^{0.5}) \quad T < T_N$$

Eight free parameters plus choice of fitting temperature range

Conclude $\alpha = -0.11 \pm 0.01$ 3D Heisenberg value

5. X-ray magnetic dichroism

Peter Wadley tomorrow



Lab x-ray source

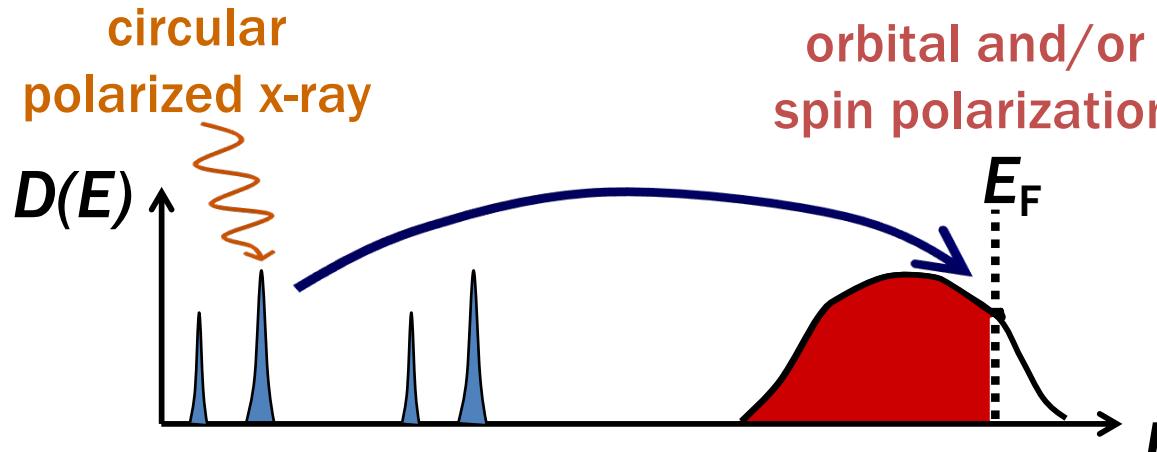
- fixed energy (*e.g.*, Cu K α)
- unpolarized



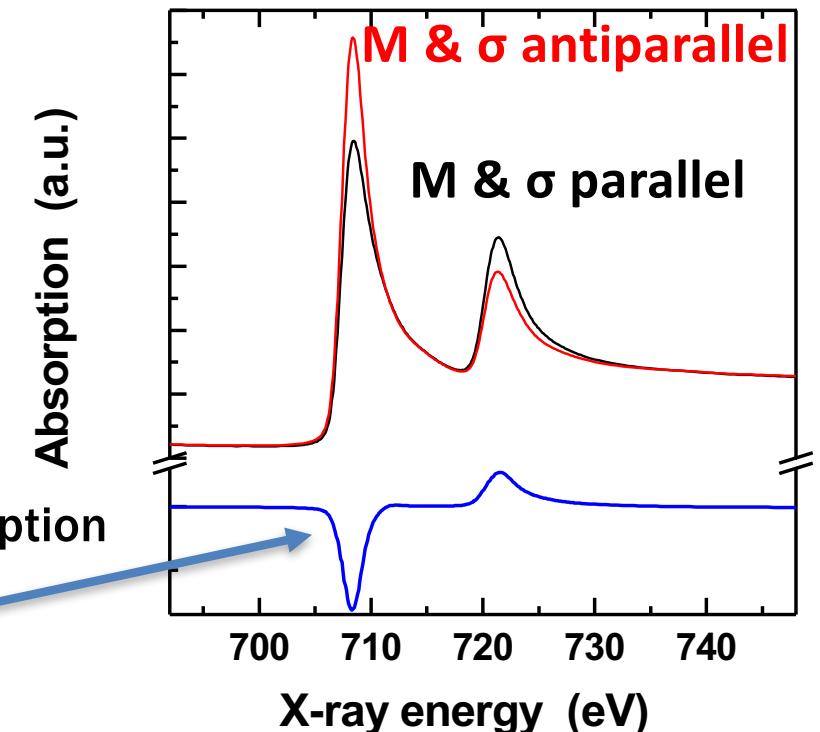
Synchrotron x-ray source

- controllable energy (sub-eV → 100s of keV)
- high flux, high coherence
- controllable polarization (linear or circular)
- **BUT** limited availability

X-ray Magnetic Circular Dichroism (XMCD)



Magnetic Dichroism: Polarization Dependent Absorption

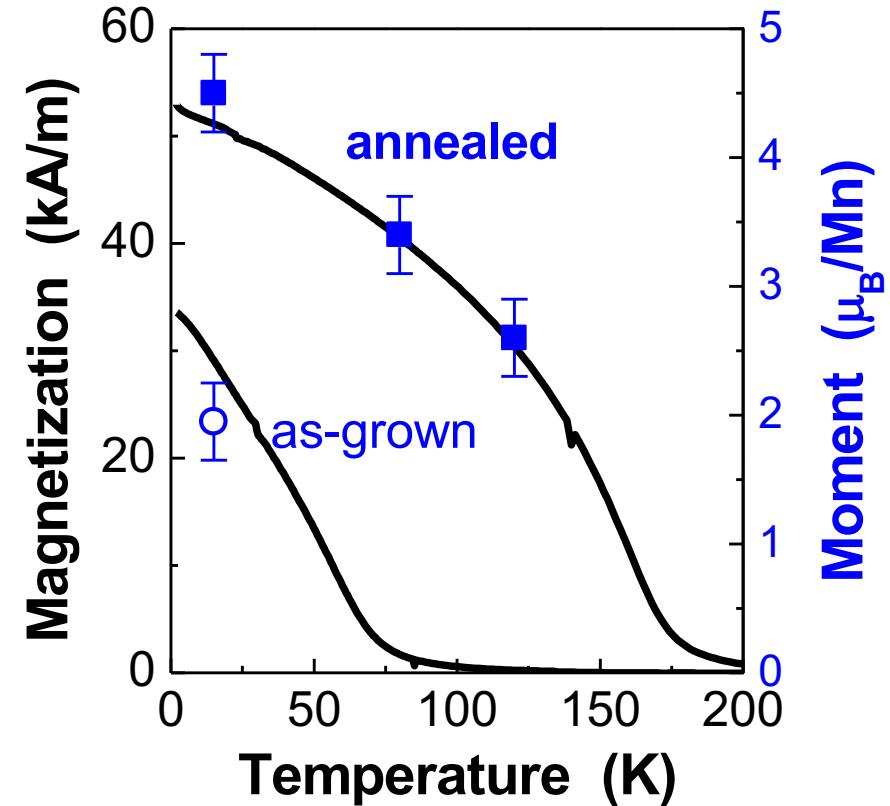
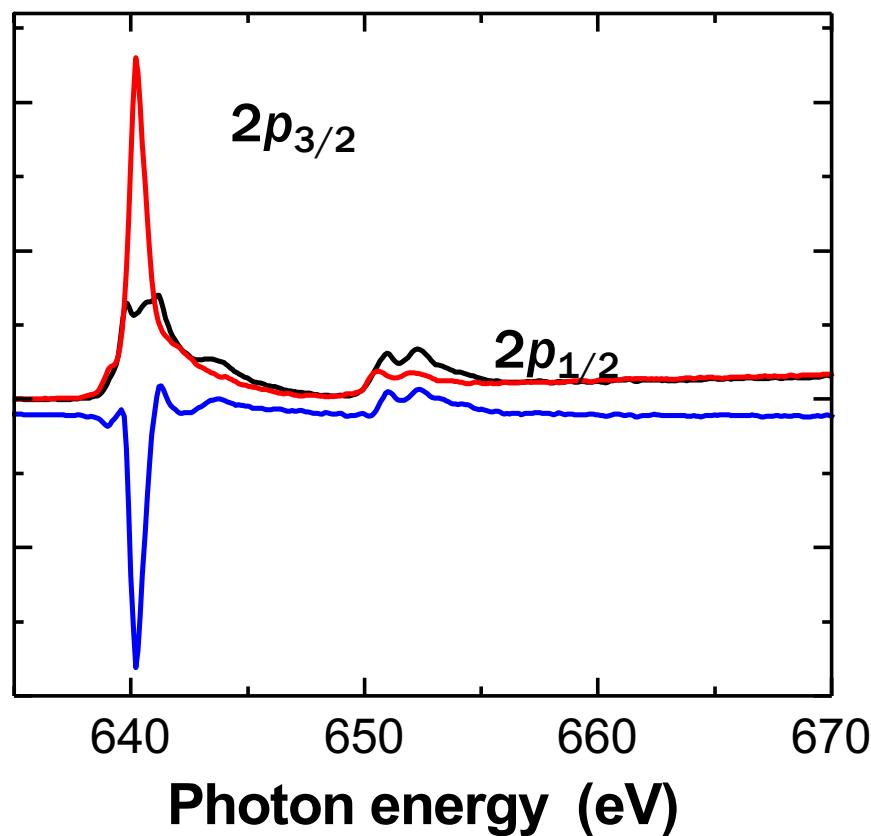


Difference = x-ray magnetic circular dichroism (XMCD)

- Probes unfilled local density of states around excited atom
- Combined chemical and magnetic probe
- Element-specific magnetometer
- Depth-dependent information
- Semi-quantitative (orbital and spin magnetic moments per atom)⁷⁶

X-ray Magnetic Circular Dichroism (XMCD)

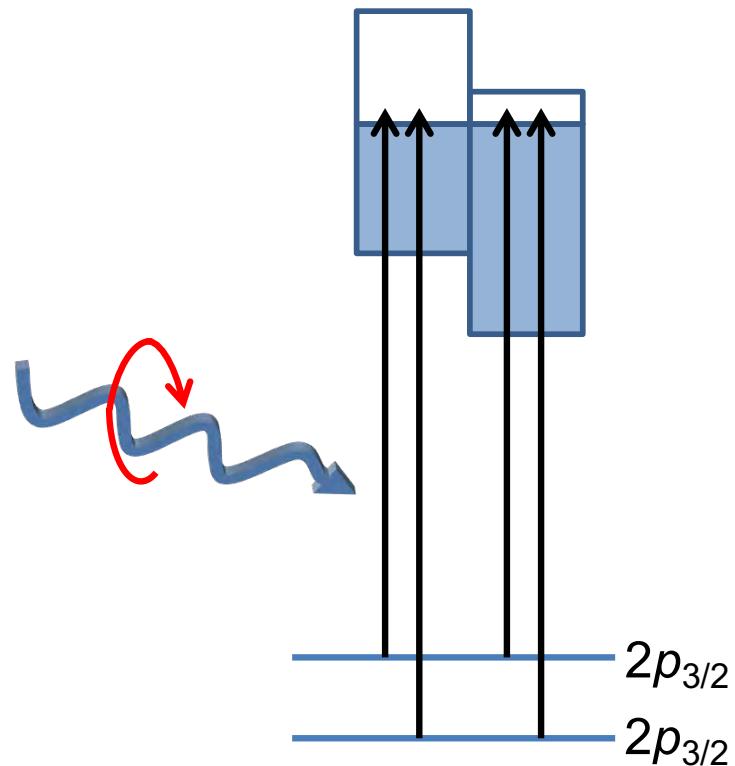
FM (Ga,Mn)As
 $2p \rightarrow 3d$ transitions



Sum rules: integrated absorption \rightarrow orbital and spin magnetic moments

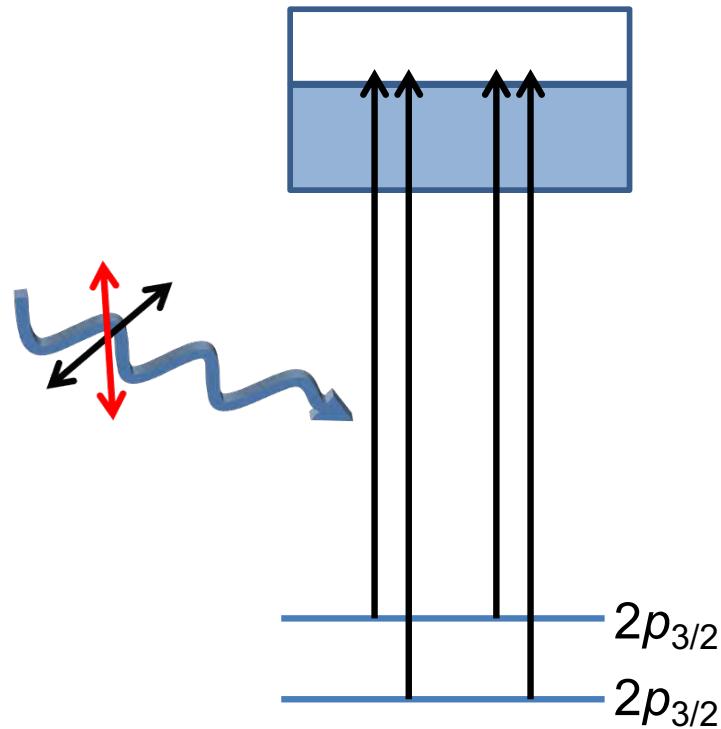
X-ray Magnetic Circular Dichroism (XMCD)

X-ray magnetic circular dichroism (XMCD)



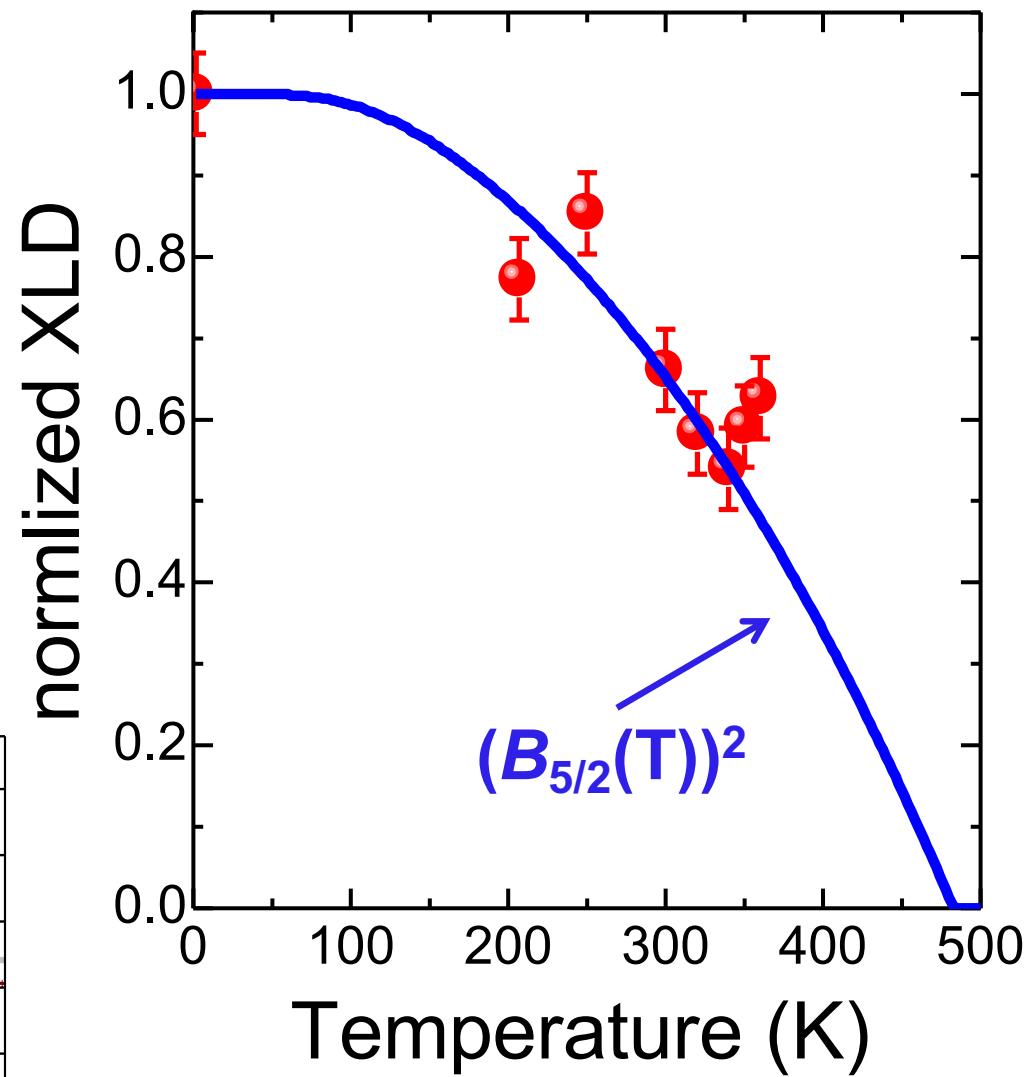
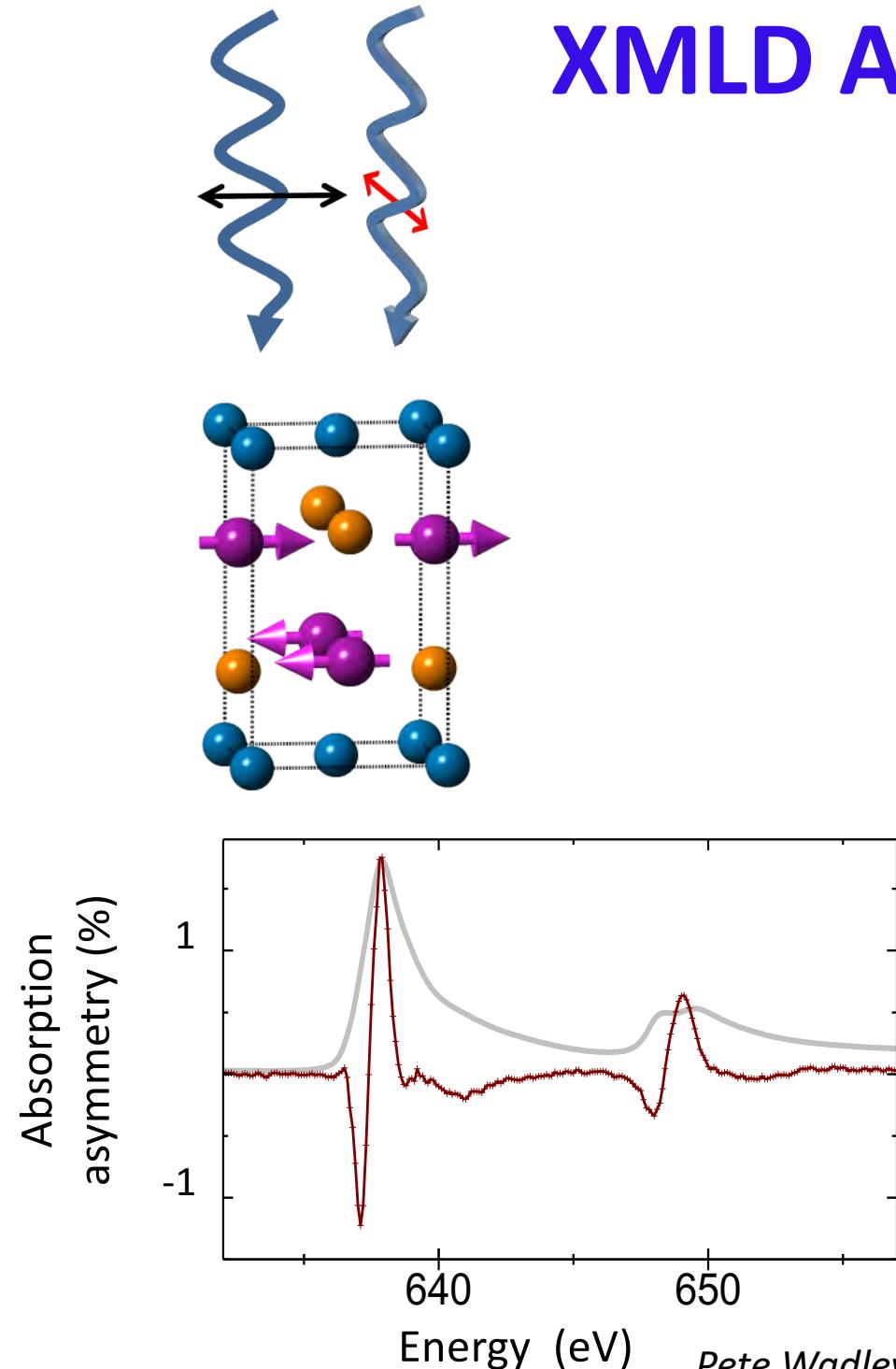
Sensitive to unpaired spin or orbital magnetic moments, *i.e.* ferromagnetic order

X-ray linear dichroism (XLD)

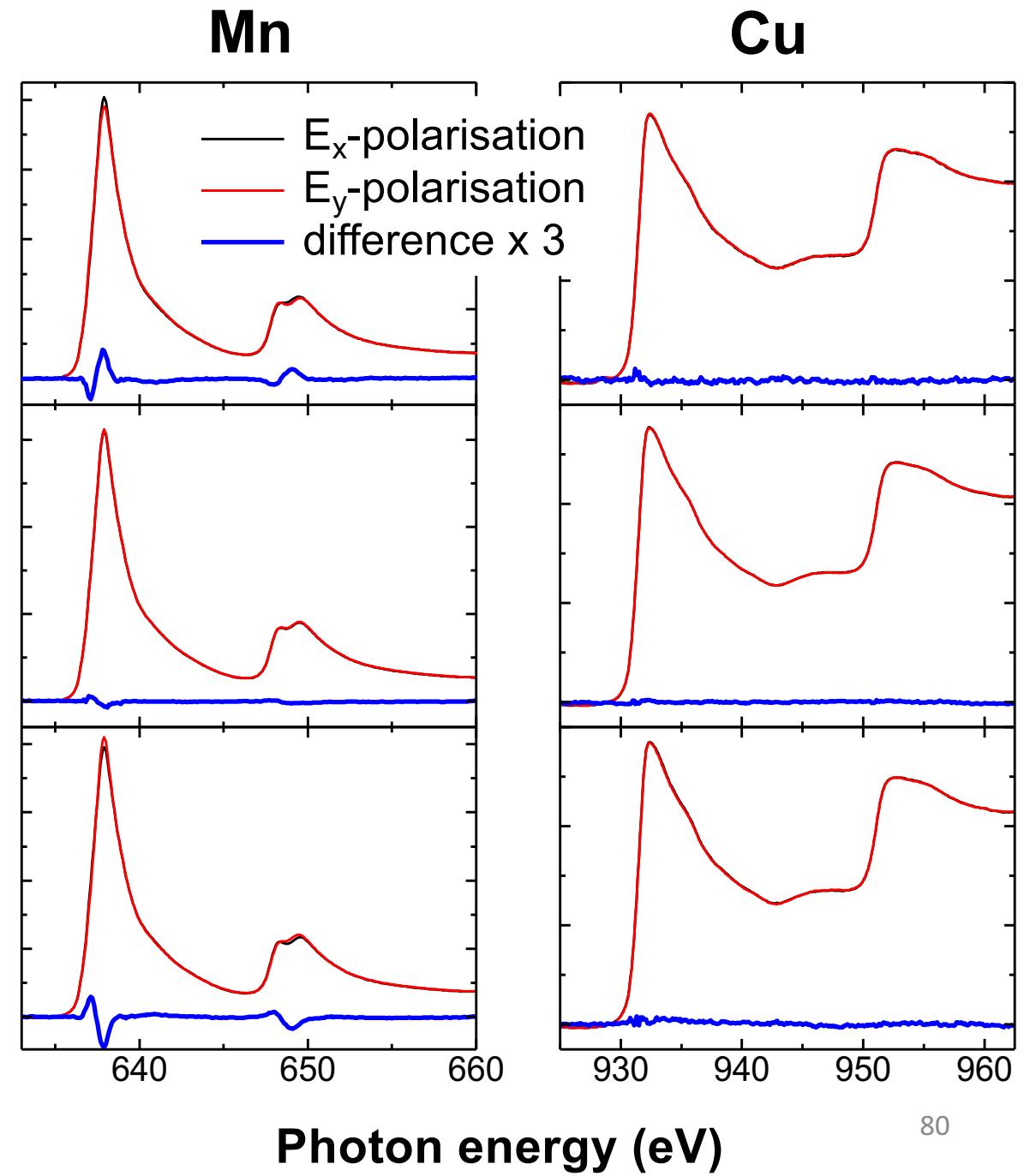
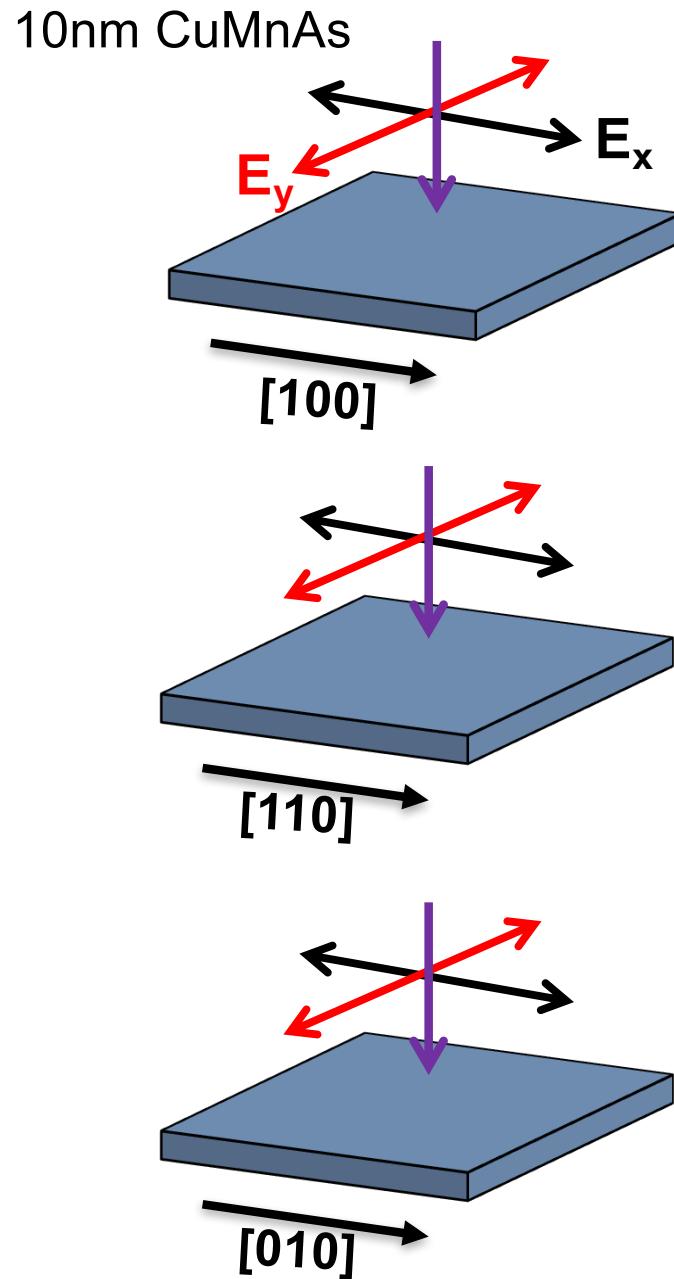


Sensitive to electric, magnetic or structural anisotropies, *including* antiferromagnetic order

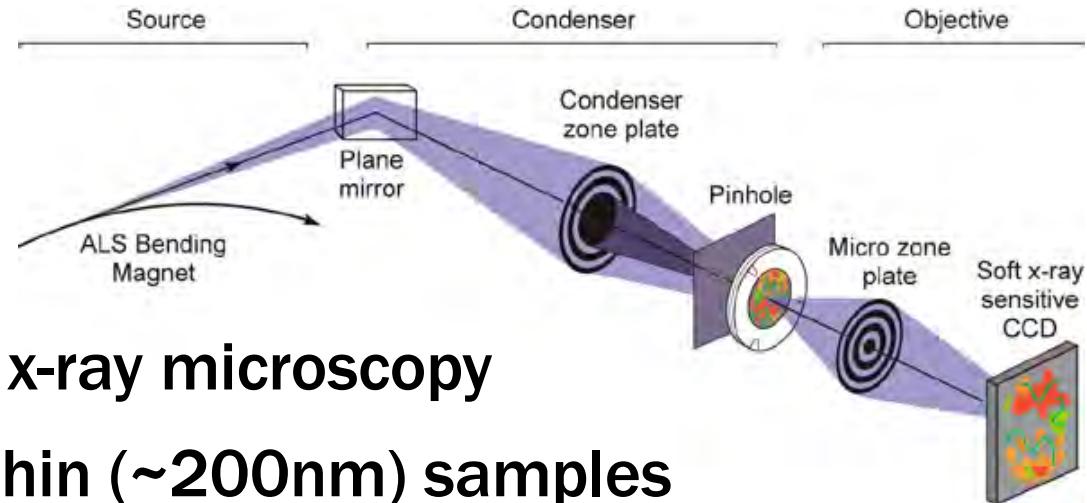
XMLD AF CuMnAs



XMLD \rightarrow thin CuMnAs layers uniaxial



Spatial resolution

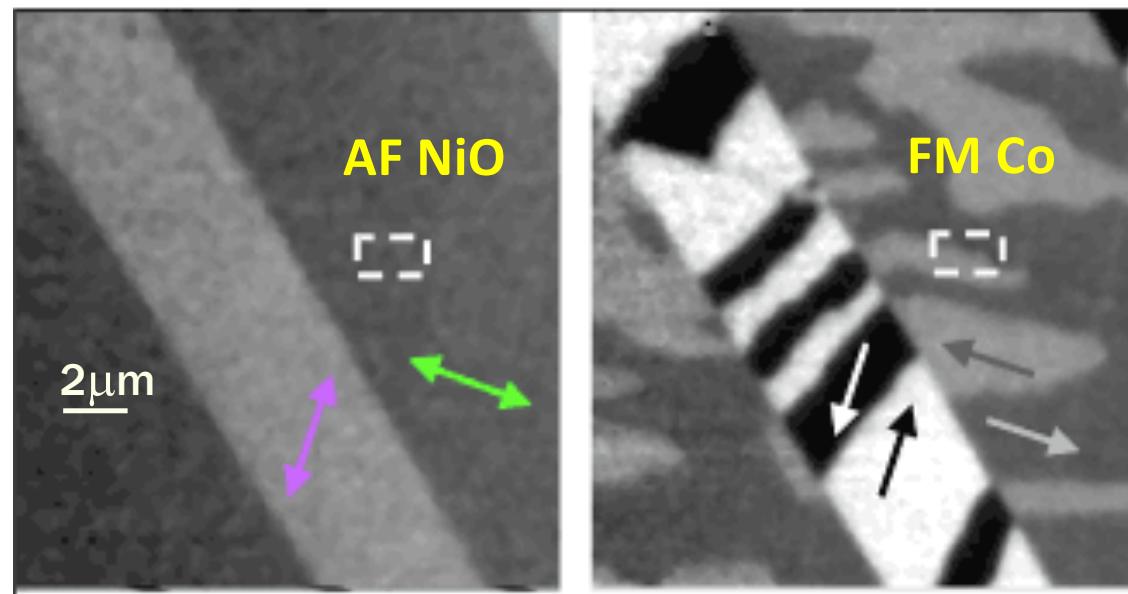
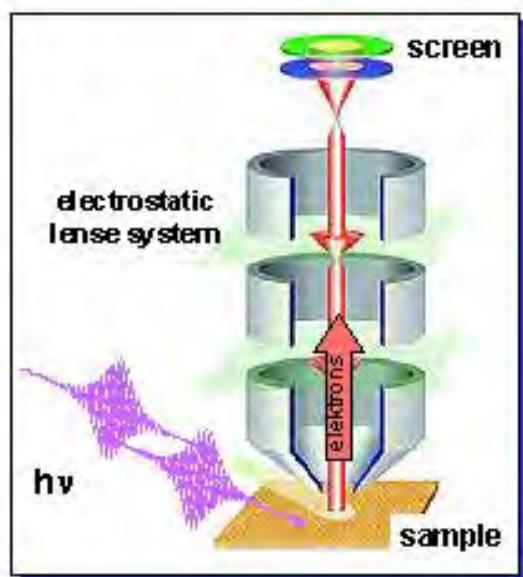


1. Scanning transmission x-ray microscopy

- powerful, but limited to thin ($\sim 200\text{nm}$) samples

2. X-ray photoelectron emission microscopy (X-PEEM)

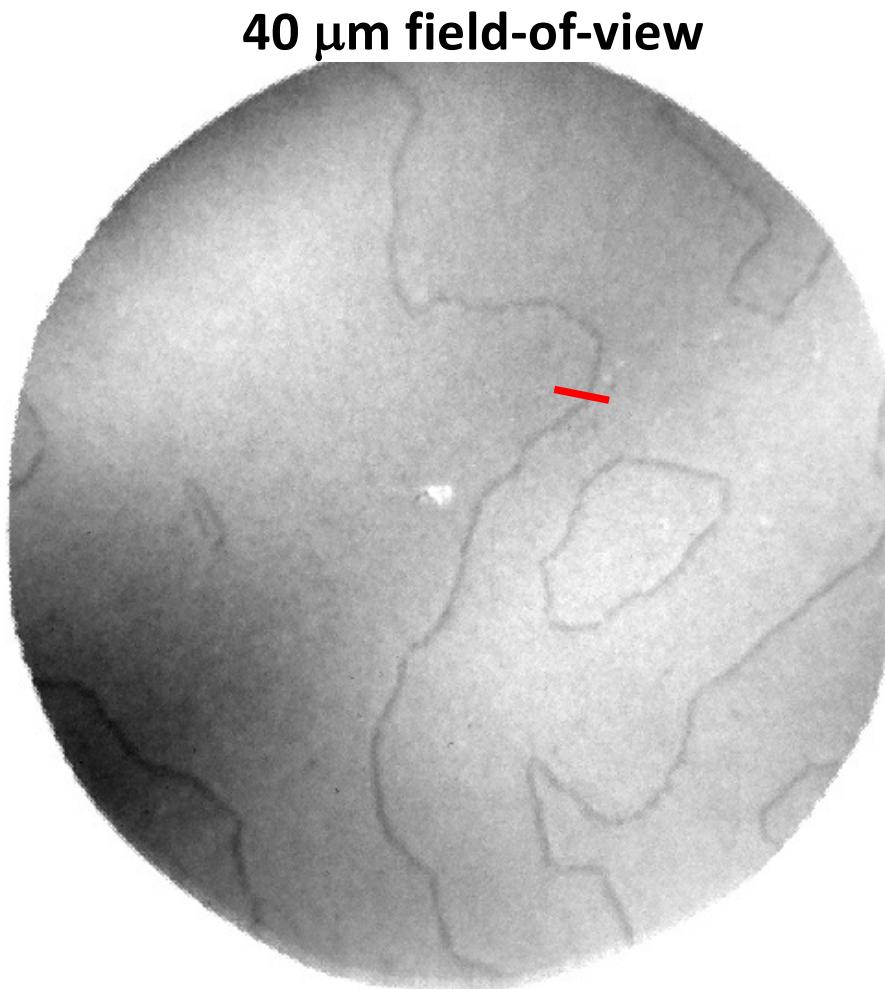
- collect and focus secondary electrons
- surface sensitive



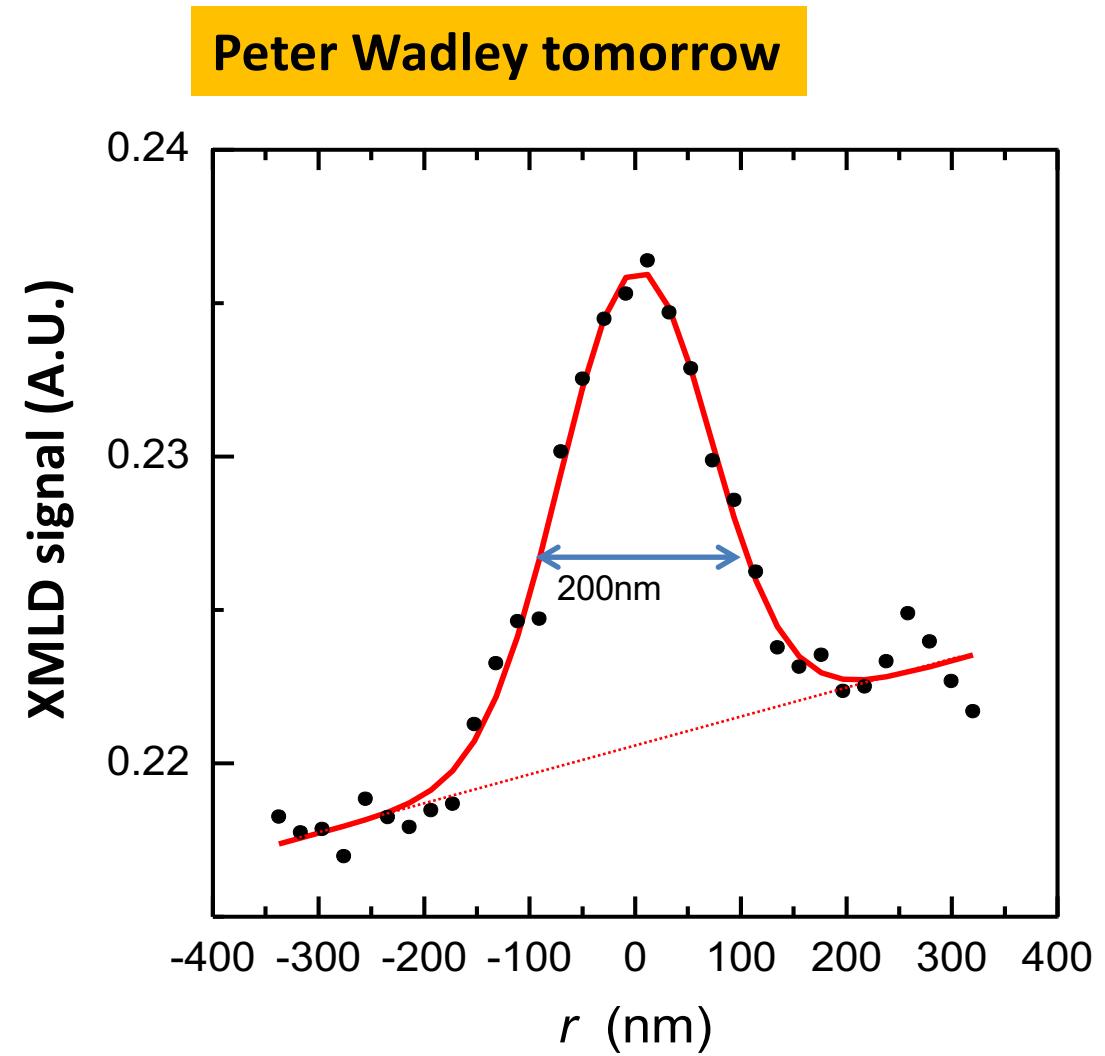
Imaging exchange coupling in AF NiO/ FM Co bilayers
Ohldag et al., PRL 86, 2878 (2001)

XMLD X-PEEM CuMnAs domains

40nm CuMnAs layer: uniaxial anisotropy with 180° domain walls



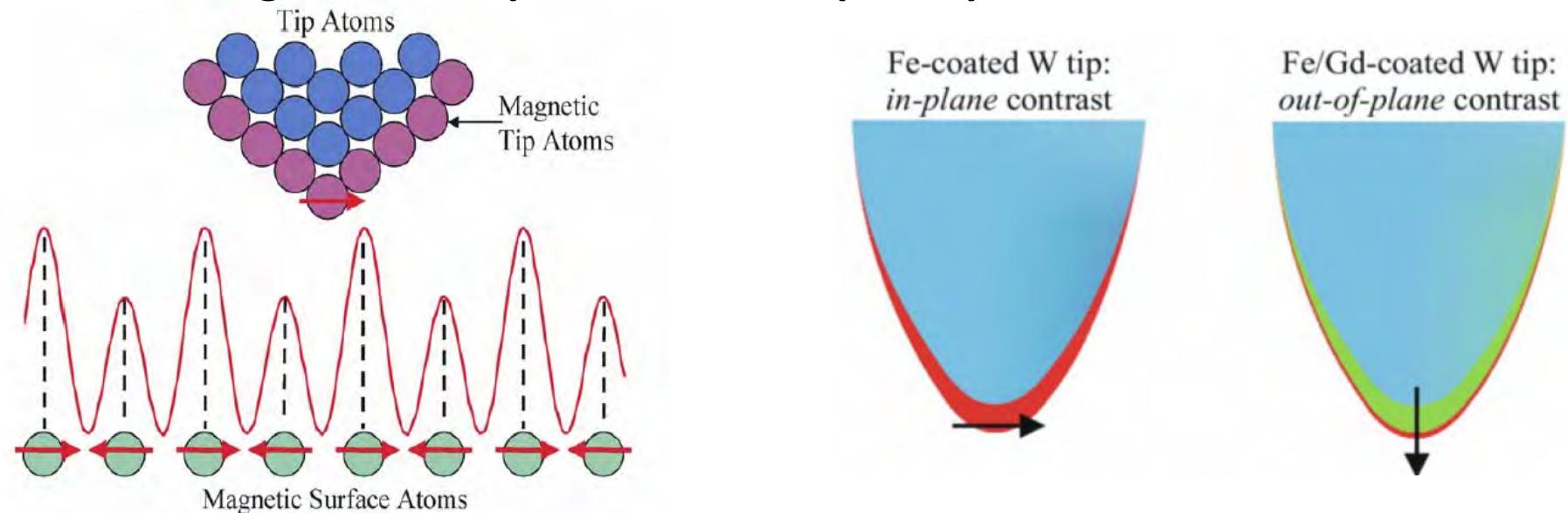
Uniaxial Domains



Domain Wall Profile

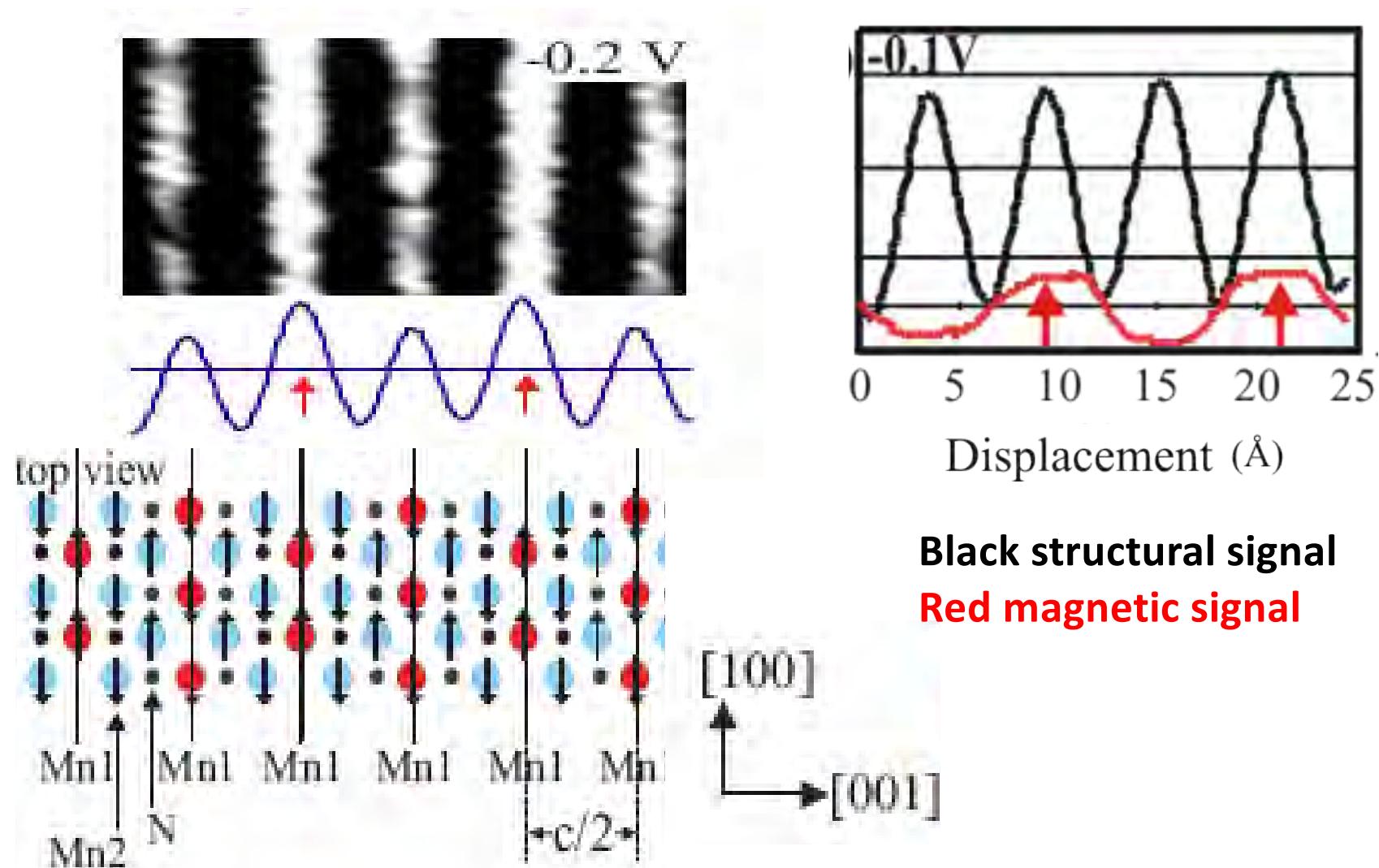
6 Spin-Polarised Scanning Tunnelling Microscopy (SP-STM)

Tunnelling current depends on local spin dependent densities of states



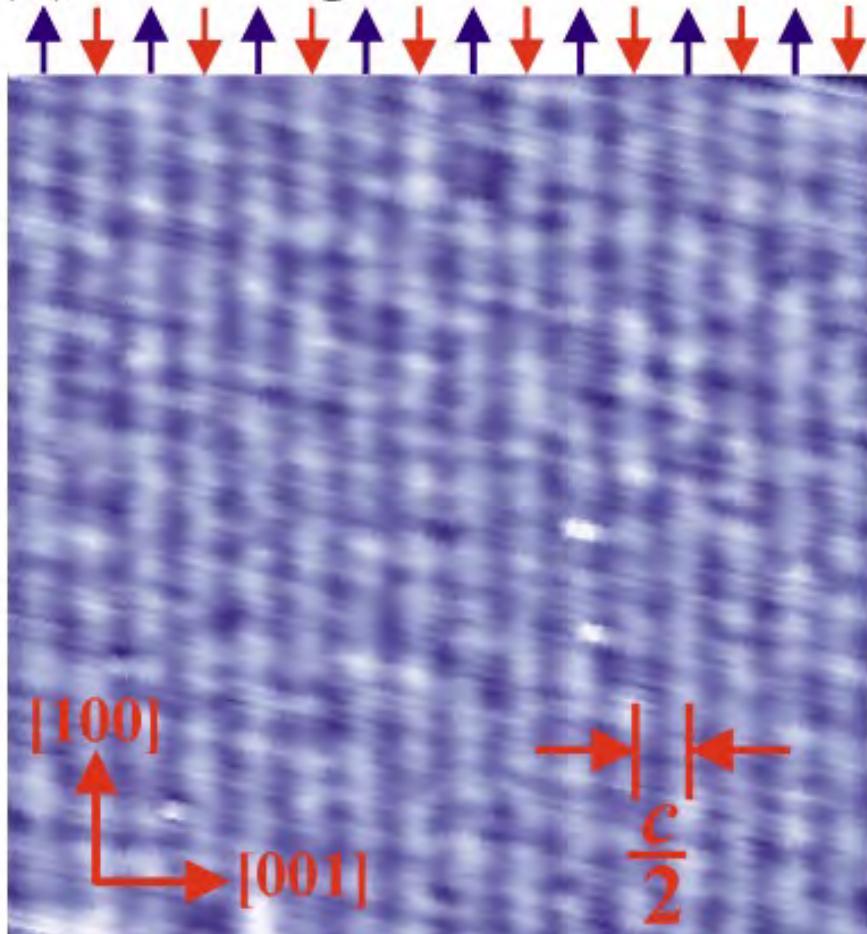
- Atomic resolution
- Sensitive to surface monolayer and few subsurface monolayers
- Need near perfect surfaces: no oxidation etc
- Usually produce sample within UHV STM system by deposition or cleavage.
- Surface properties may not reflect bulk properties: surface reconstructions, uncompensated moments etc.

SP-STM AF Mn_3N_2

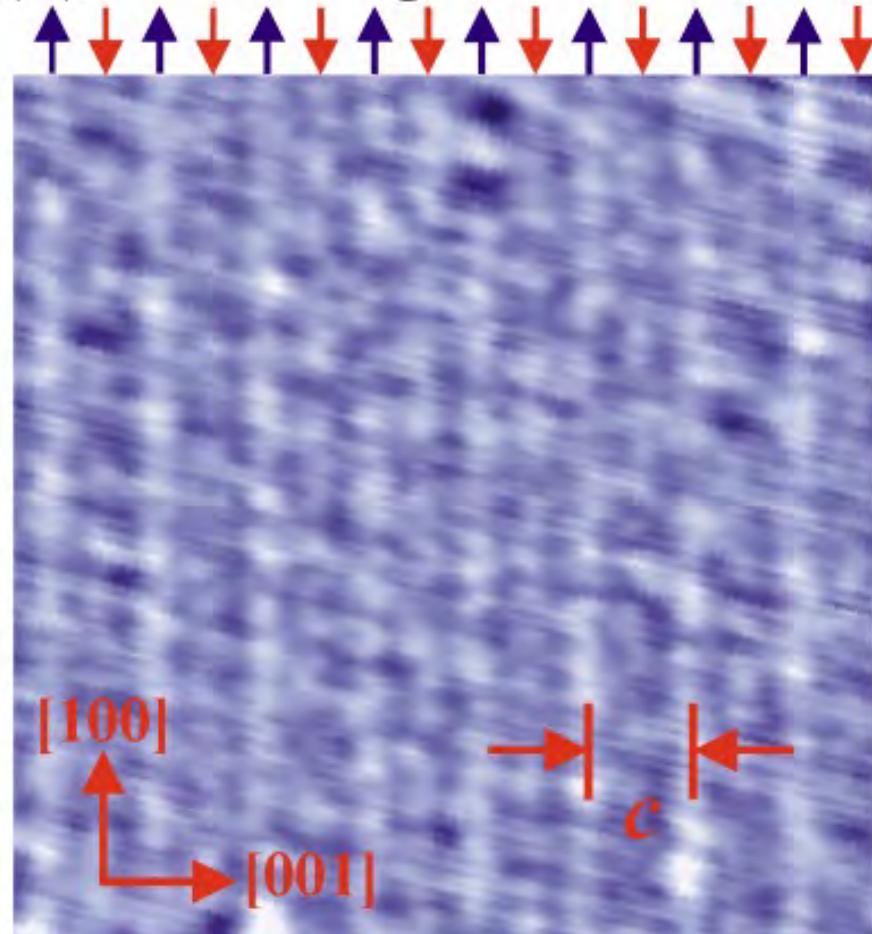


SP-STM AF Mn_3N_2

(a) no magnetic contrast



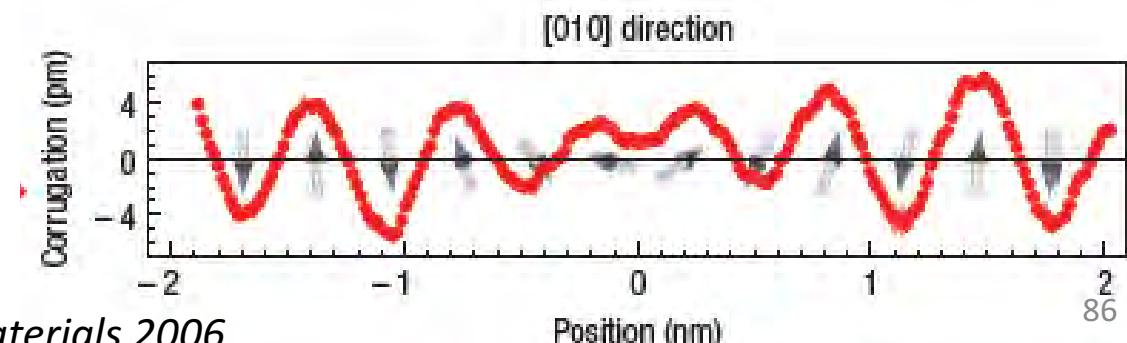
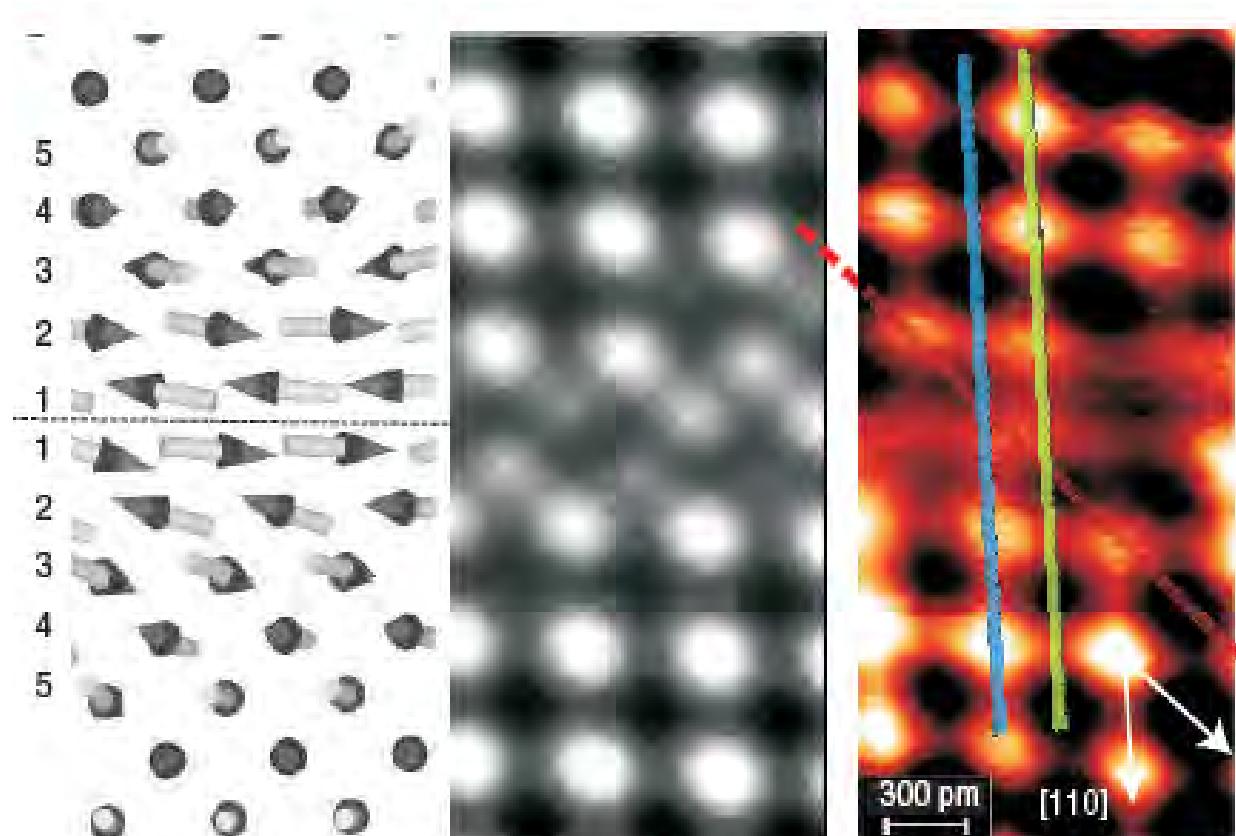
(b) with magnetic contrast



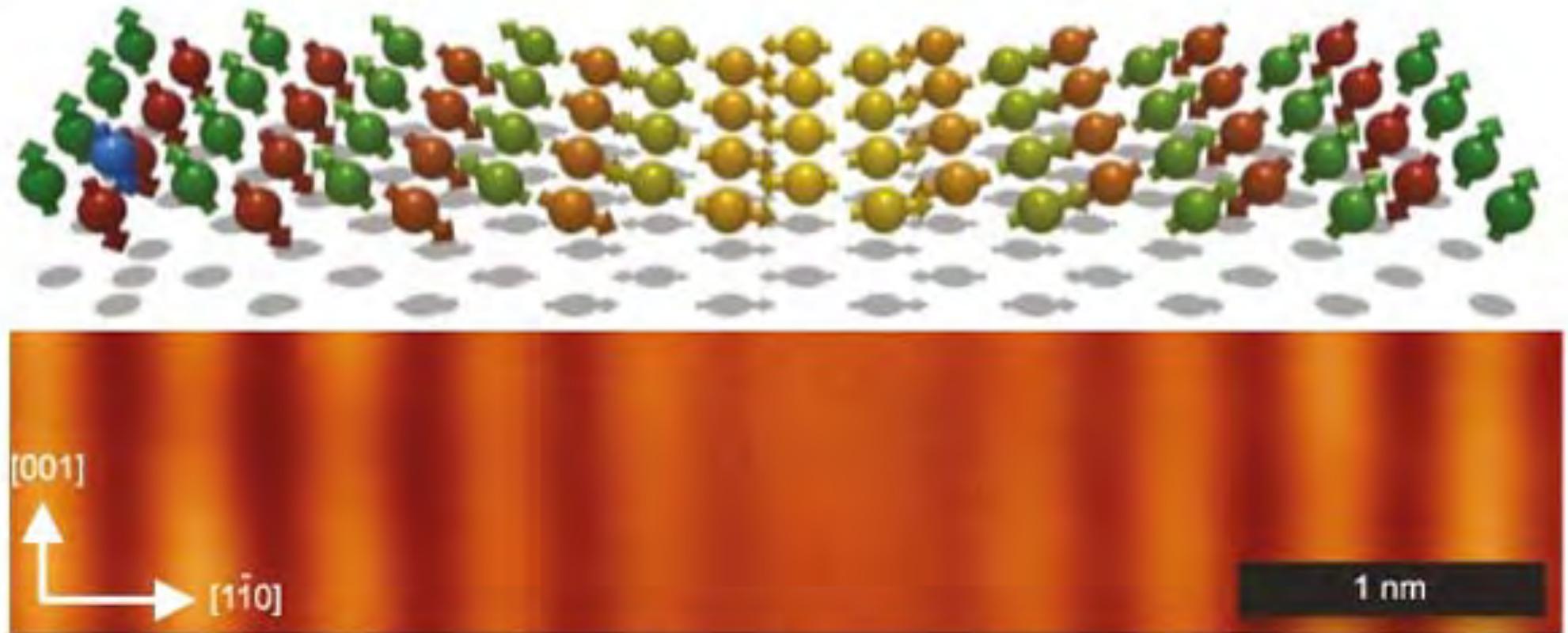
SP-STM: Atomic spin structure of an antiferromagnetic wall

Deposit Iron monolayer on
(001) surface of Tungsten.
Iron layer antiferromagnetic

Right: Simulated (grey) and
measured (colour) SP-STM
signals



SP-STM AF Mn_3N_2



(a) The spins of the antiferromagnetic manganese surface in a spin-spiral state.

(b) Out of plane SP-STM image showing the anti-parallel spins.

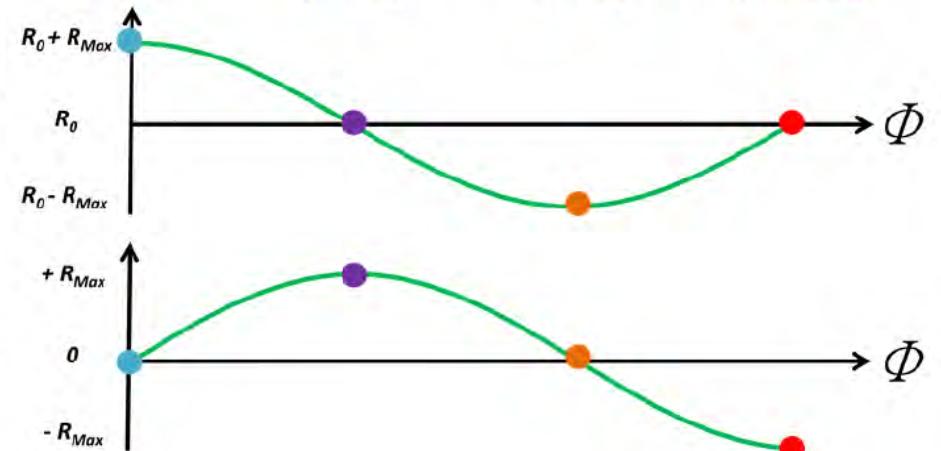
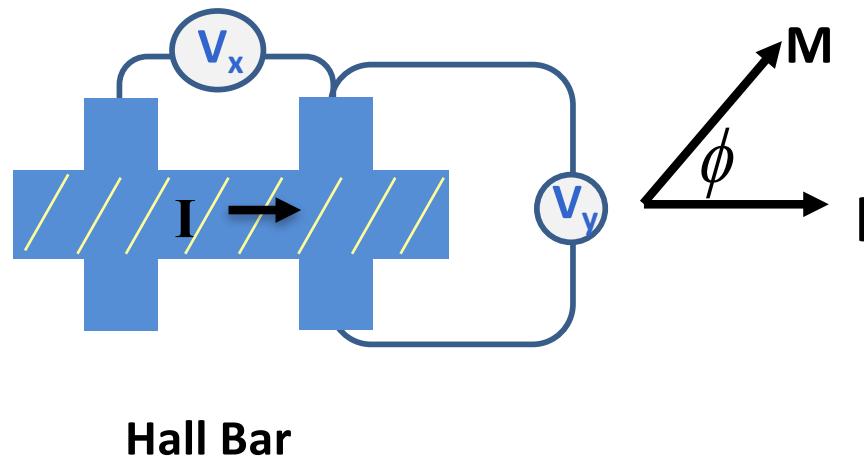
End / Jump ?

AMR ?



5.1 Anisotropic Magnetoresistance (AMR)

“Normal” Non Crystalline AMR



$$\rho_{xx} = \rho_0 + \Delta\rho \cos 2\phi \quad \rho_{xy} = \Delta\rho \sin 2\phi$$

Resistance depends on relative orientation of Magnetisation and current.

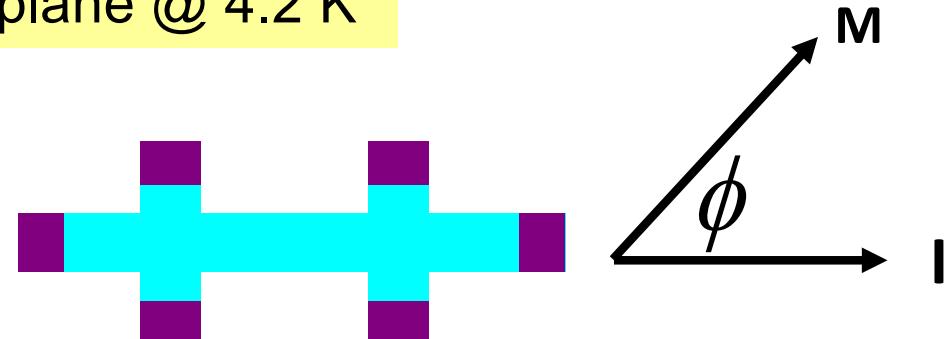
Only contribution for isotropic (e.g. polycrystalline) materials.

FM (Ga,Mn)As: Measured AMR

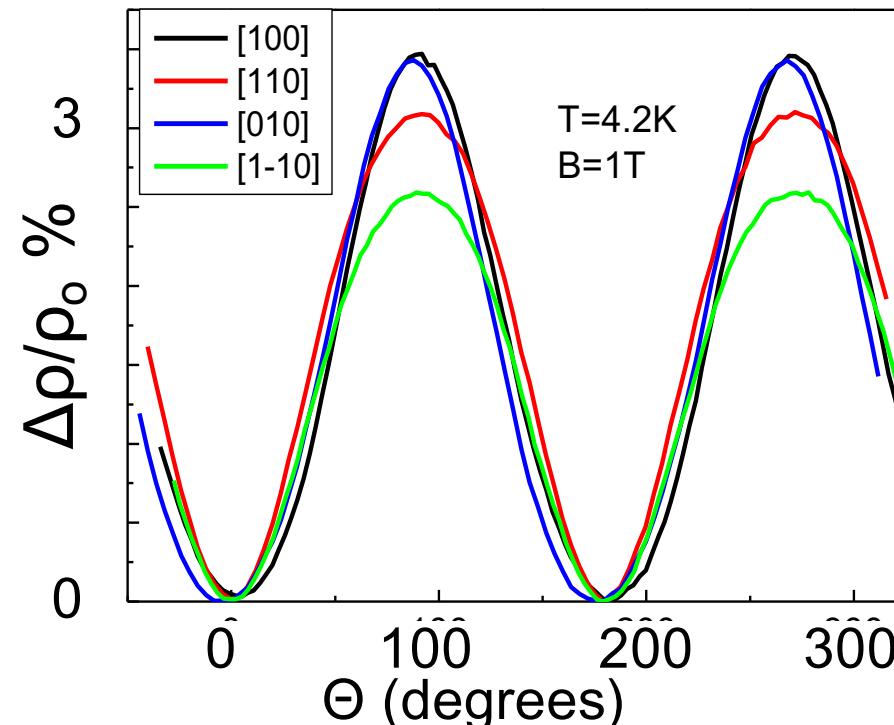
GaMnAs 5% Mn 25nm epilayer.

Fixed field of 1 Tesla rotated in plane @ 4.2 K

4 Hall Bars along 4 crystal directions

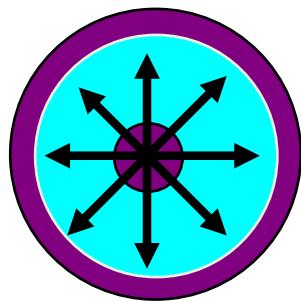


Find size of measured AMR
Depends on crystallographic orientation of the Hall bar



FM (Ga,Mn)As: Measured AMR

Corbino Disk. Current radial.
Zero non-crystalline AMR



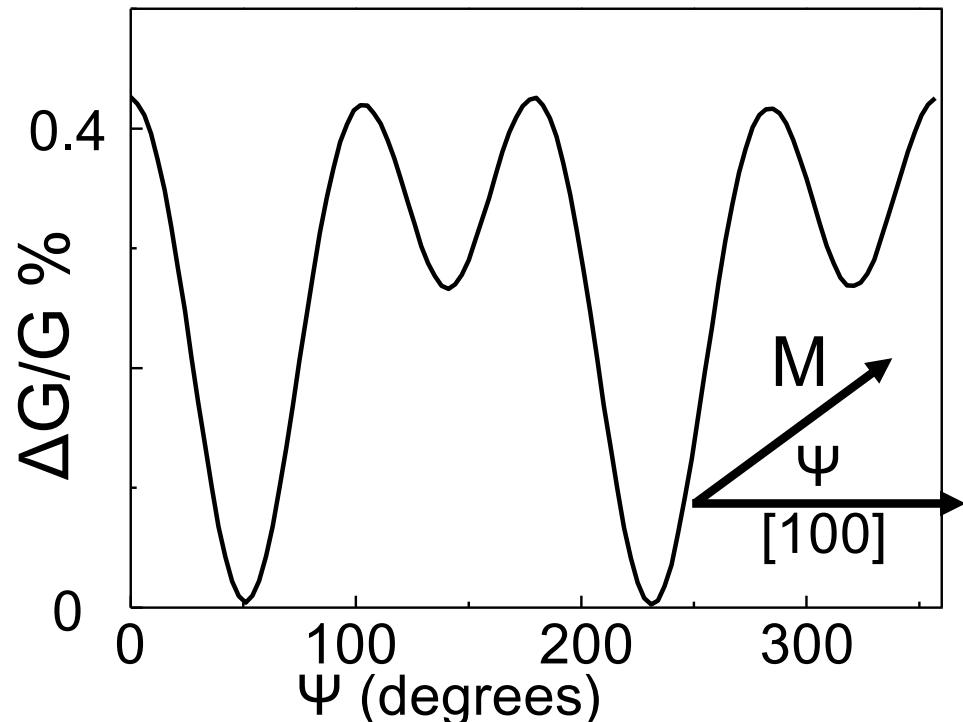
Measured AMR depends on
relative orientation of magnetisation
and crystallographic axis.

“Small” Bi-axial and uniaxial
magneto-crystalline contributions

$$\frac{\Delta\rho_{xx}}{\rho_{av}} = C_I \cos 2\phi + C_U \cos 2\psi + C_C \cos 4\psi + C_{I,C} \cos(4\psi - 2\phi) + C_{I,U} \sin(6\psi - 2\phi) + \dots$$

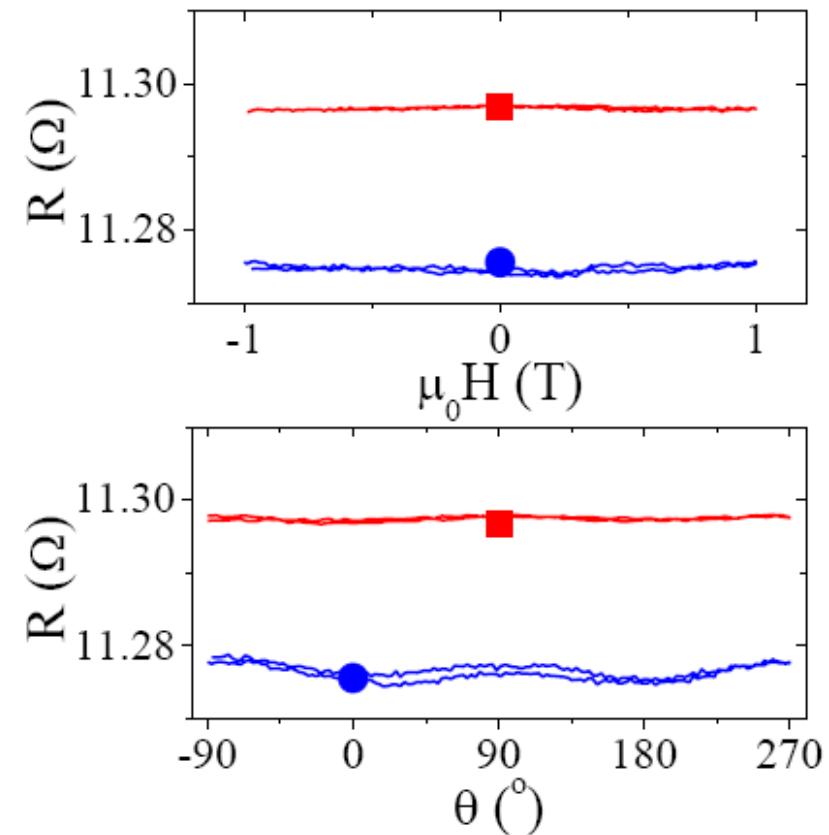
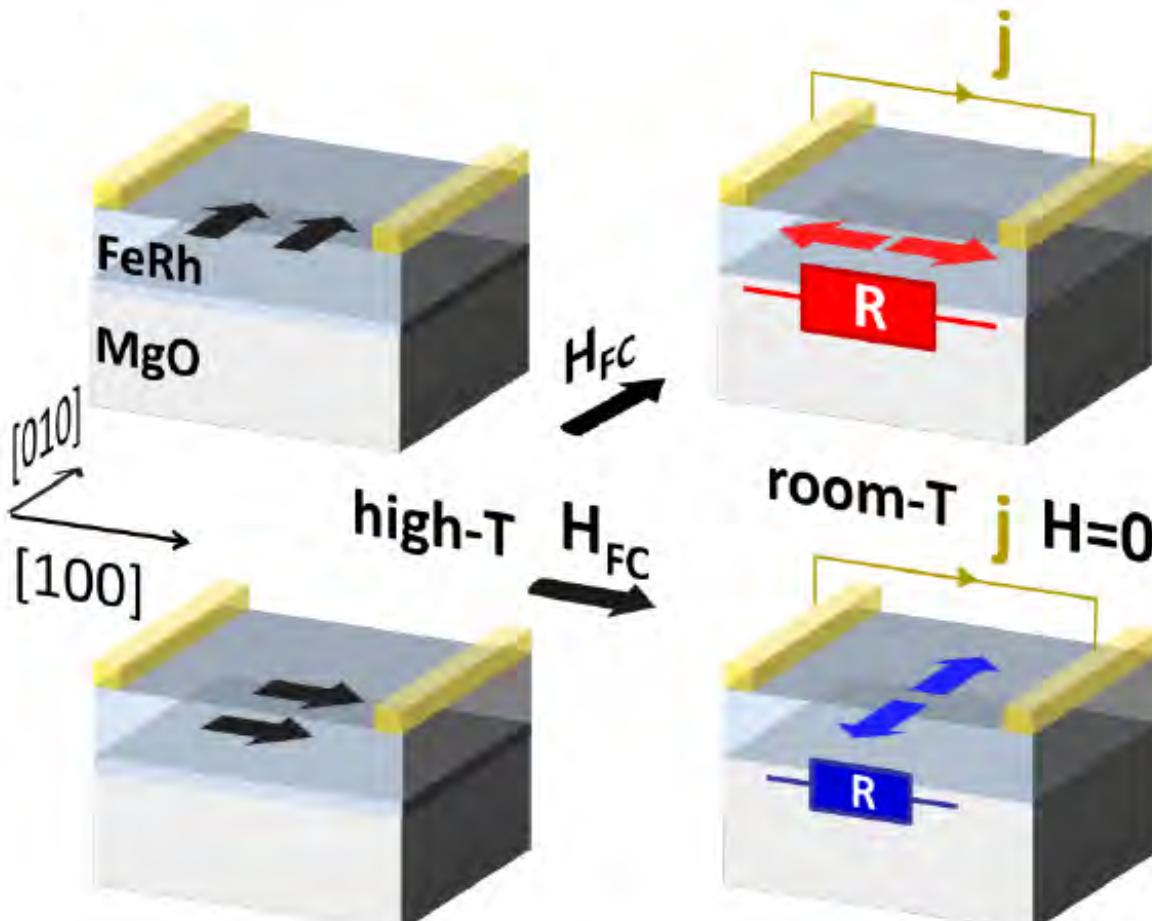
$$\frac{\Delta\rho_{xy}}{\rho_{av}} = C_I \sin 2\phi + C_{I,C} \sin(4\psi - 2\phi) + C_{I,U} \cos(6\psi - 2\phi) + \dots$$

GaMnAs 5% Mn 25nm epilayer.
1 Tesla field rotated in plane @ 4.2 K



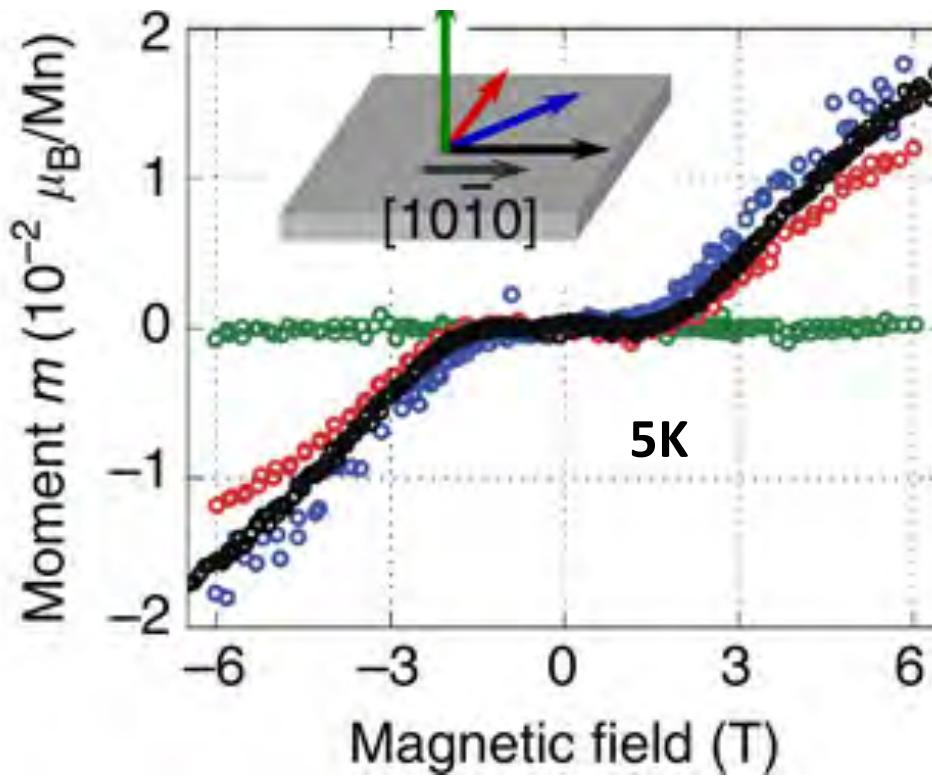
Antiferromagnetic AMR: FeRh

FeRh: AF to FM transition at $\sim 400\text{K}$.

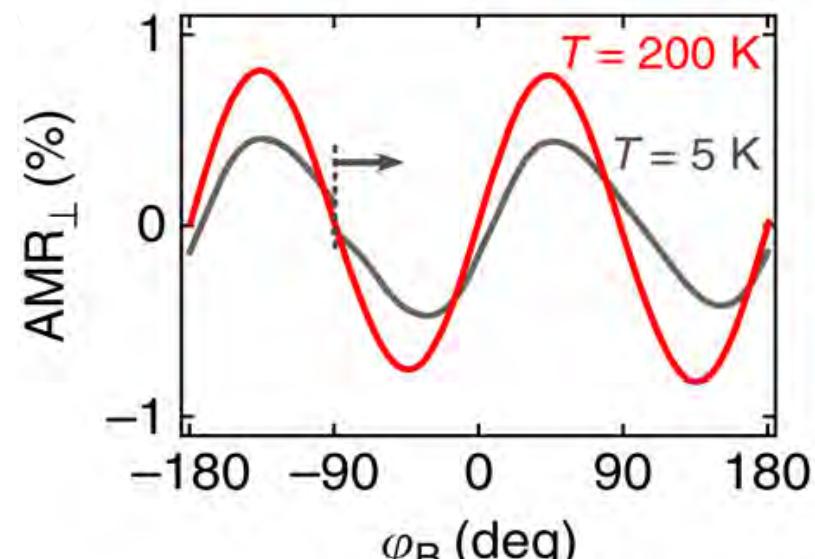


Cool in field through the FM to AF transition temperature:
Spins flop into direction perpendicular to applied field.

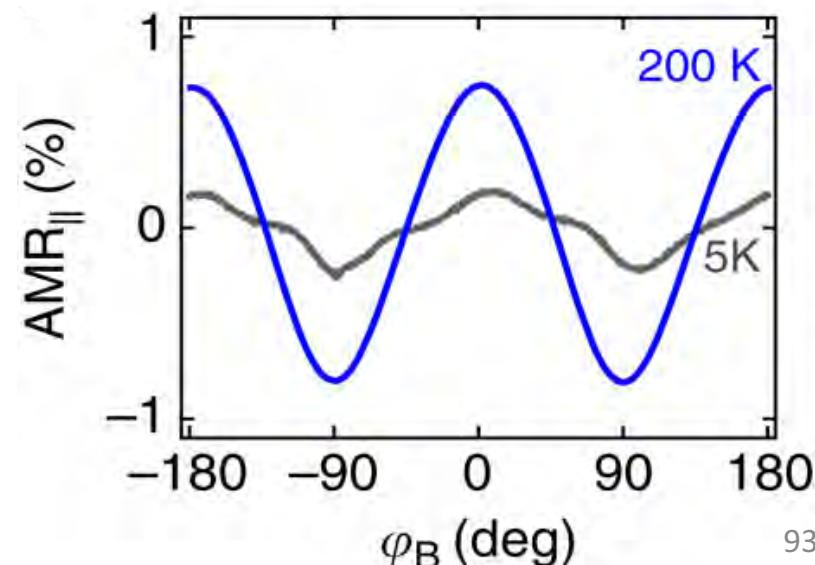
Antiferromagnetic AMR: α -MnTe



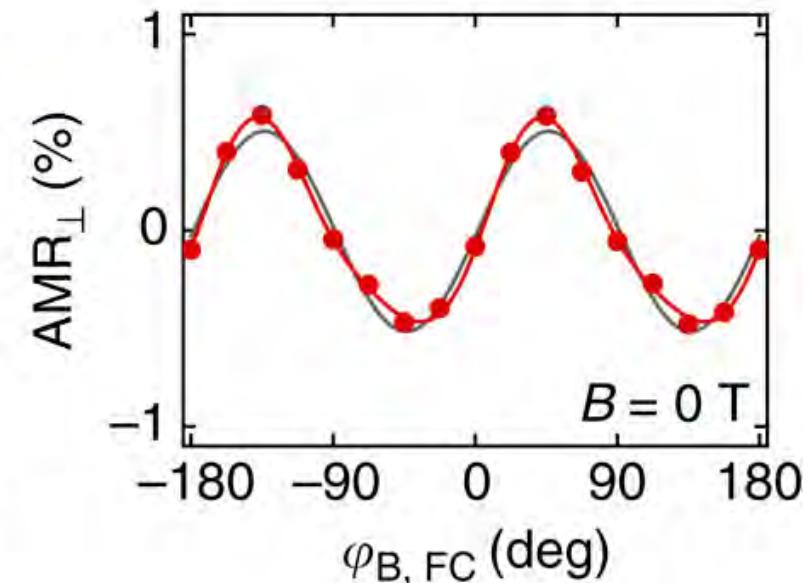
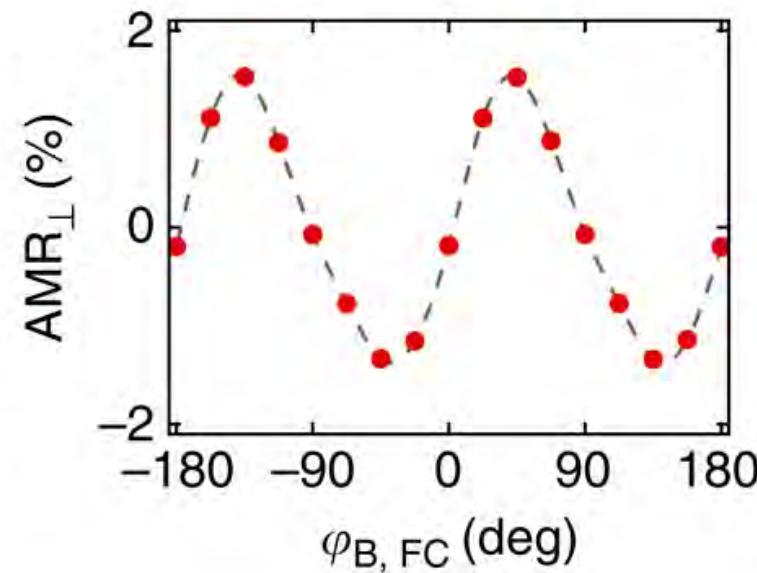
Applied field leads to Spin flop
plus domain modification



2 Tesla field rotated in plane

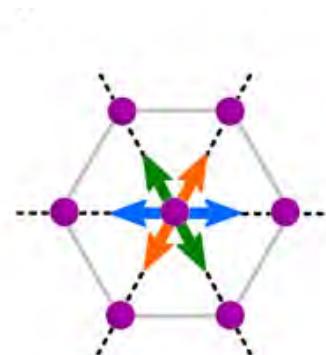
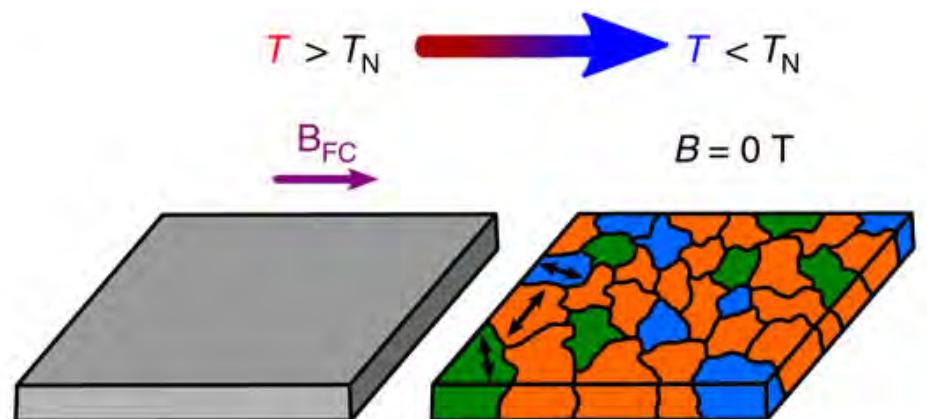


Antiferromagnetic AMR: α -MnTe



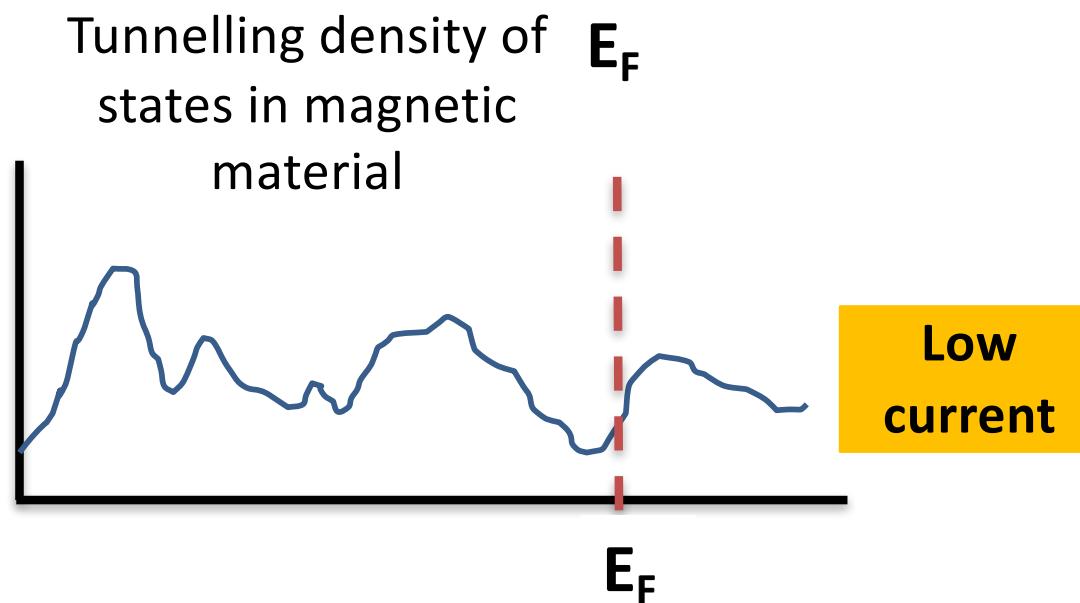
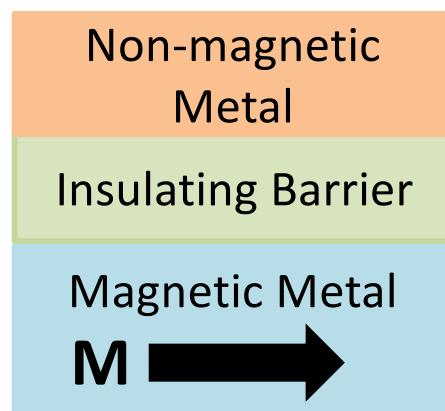
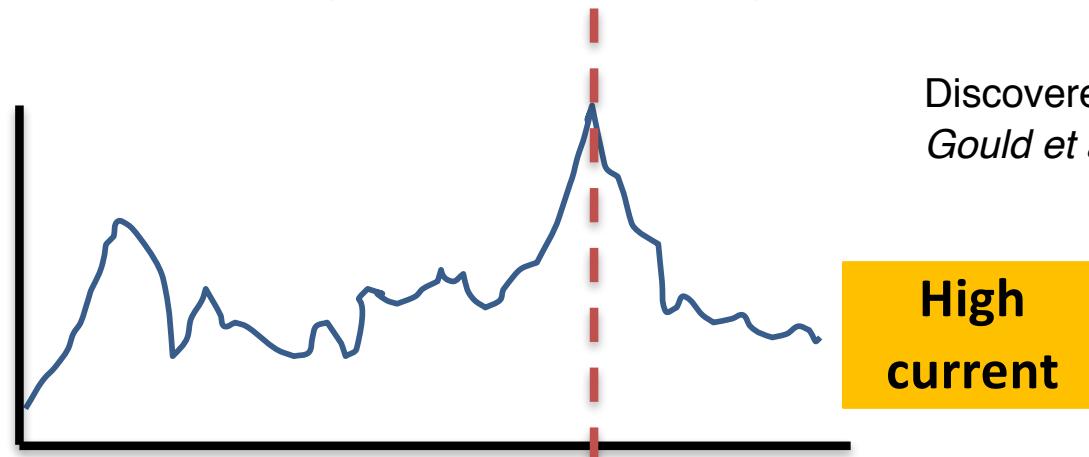
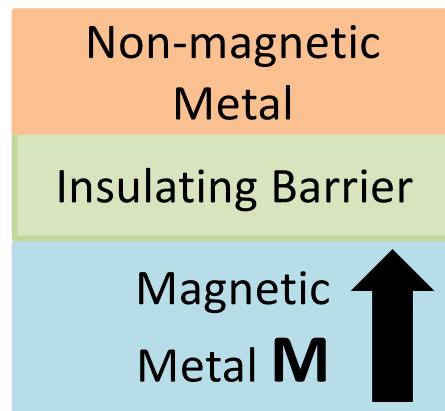
Cooled in 2 Tesla field from 350K
to 200K then field left on.

Cooled in 2 Tesla field from 350K
to 200K then field removed.



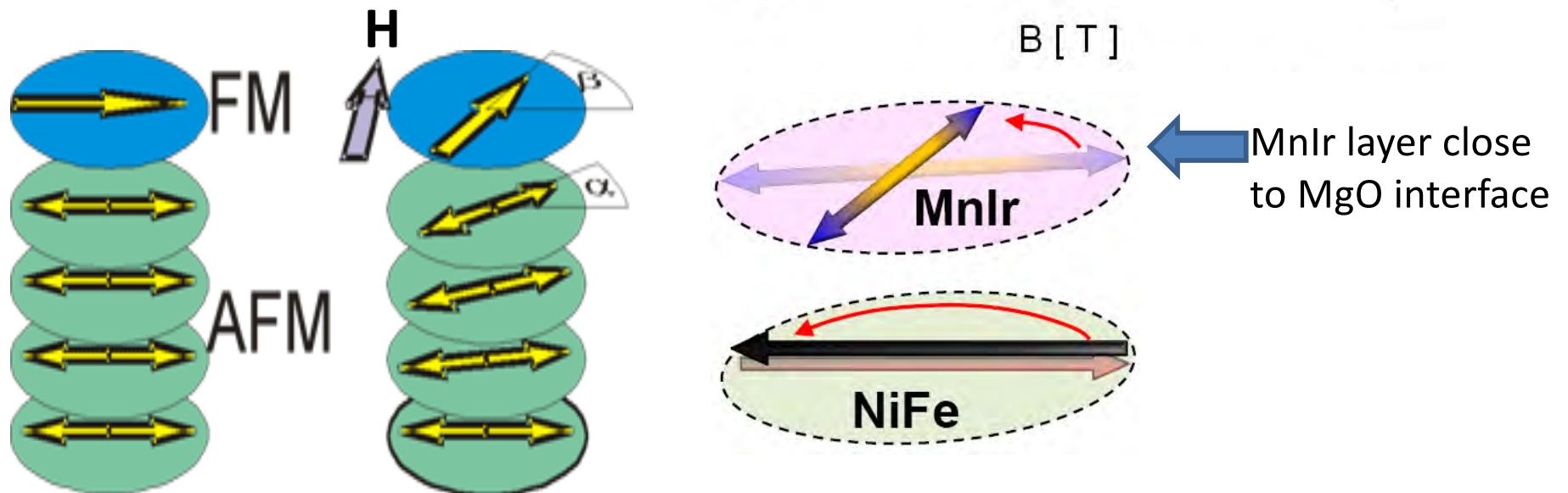
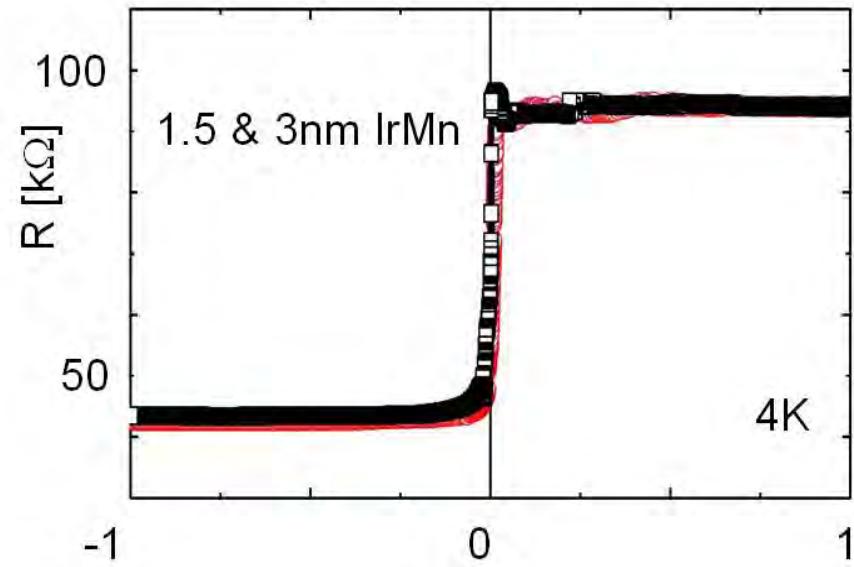
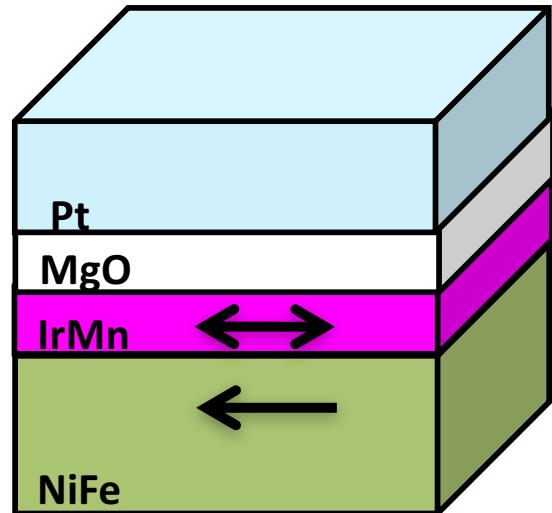
5.2 Tunnelling Anisotropic Magnetoresistance (TAMR)

Single magnetic contact. Tunnelling density of states depends on relative orientation of magnetisation and crystal axes due to spin-orbit interaction.

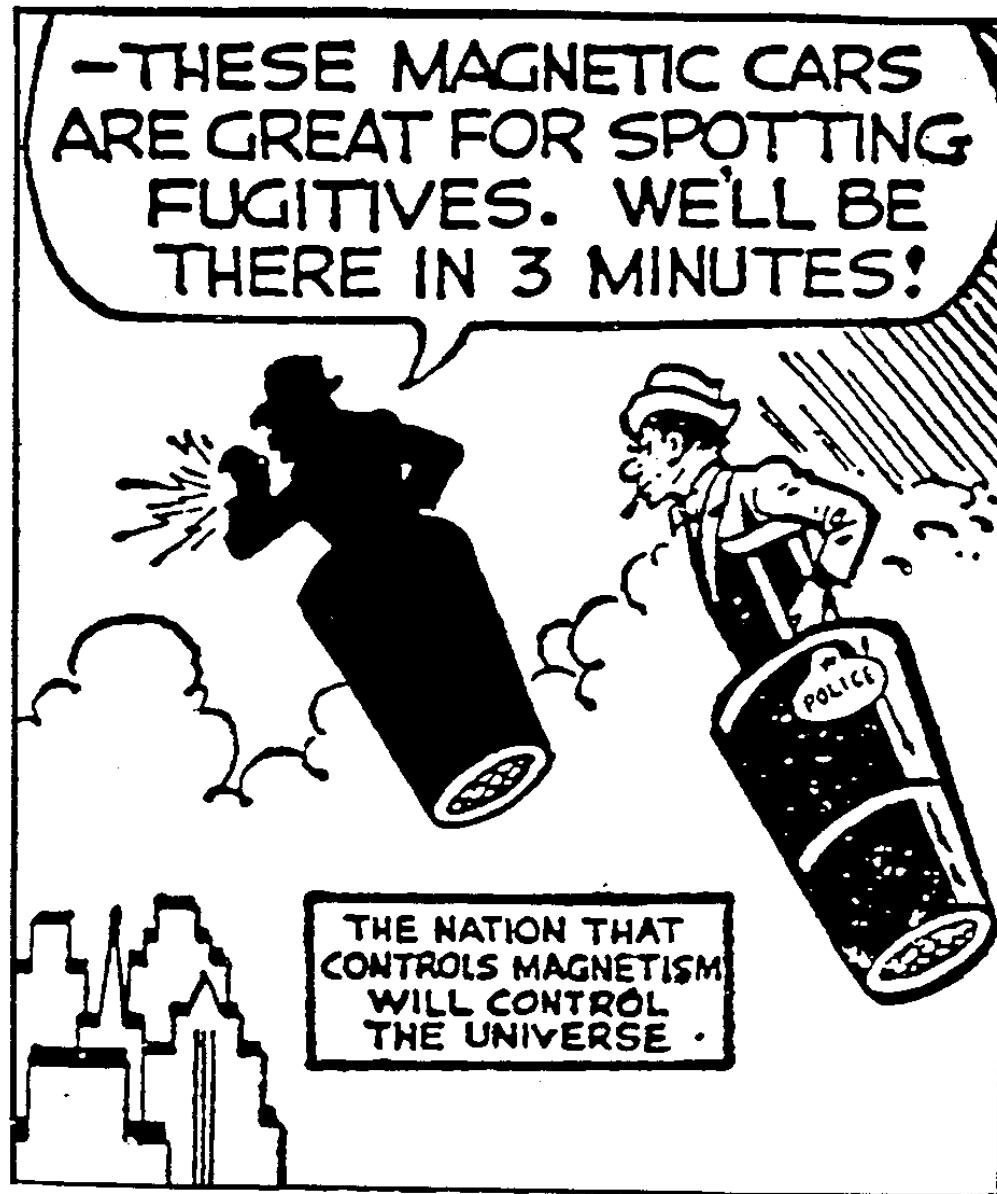


Antiferromagnetic TAMR

Exchange bias used to control AFM moments



The End



The Detective Dick Tracy (1930s)